Crystal Growth: Physics, Technology and Modeling

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Lecture 4. Epitaxy of semiconductor structures

http://www.unipress.waw.pl/~stach/cg-2024-25

Epitaxy of semiconductor structures

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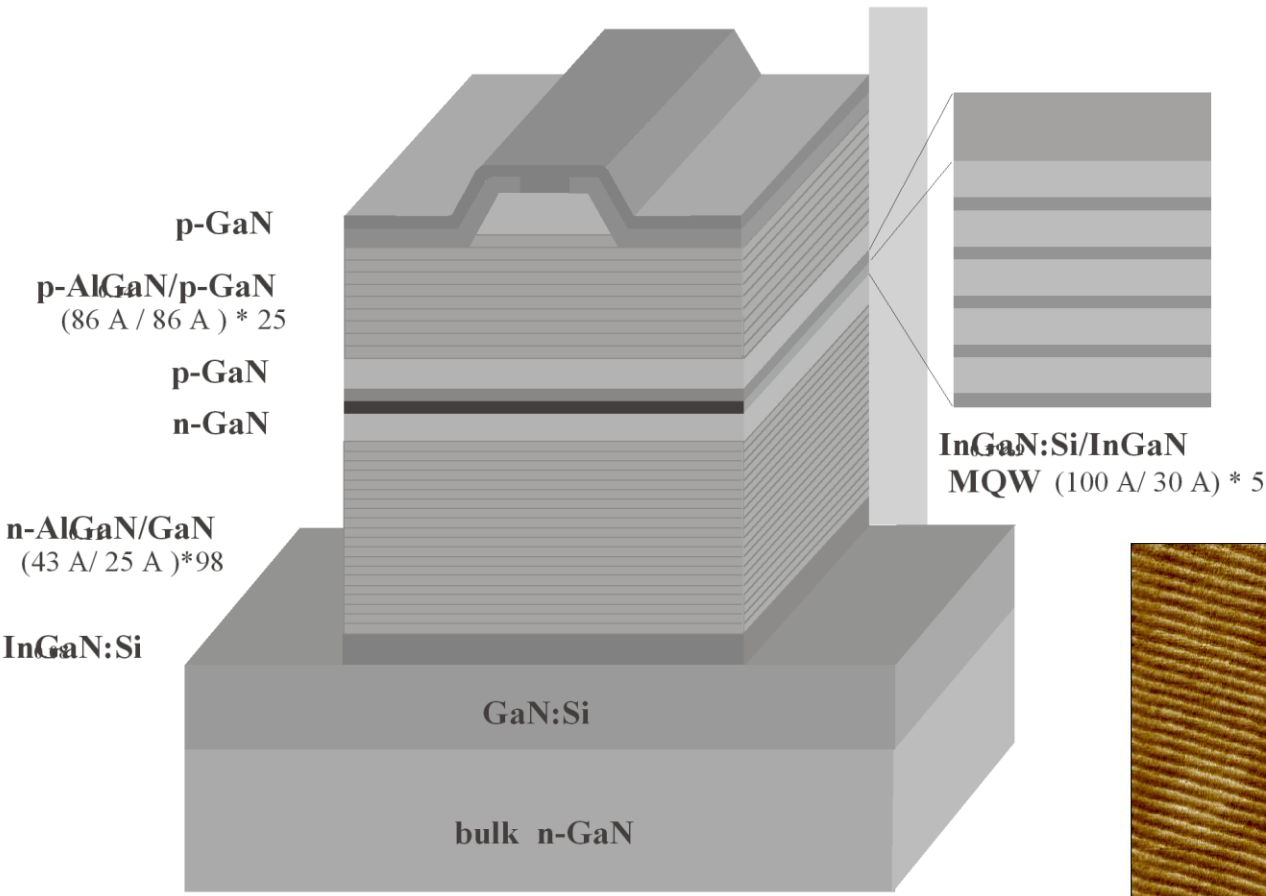
Instytut Wysokich Ciśnień PAN

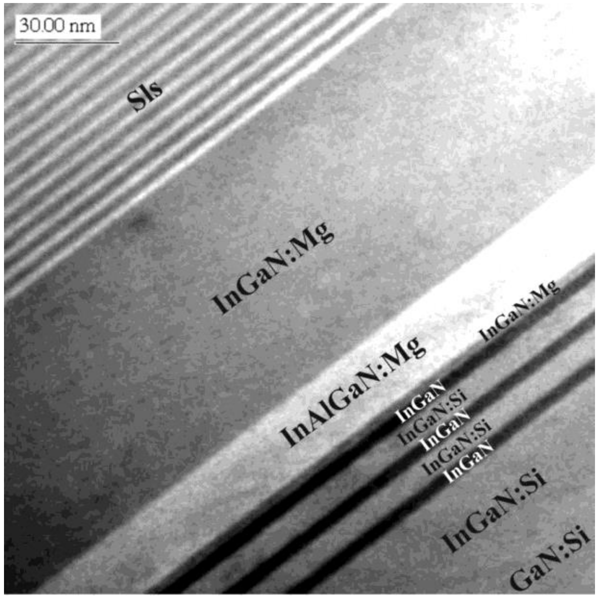
And TopGaN

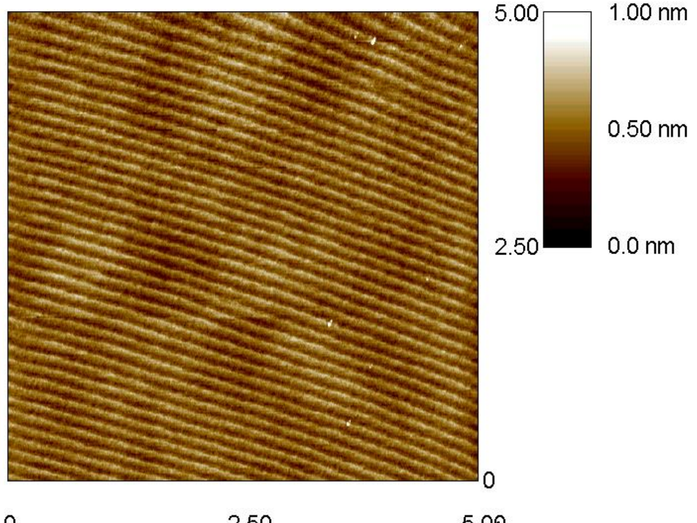
Outline

1. Epitaxial reactors
2. Substrate off-cut
3. Problem of lattice mismatch
4. Non-lateral growth
5. Atom incorporation efficiency
6. Examples of MOVPE growth: InGaN Quantum Wells

What is to be grown? For example, blue LD.







What we need from epitaxy:

1. Proper chemical composition (including doping)
2. No defects (dislocations, vacancies, etc)
3. Perfect interfaces
4. Perfect uniformity across the wafers

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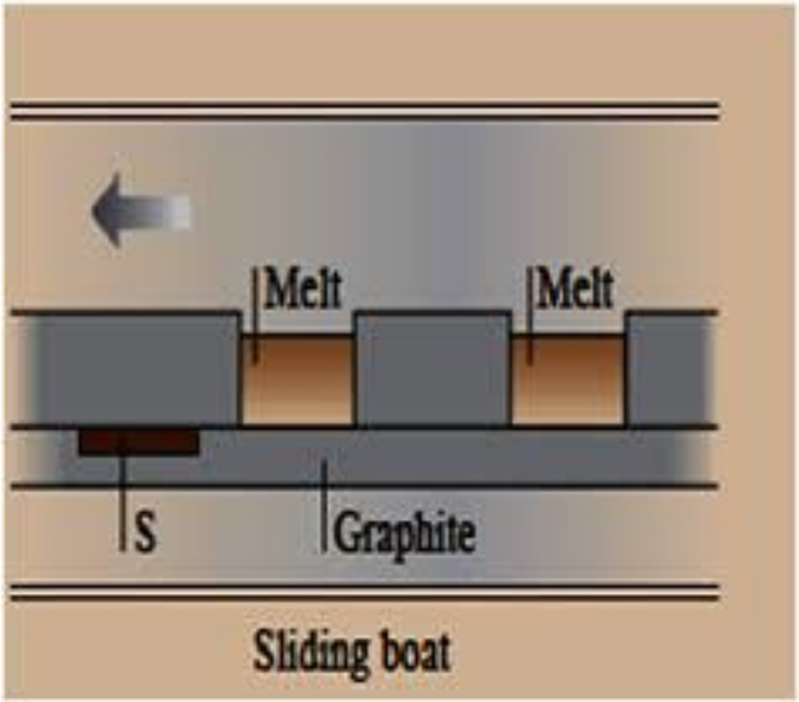
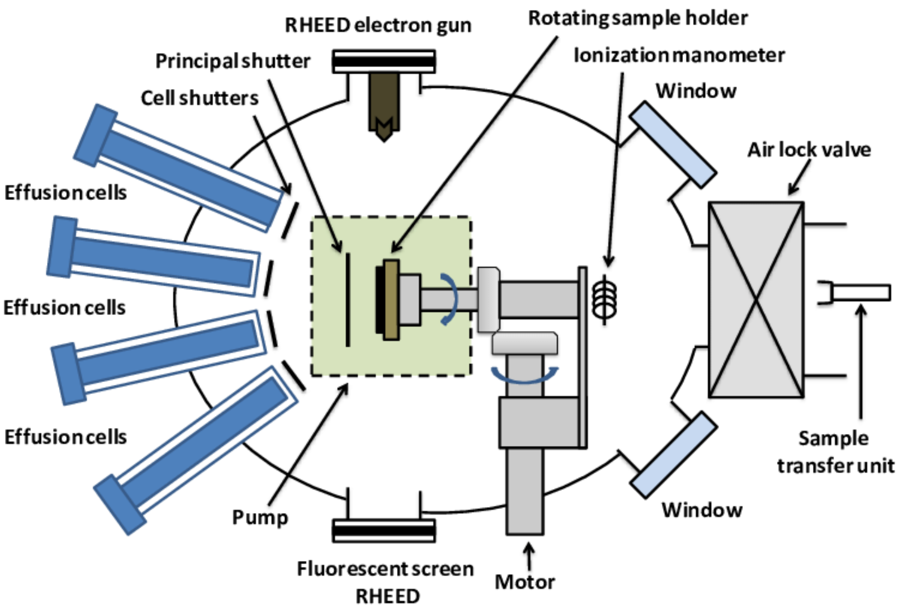
Liquid Phase Epitaxy (LPE)

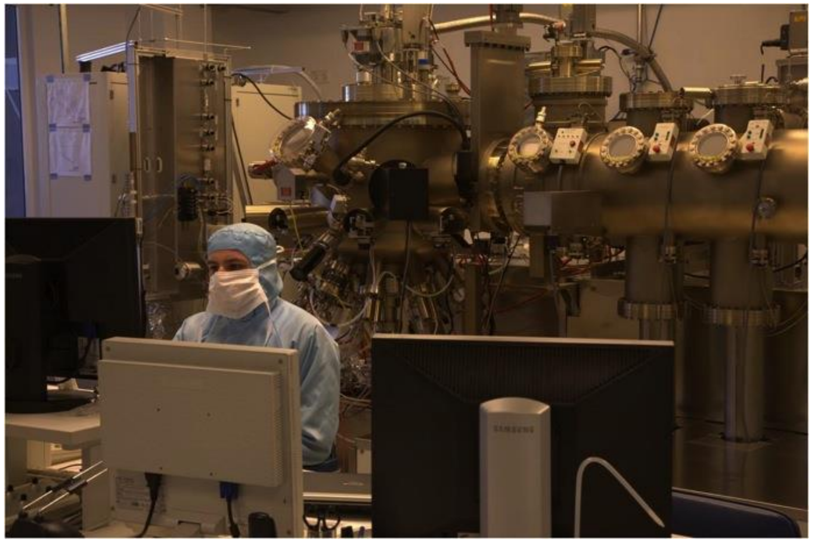


Technology abandoned by industry

- **Only small wafers**
- **Not perfect interfaces**
- **Not highest purity**

Molecular Beam Epitaxy (MBE)

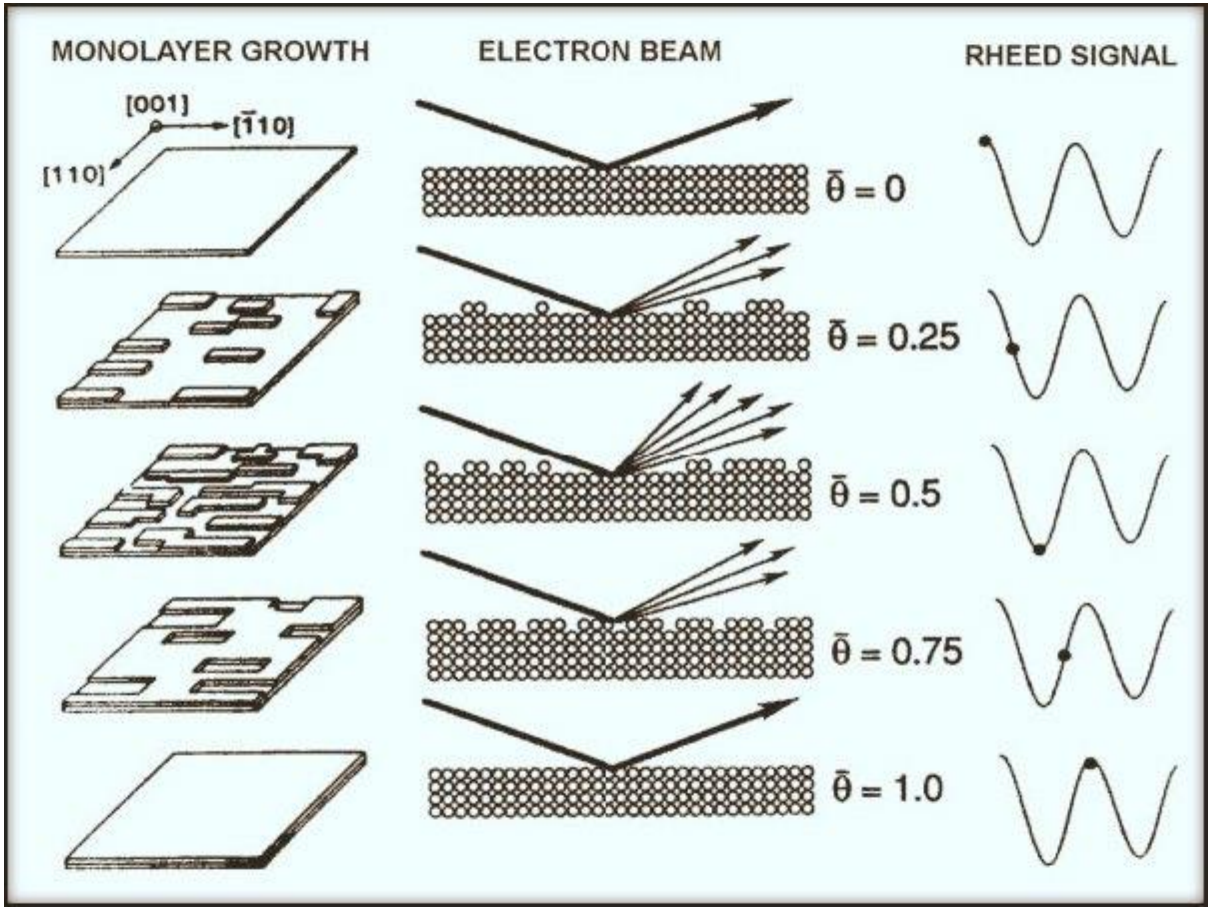




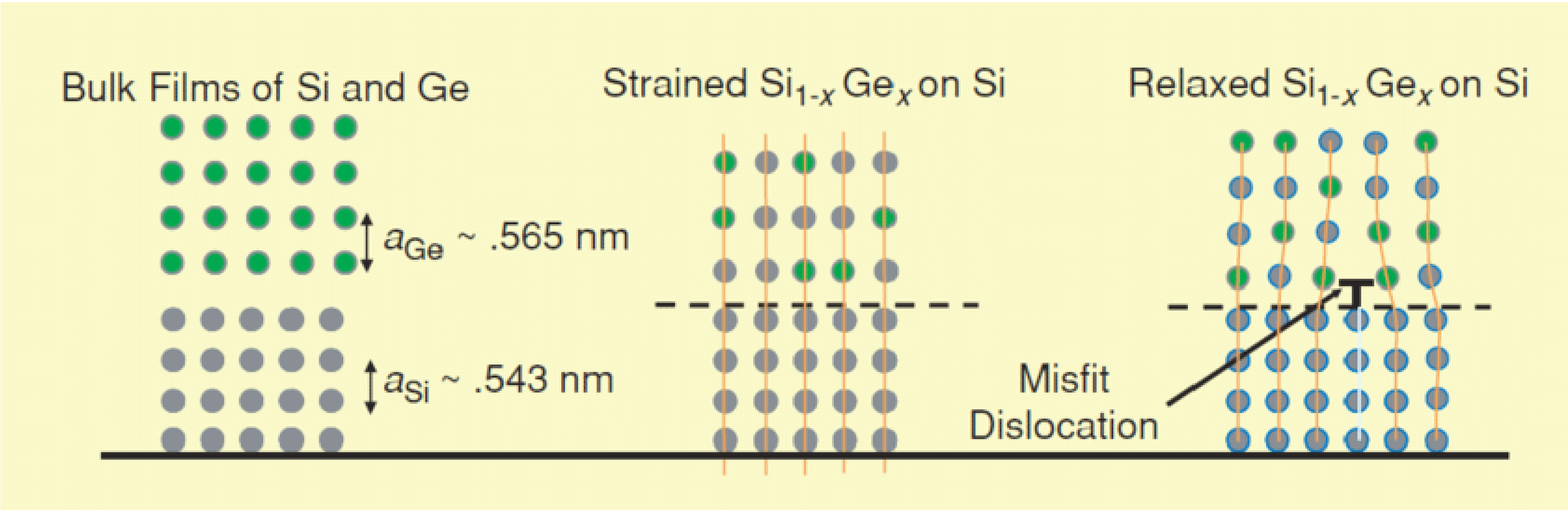
+ High purity, perfect interfaces, no hydrogen in the system

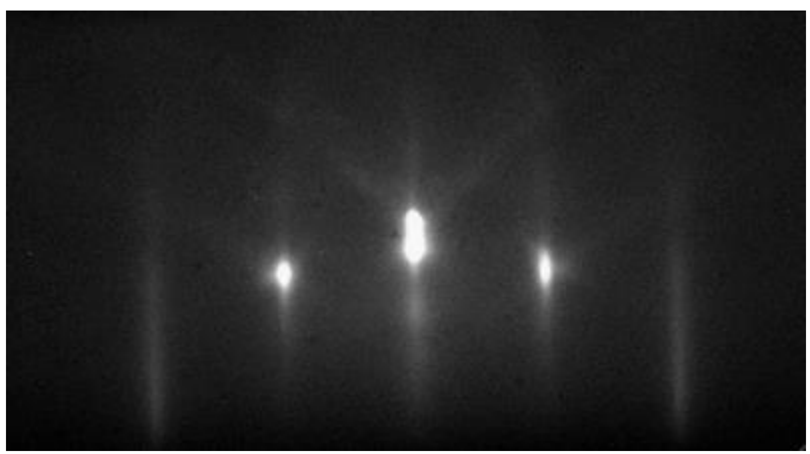
- Costs, no multiwafer machines (one six-inch wafer)

RHEED growth rate and roughness control



RHEED: lattice relaxation control

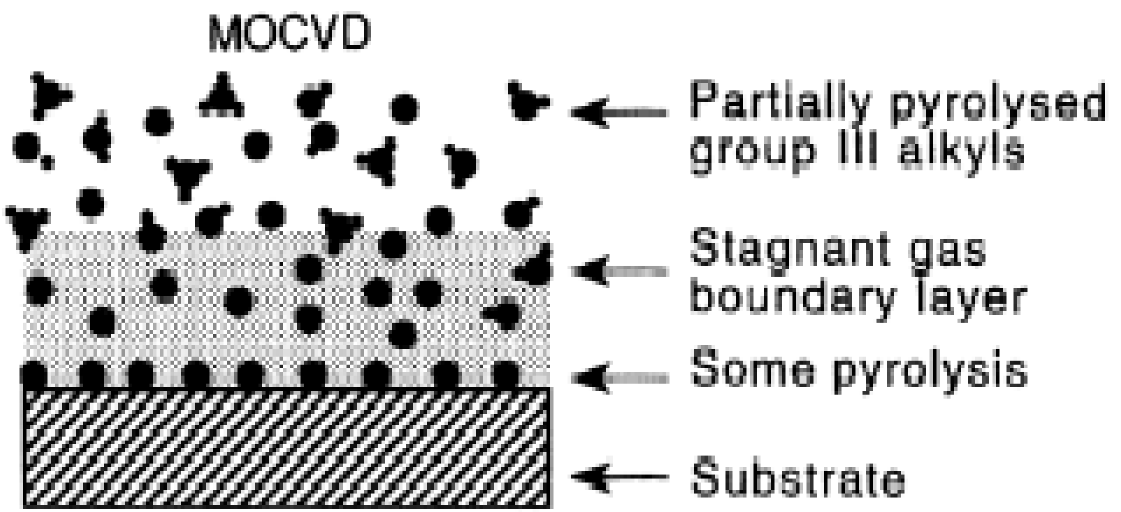




Metalorganic Chemical Vapour Phase Epitaxy MOVPE

A(CH3)3+BH3->AB+3CH4

A= Ga, In, Al, B=N, As, P

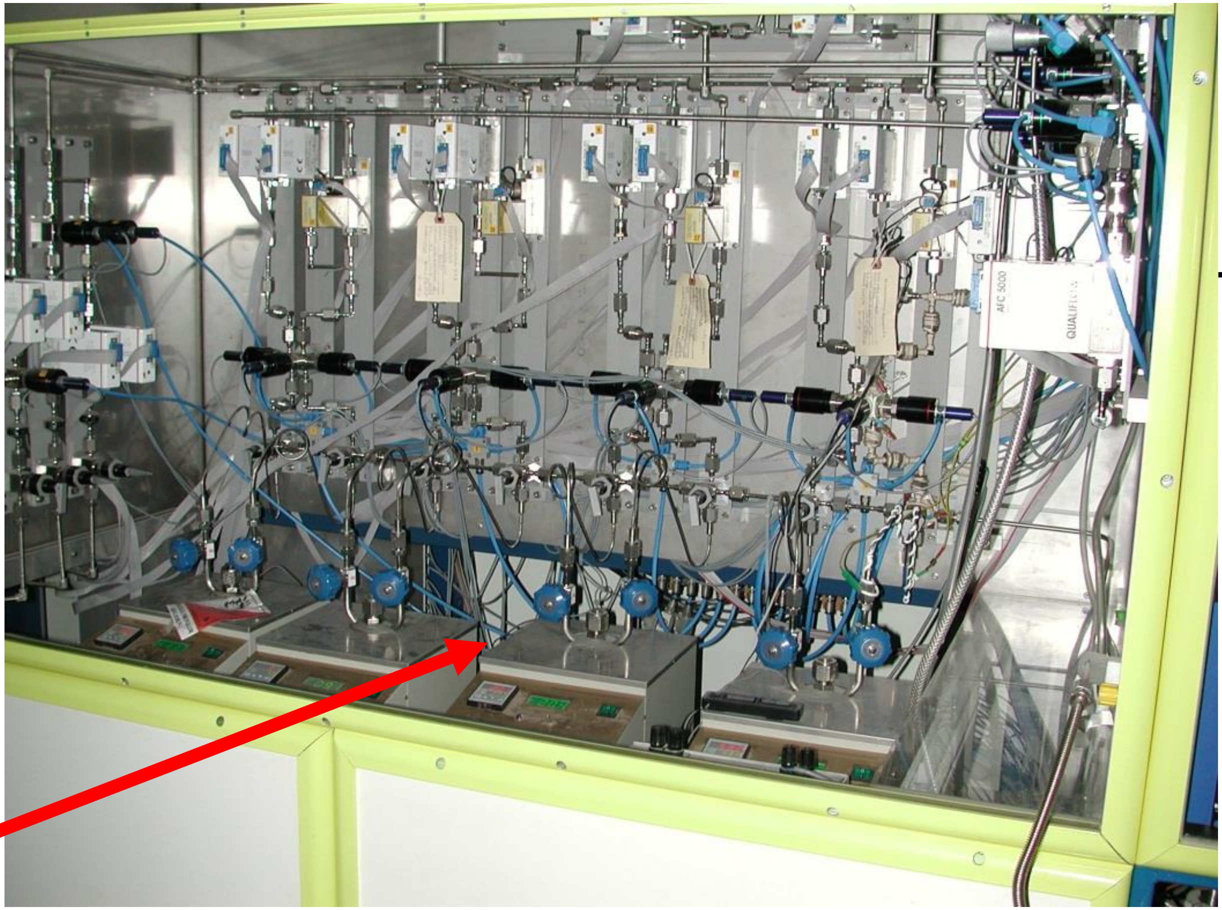


Gas panel, MOVPE, AlGaInN:Mg, Si

Purifiers

Carrier gas
N2, H2

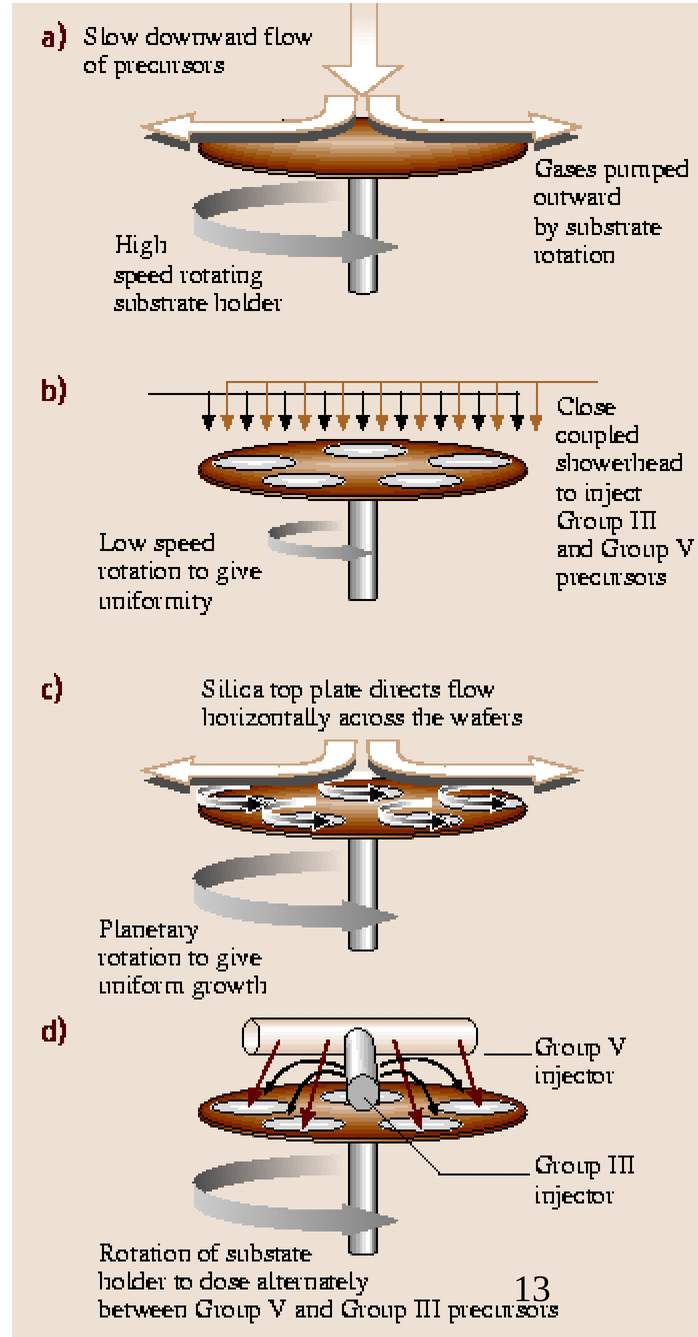
NH3
SiH4



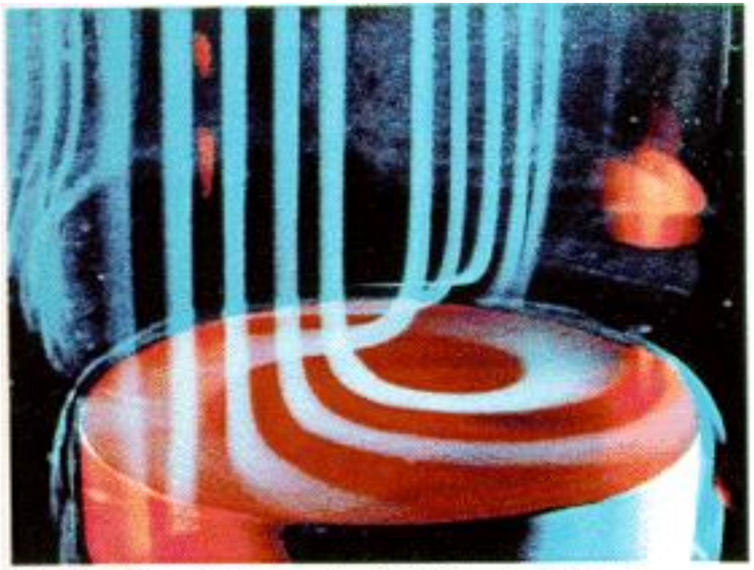
To reactor

Bubblers
TMGa
TEGa
TMAl
TMIn
Cp2Mg

MOVPE reactors



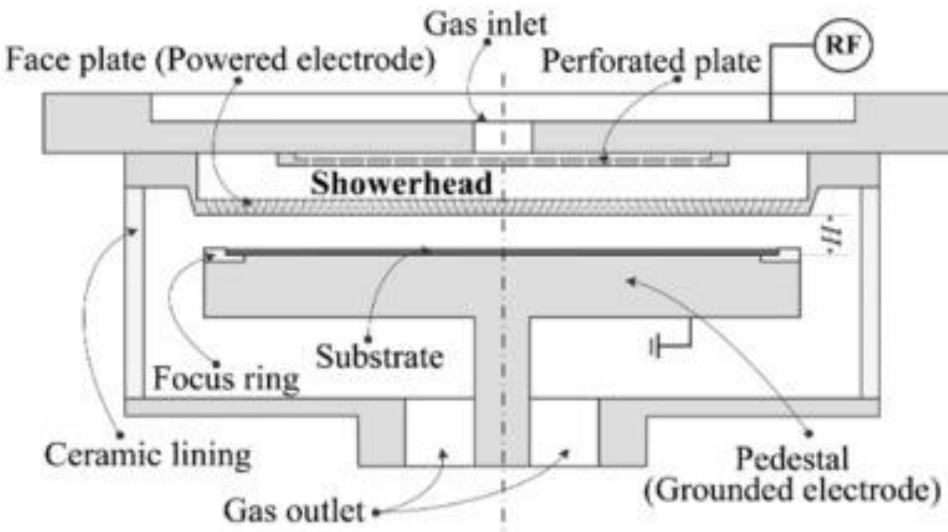
Turbodisc VEECO



Smoke Flow Patterns in Rotating Disk System
(EXPERIMENTAL).
(Data courtesy of Sandia National Laboratories)

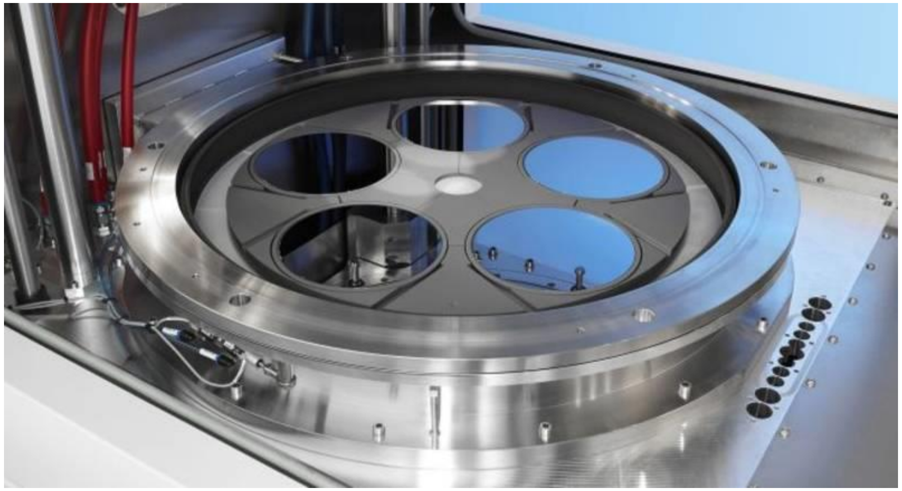
The largest system:31×4”, 12×6” and 6×8”

Showerhead (AIXTRON)

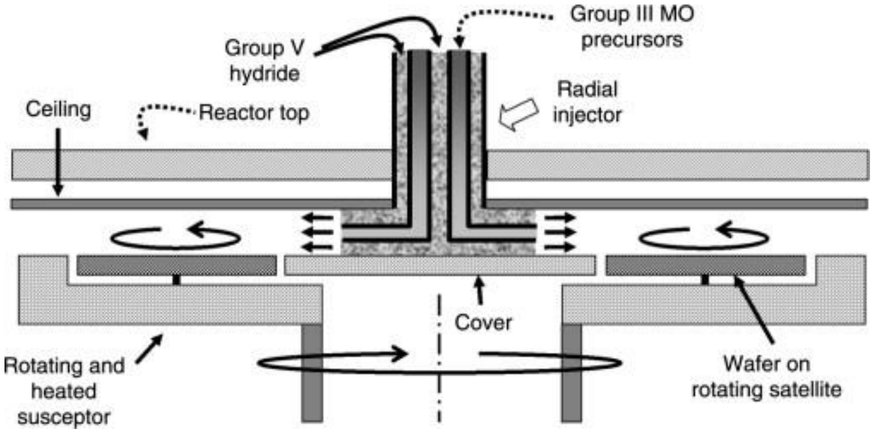


Planetary reactors

(AIXTRON)



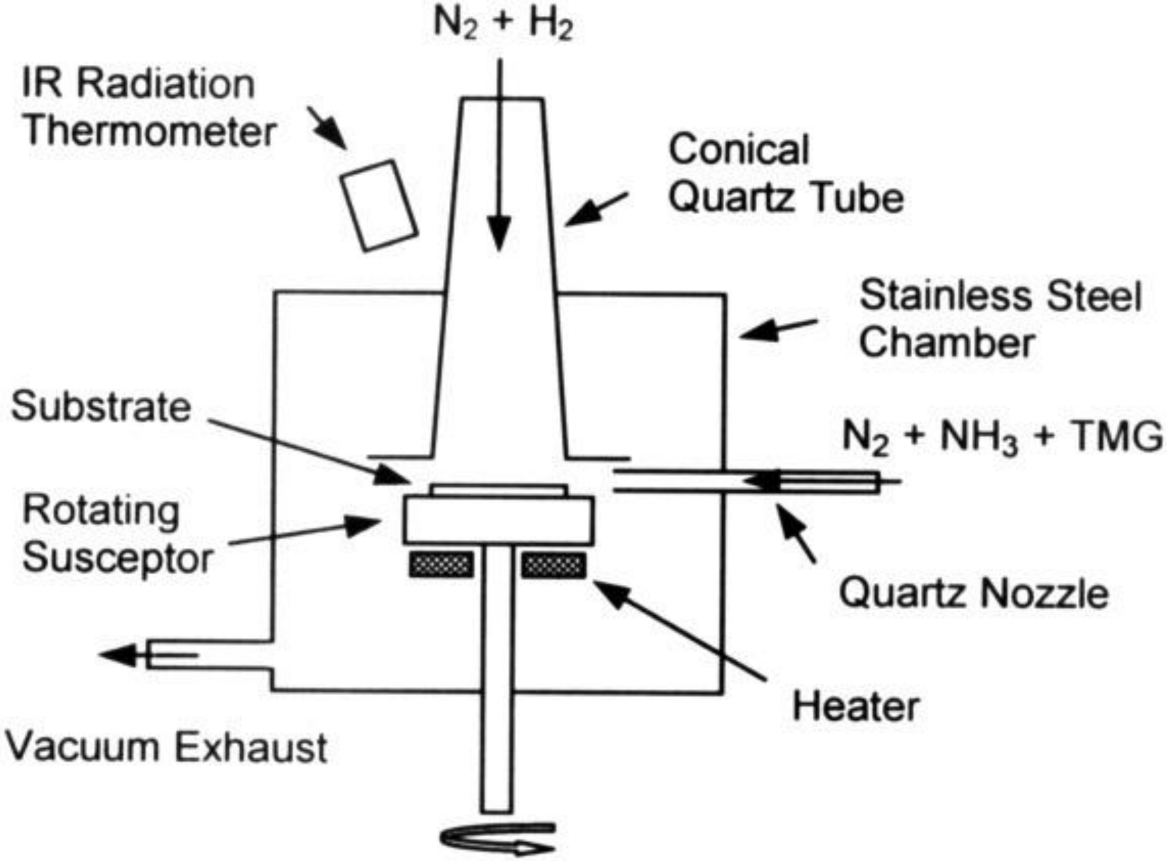
5x8-inch



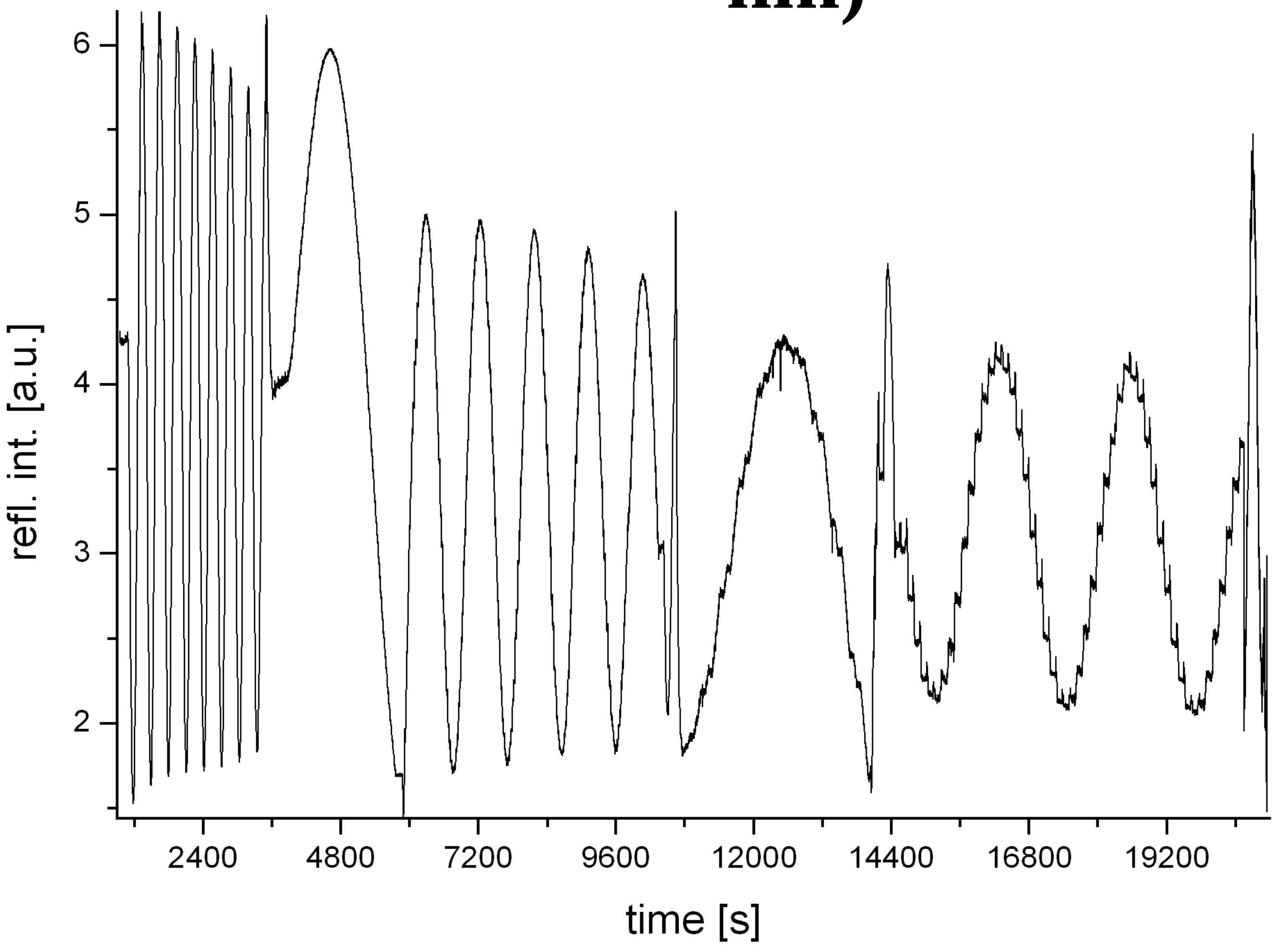
56x2-inch

Nakamura's reactor (Nichia)

A little bit similar reactors are offered by Nippon Sanso



Laser reflectometry used in every MOVPE (every oscillation corresponds to 100-400 nm)



1. Epitaxial reactors
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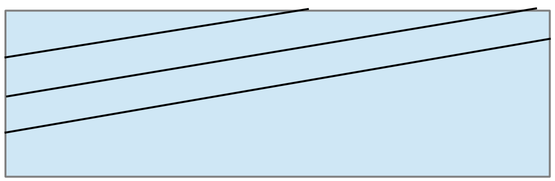
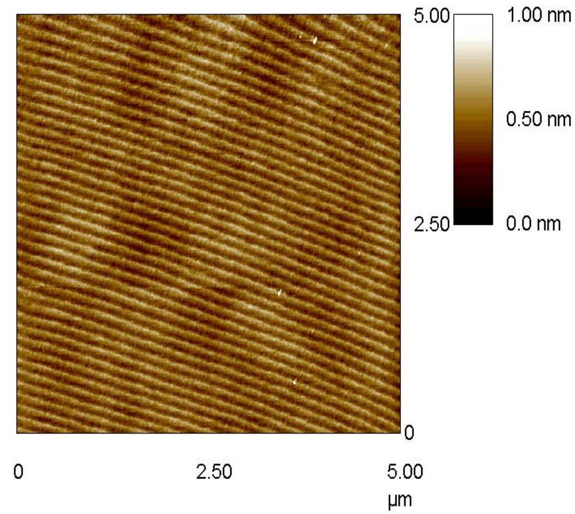
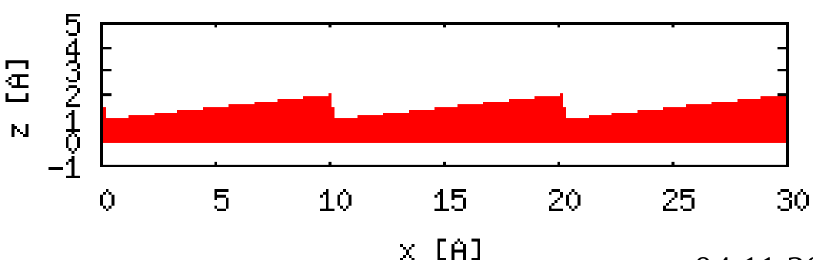
Step-flow growth mode and substrate off-orientation

- The steps flow slower for higher off-orientation

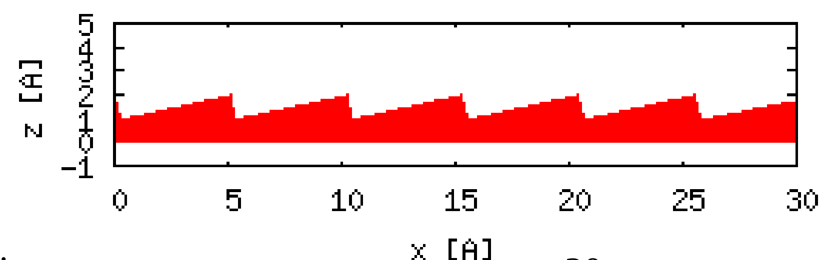




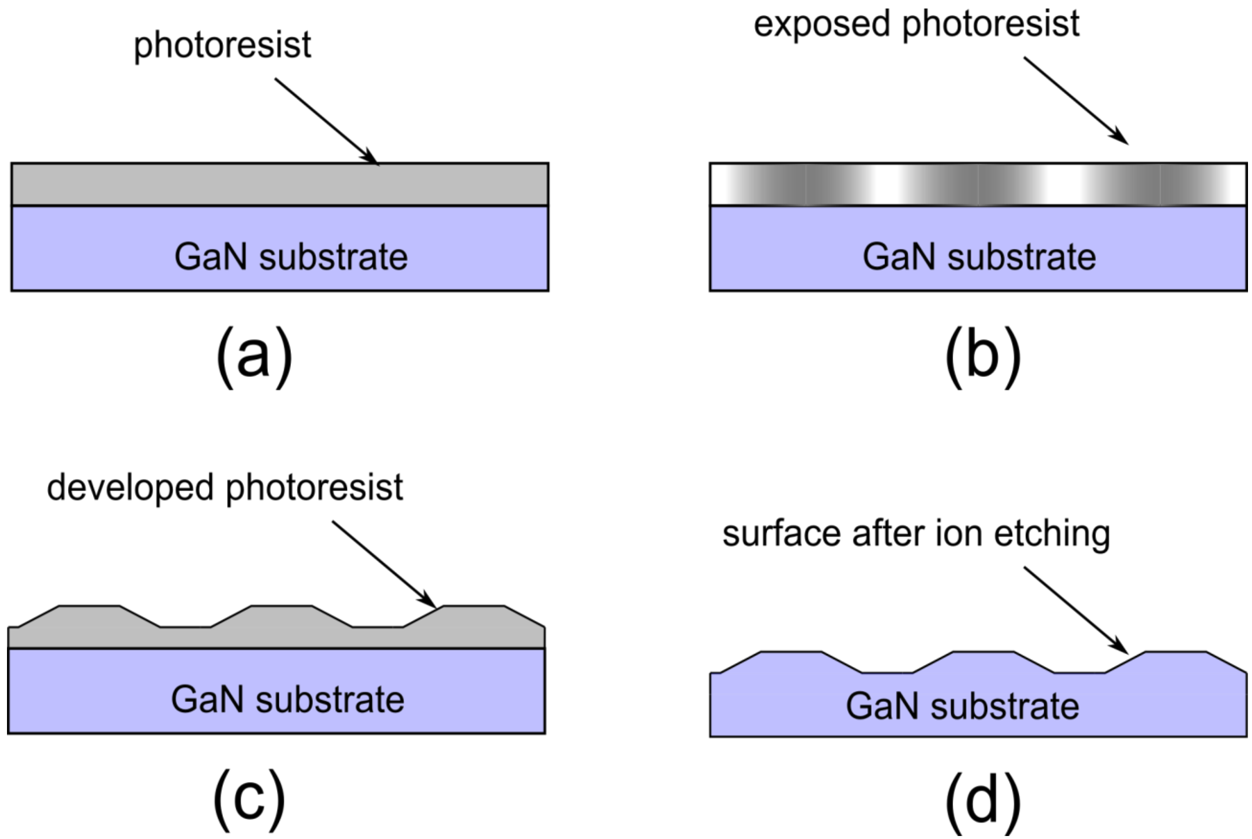
Low misorientation - fast atomic steps



Higher misorientation - slower atomic steps

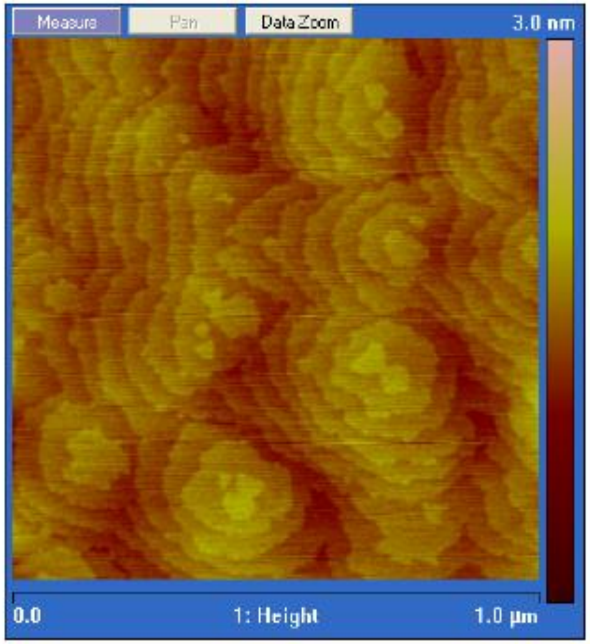


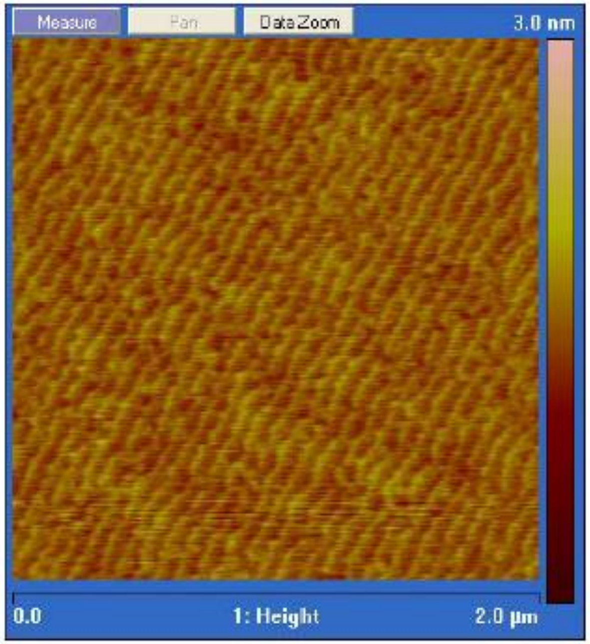
Lateral patterning

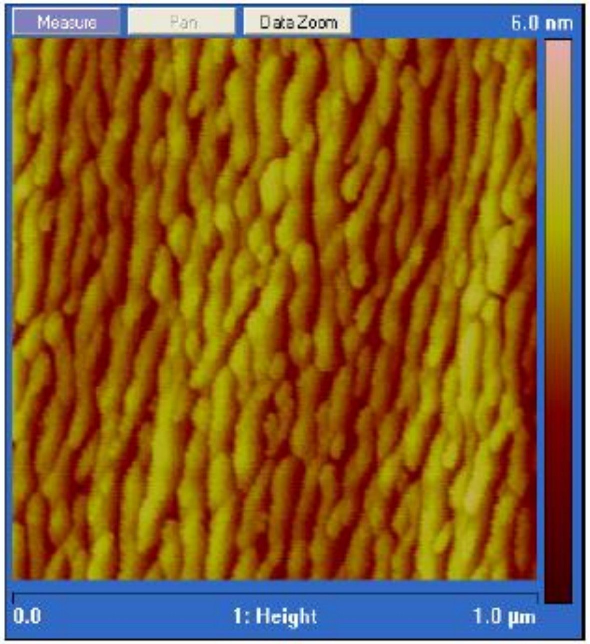


M. Sarzynski idea
Patented by TopGaN/Unipress

Morphology of InGaN layers grown on GaN substrates of different misorientation, Tgrowth = 820oC







0.2 degree

0.8 degree

1.8 degree

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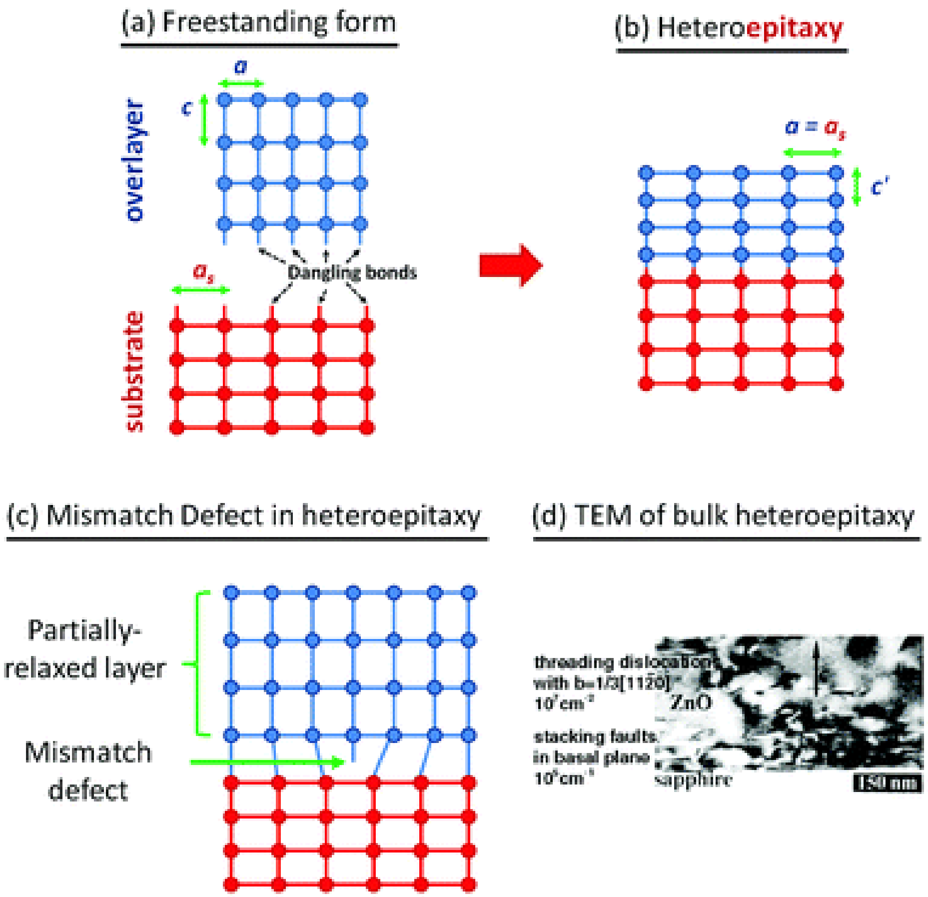
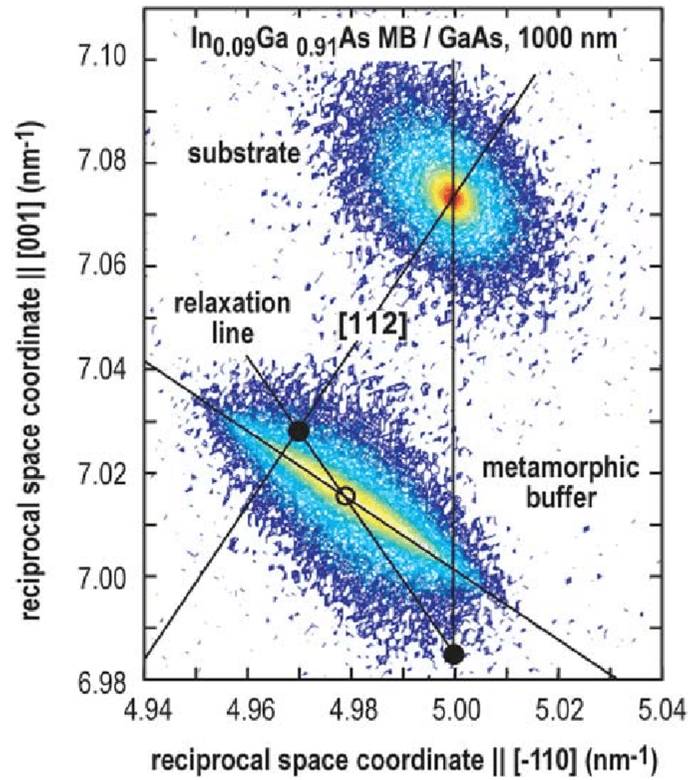
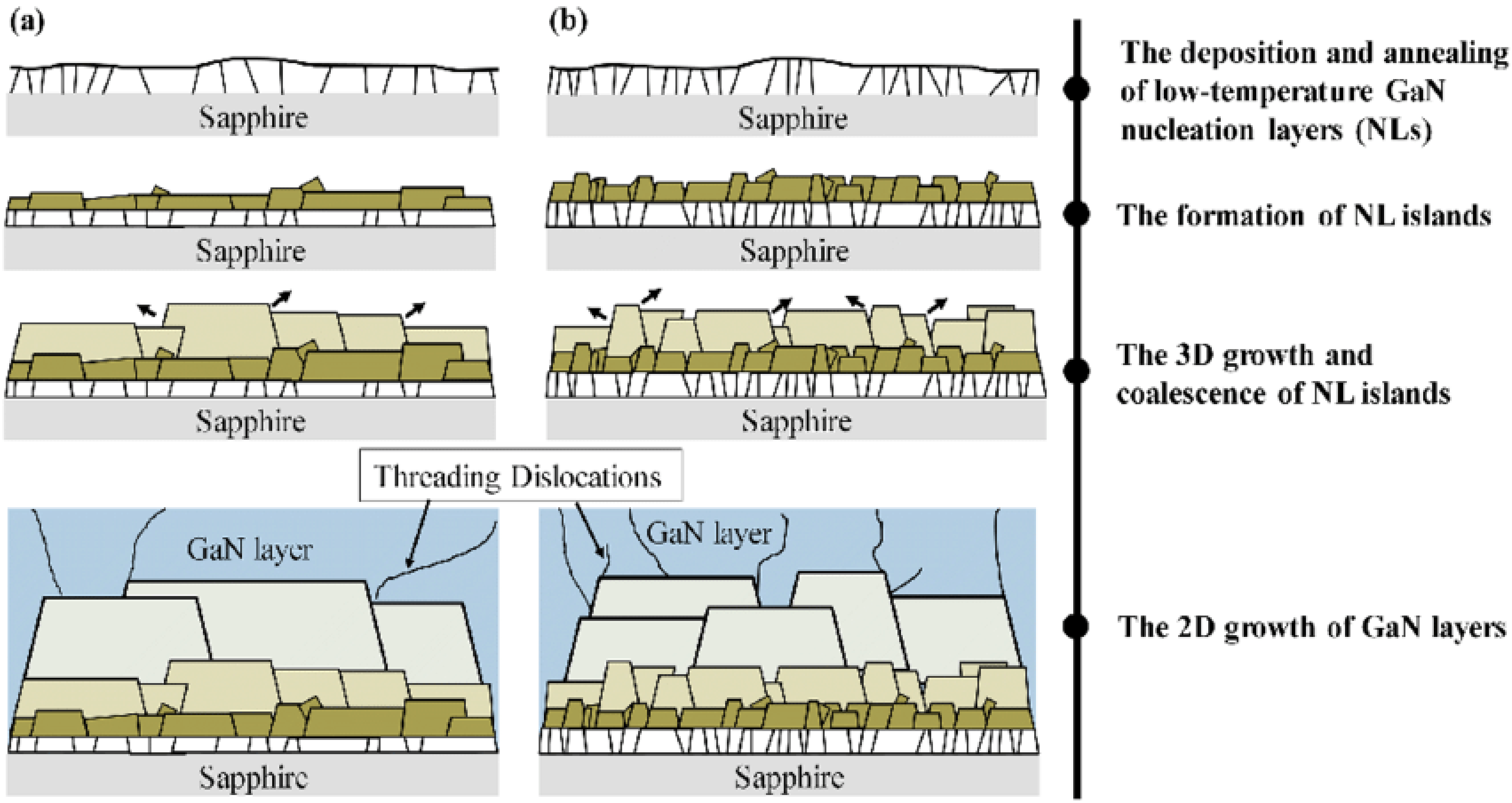
Lattice relaxation



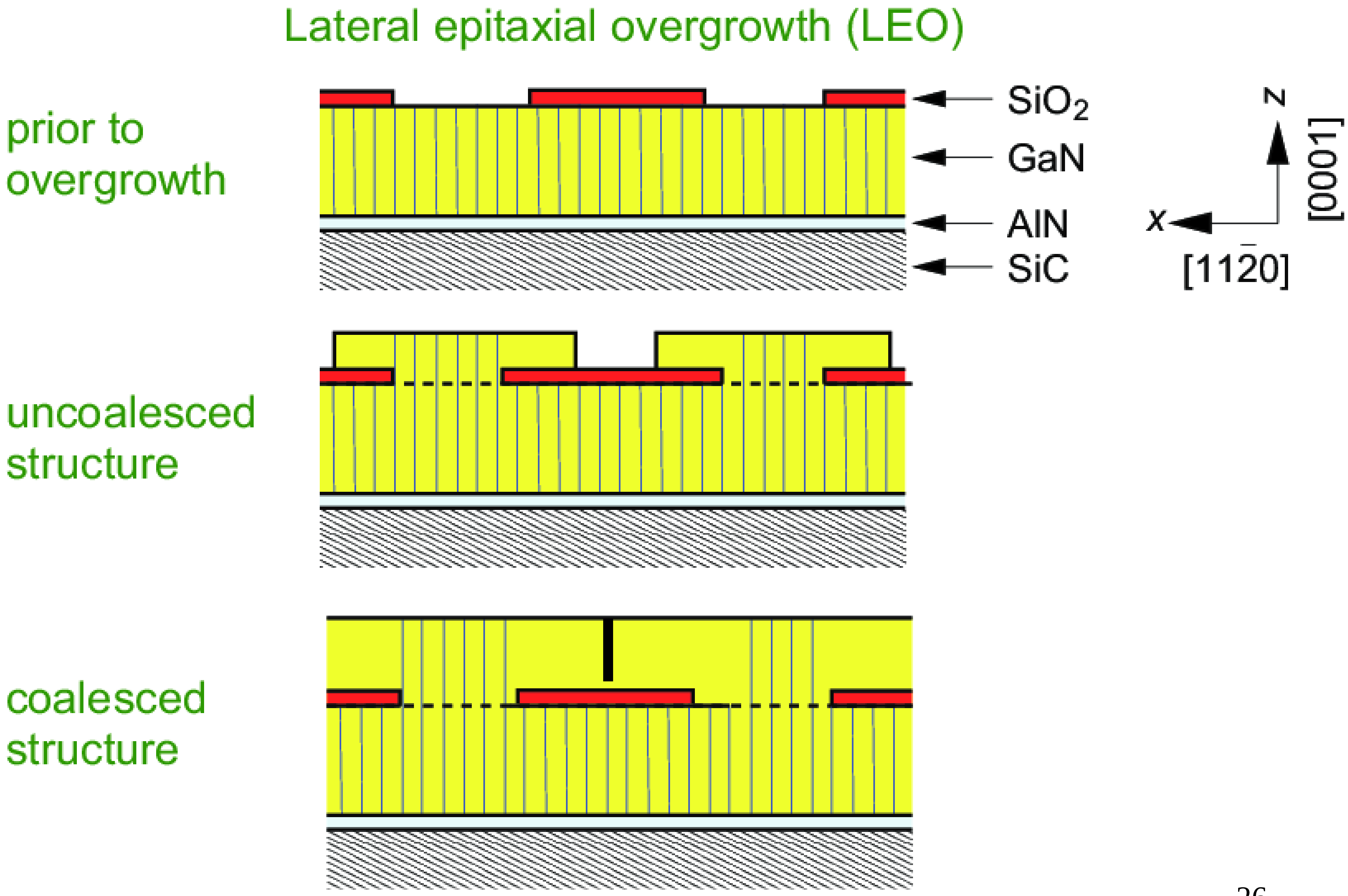


Growth on highly mismatched substrates



Schematic of the growth process of GaN layers for (a) samples S1 and (b) S2 with H 2 and N 2 as nucleation layer carrier gases.

Epitaxial Lateral Overgrowth



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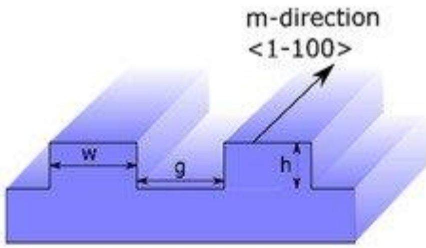
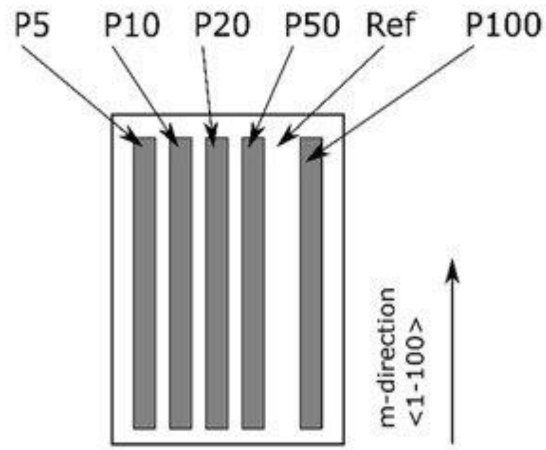
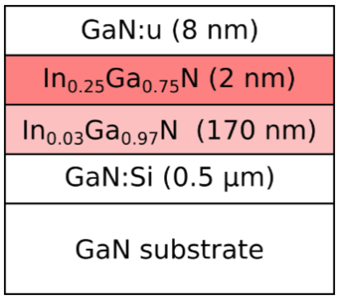
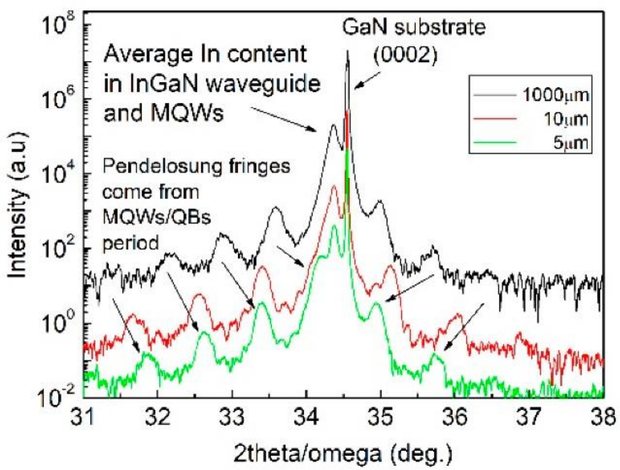

m-direction
<1-100>
w
g
h

(a)

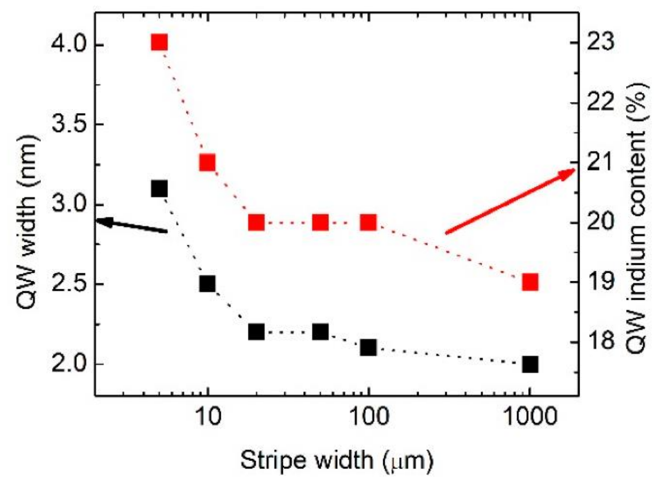

P5
P10
P20
P50
Ref
P100
m-direction
<1-100>

(b)

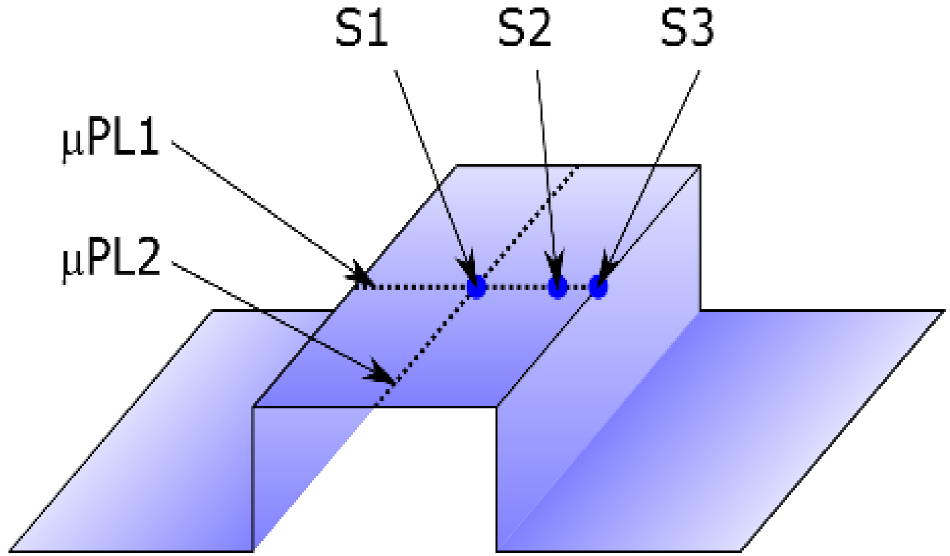

GaN:u (8 nm)
In0.25Ga0.75N (2 nm)
In0.03Ga0.97N (170 nm)
GaN:Si (0.5 µm)
GaN substrate

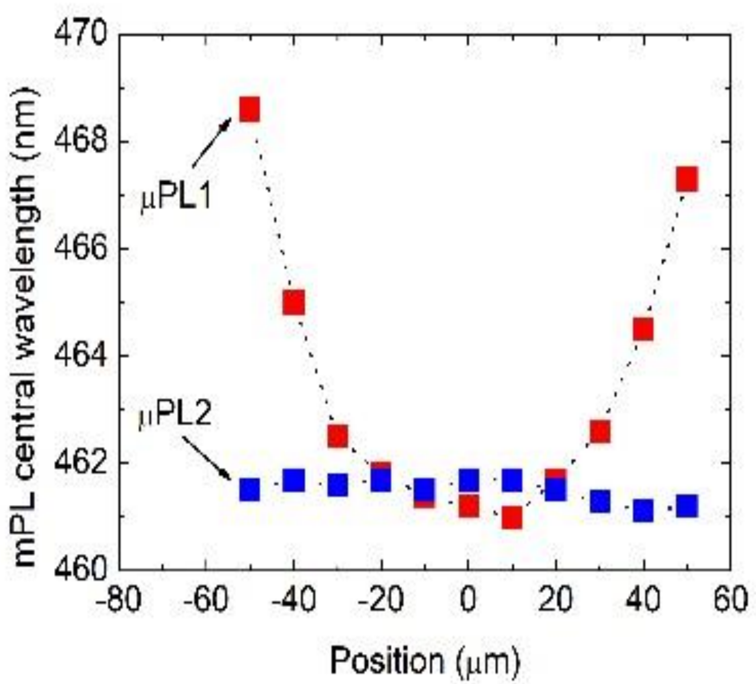

GaN substrate
(0002)
Average In content
in InGaN waveguide
and MQWs
Pendelosung fringes
come from
MQWs/QBs
period
1000µm
10µm
5µm
Intensity (a.u)
2theta/omega (deg.)

(a)

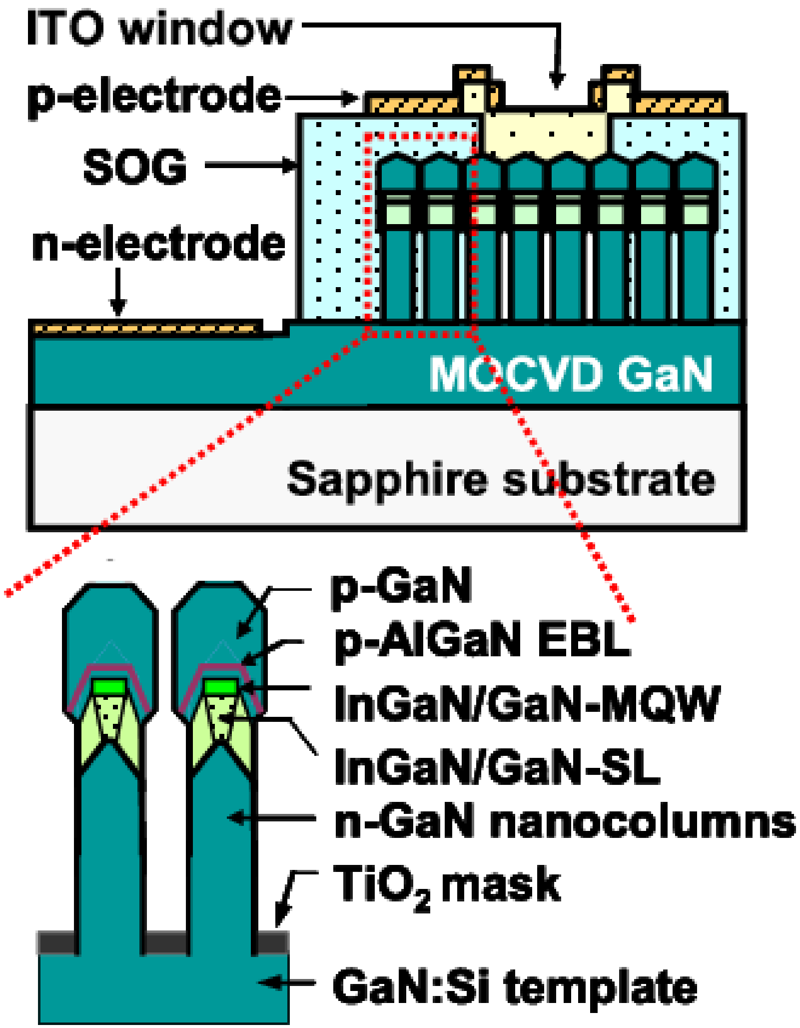

QW width (nm)
QW indium content (%)
Stripe width (µm)

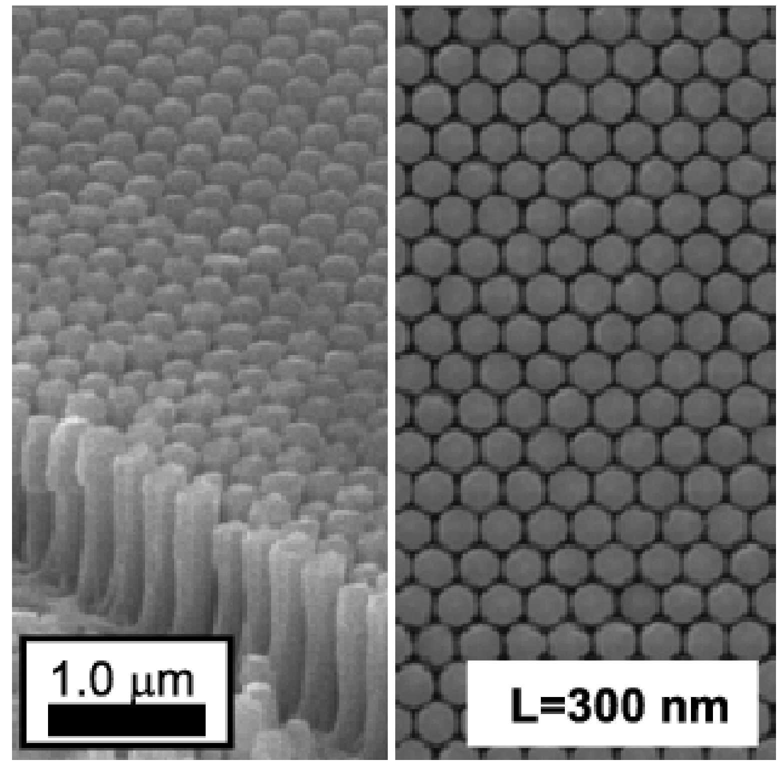
(b)


S1
S2
S3
μPL1
μPL2


470
468
466
464
462
460
-80
-60
-40
-20
0
20
40
60
mPL central wavelength (nm)
Position (μm)
μPL1
μPL2

Nanocolumns

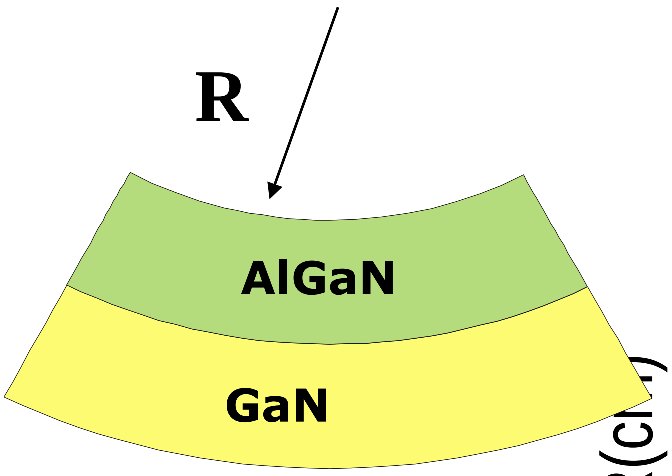


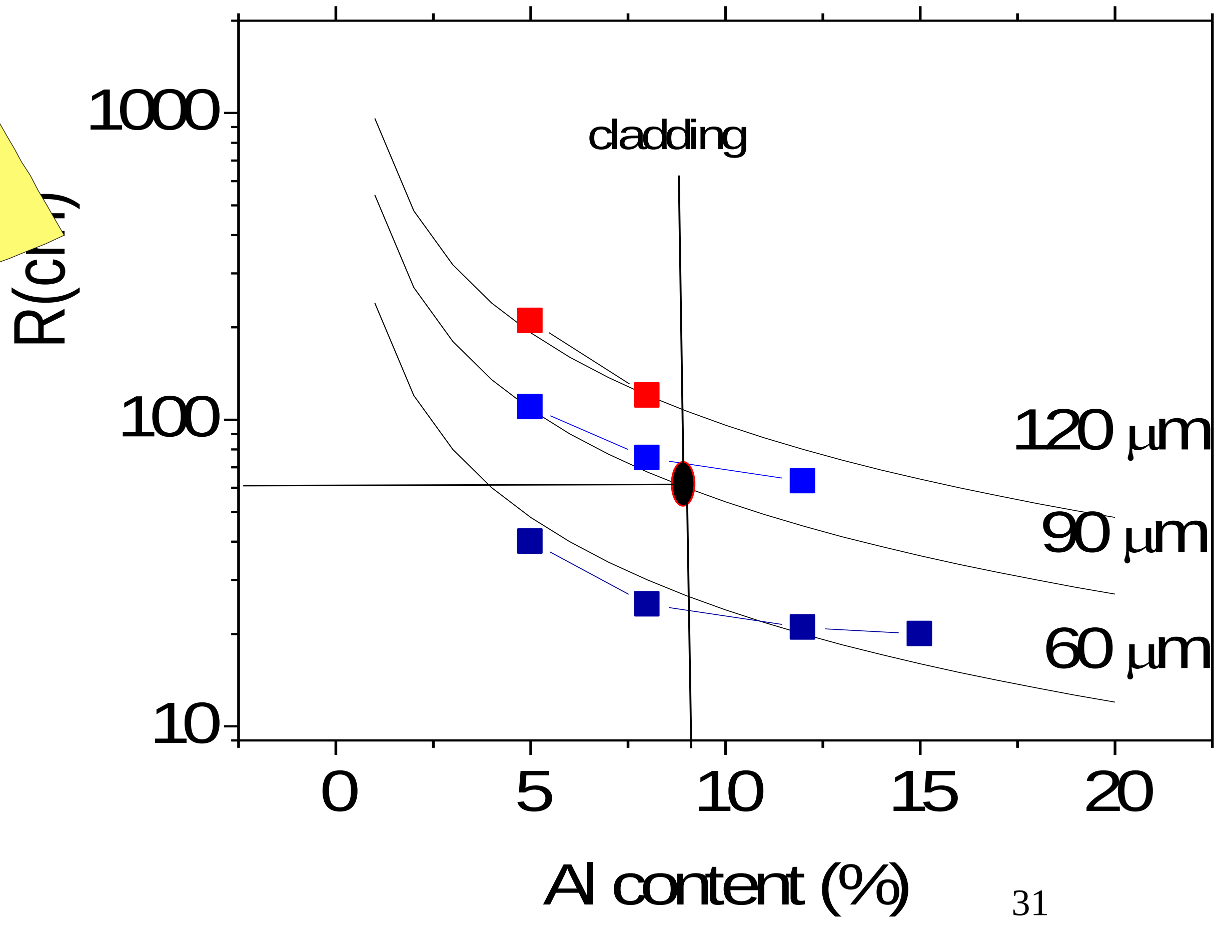


(a) (b)

In LDs, we need to have thick AlGaN cladding layers

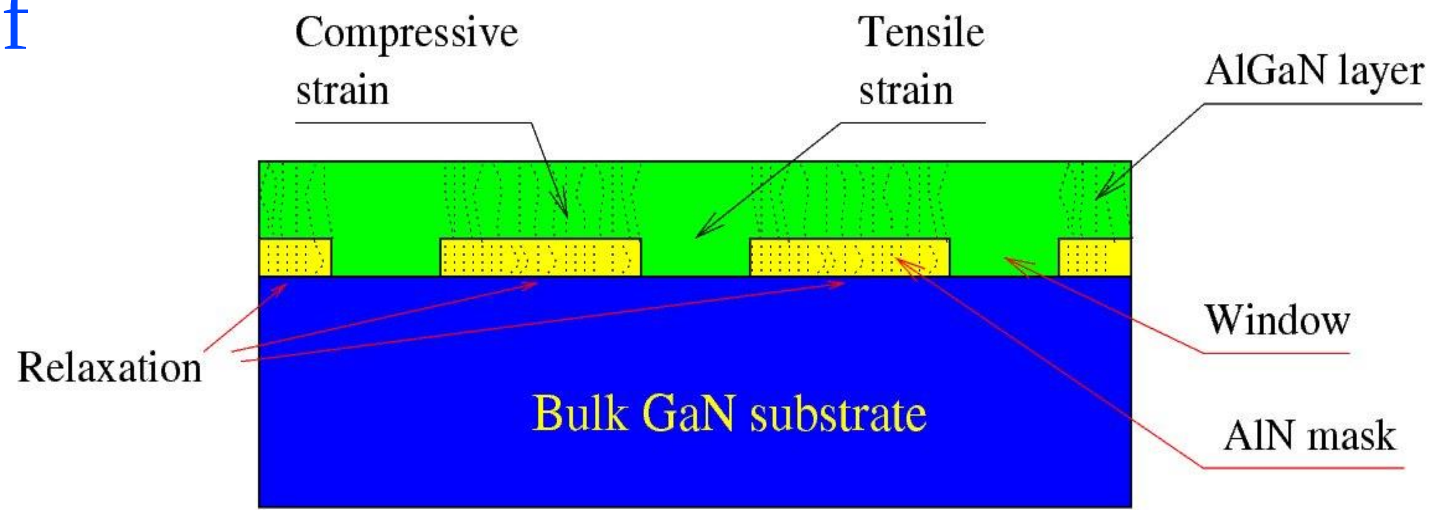
Tensile strain: cracking and bowing

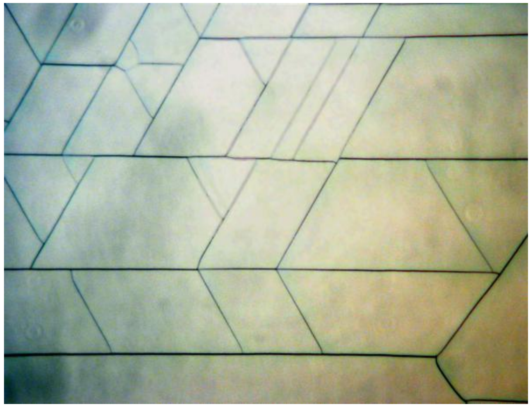




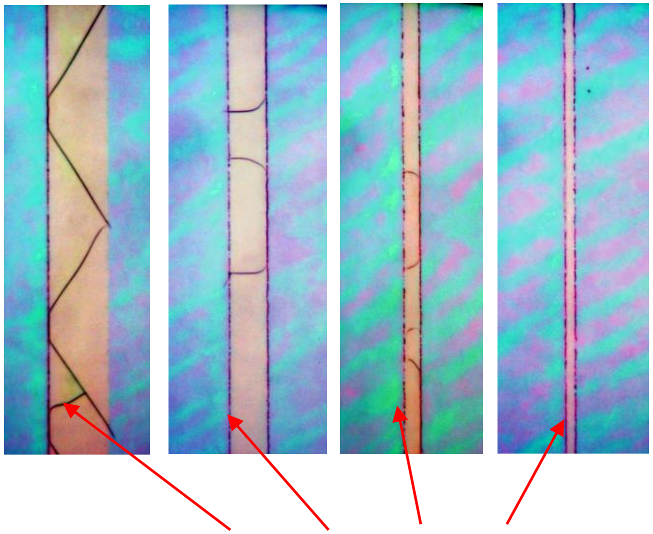
Elimination of cracking and bowing

AlGaN 27%
220 nm

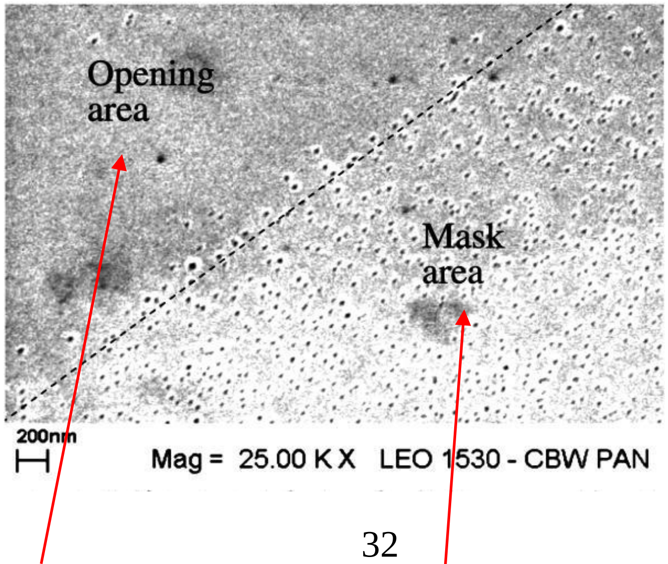




No maskAlN



With mask 15 10 5 3 μm



Density of defects: **window<10^6/cm2** **on mask 10^{10}/cm2**

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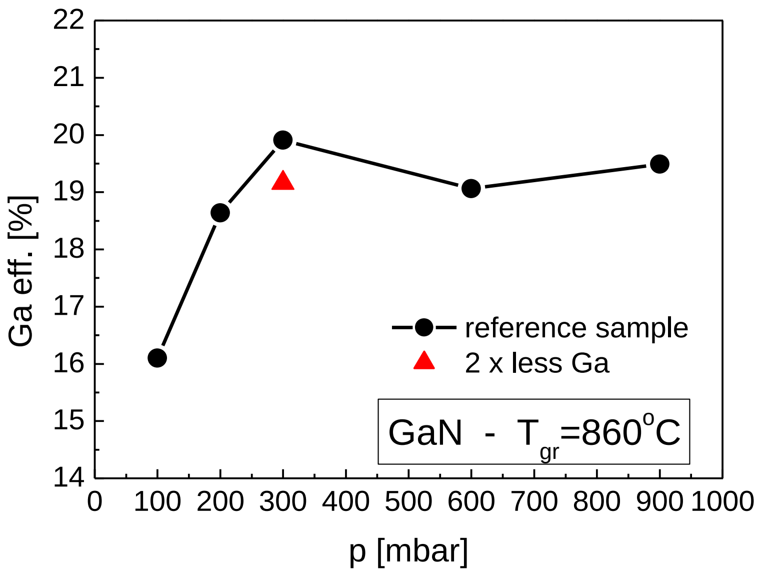
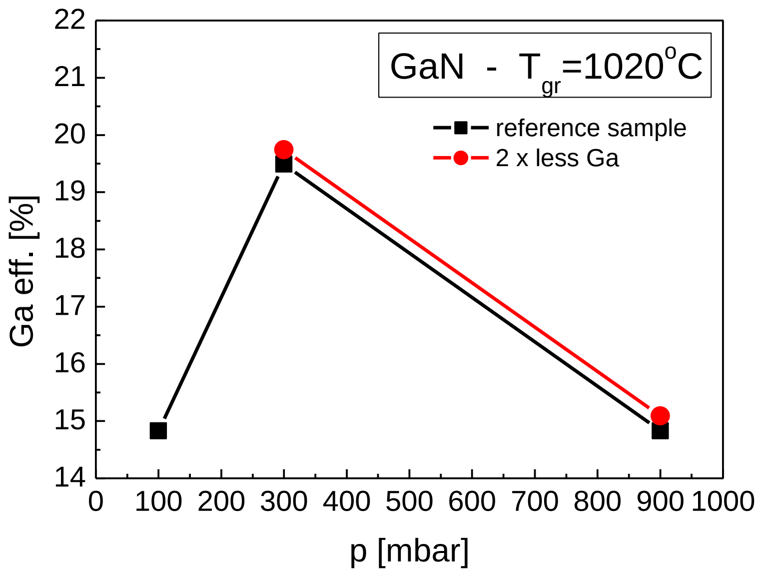
Efficiency of Ga incorporation into GaN in HM

Efficiency= number of atoms in gas phase/ number of atoms incorporated into the epi layer



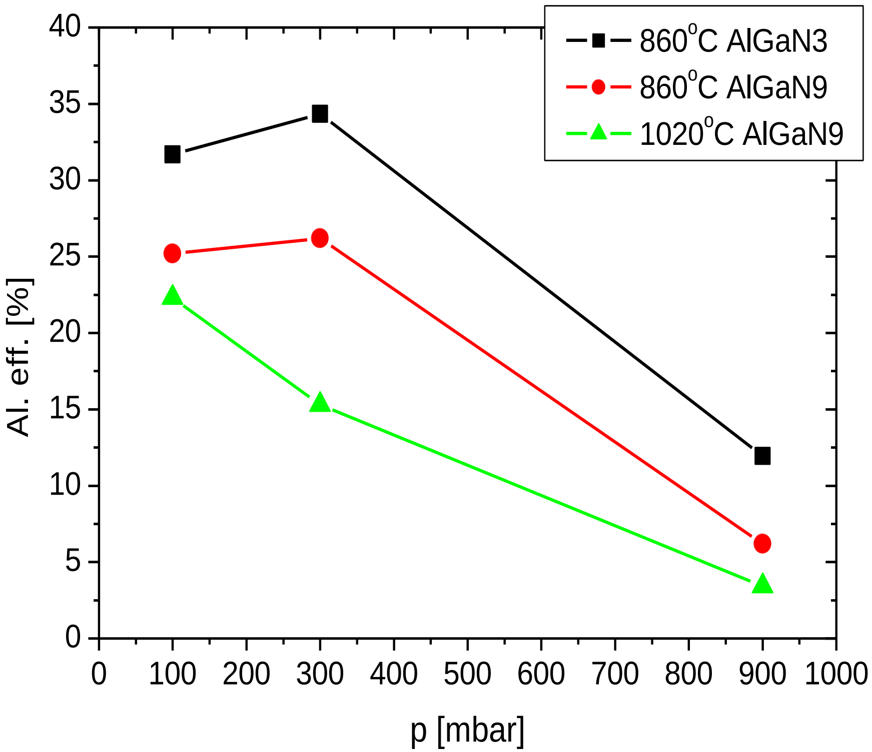


At low pressure, the Ga atoms are blown away from the surface.

At high pressure and temperature, the rate of prereactions increase which prevents Ga incorporation

Efficiency of Ga incorporation independent on amount of TEGa

Efficiency of Al incorporation into AlGaN



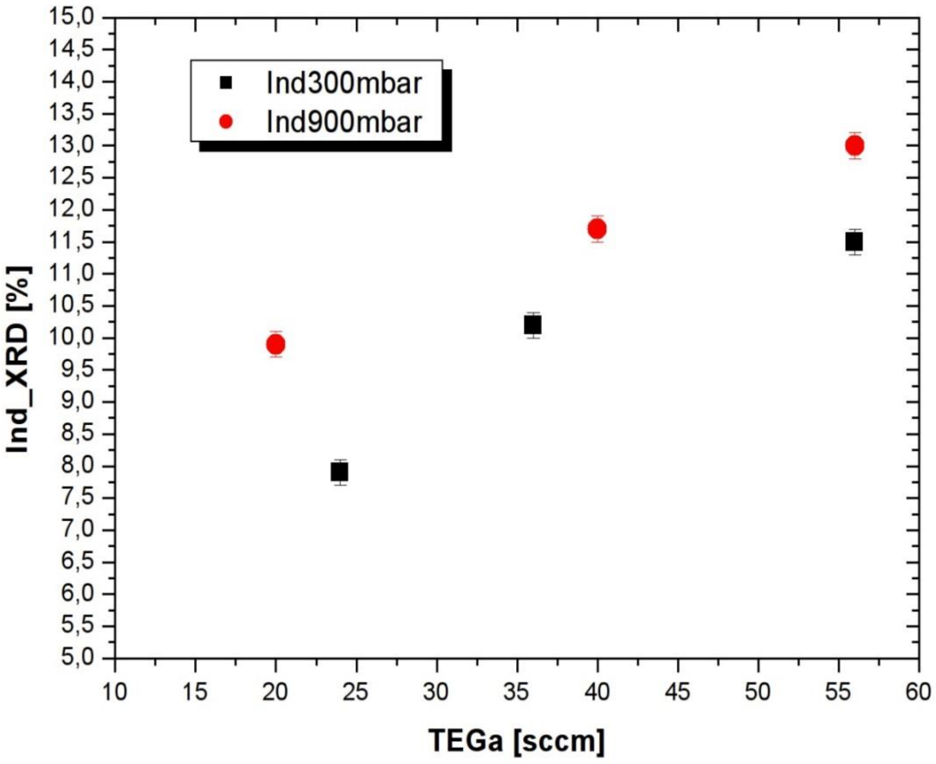
Efficiency of Al incorporation falls down with TMAl flow increase

Prereactions rate higher than in the case of GaN

More prereactions at high temperature and pressure

3 sccm of TMAl
9 sccm of TMAl

Efficiency of In incorporation into InGaN versus TEGa flow

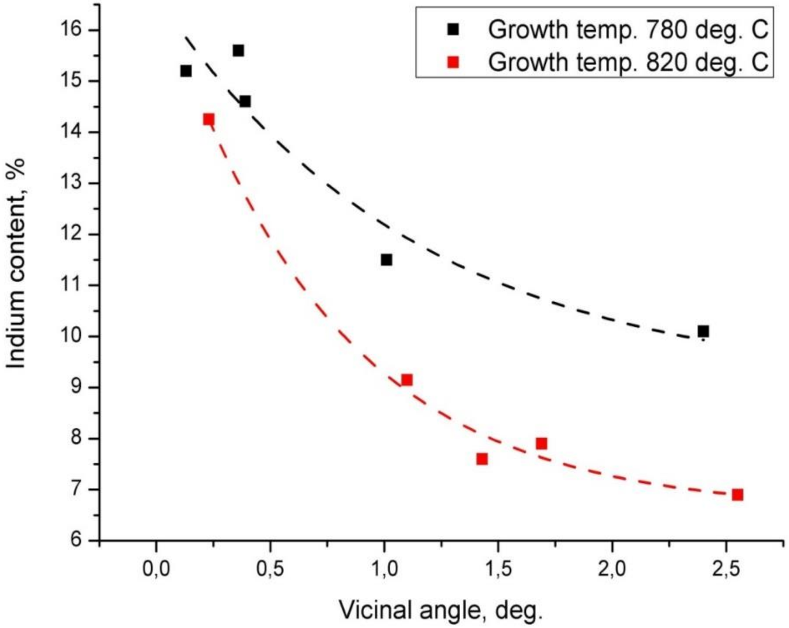


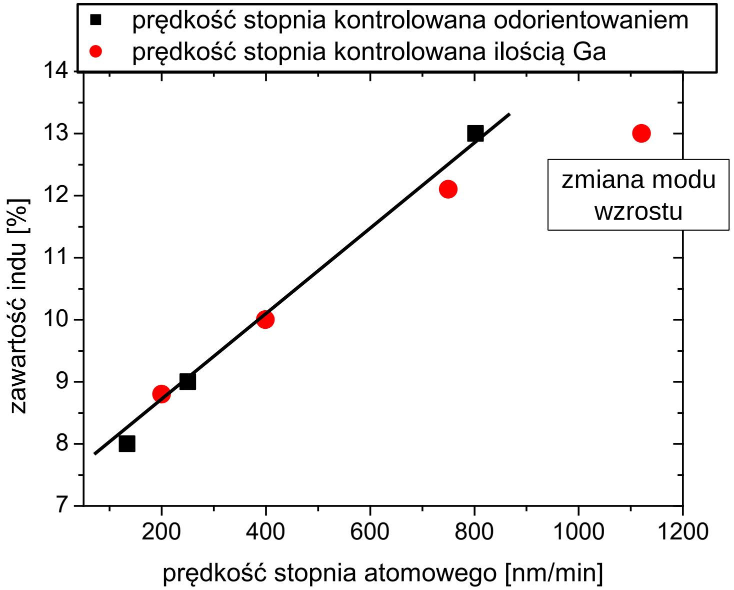
More Ga source in the gas phase, but more In in InGaN solid phase.

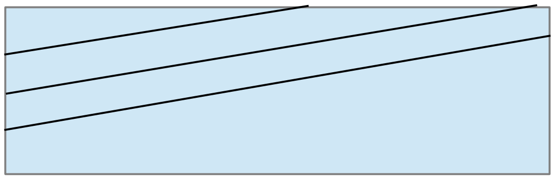
More indium at elevated pressure (not for every reactor!)

In atoms to be incorporated must be surrounded by Ga atoms

In incorporation into InGaN layers versus GaN substrate off-orientation







If we have bad morphology (steps are not identical), we deal with In inhomogeneous incorporation

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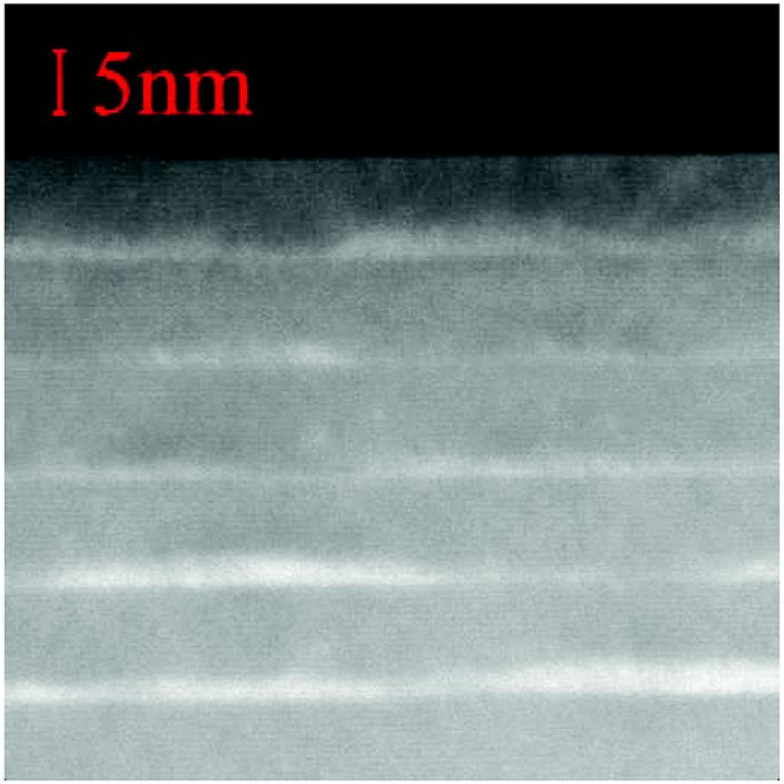
As grown InGaN QWs



Indium fluctuations

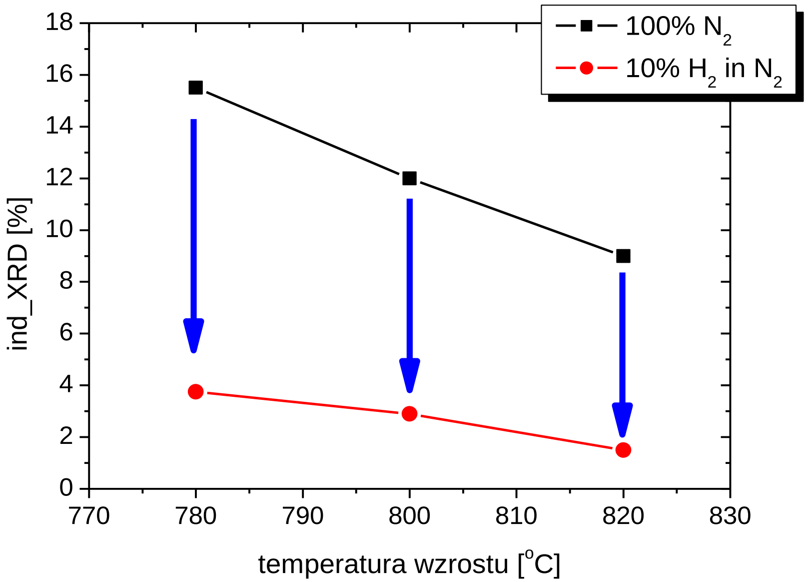
QW thickness fluctuations

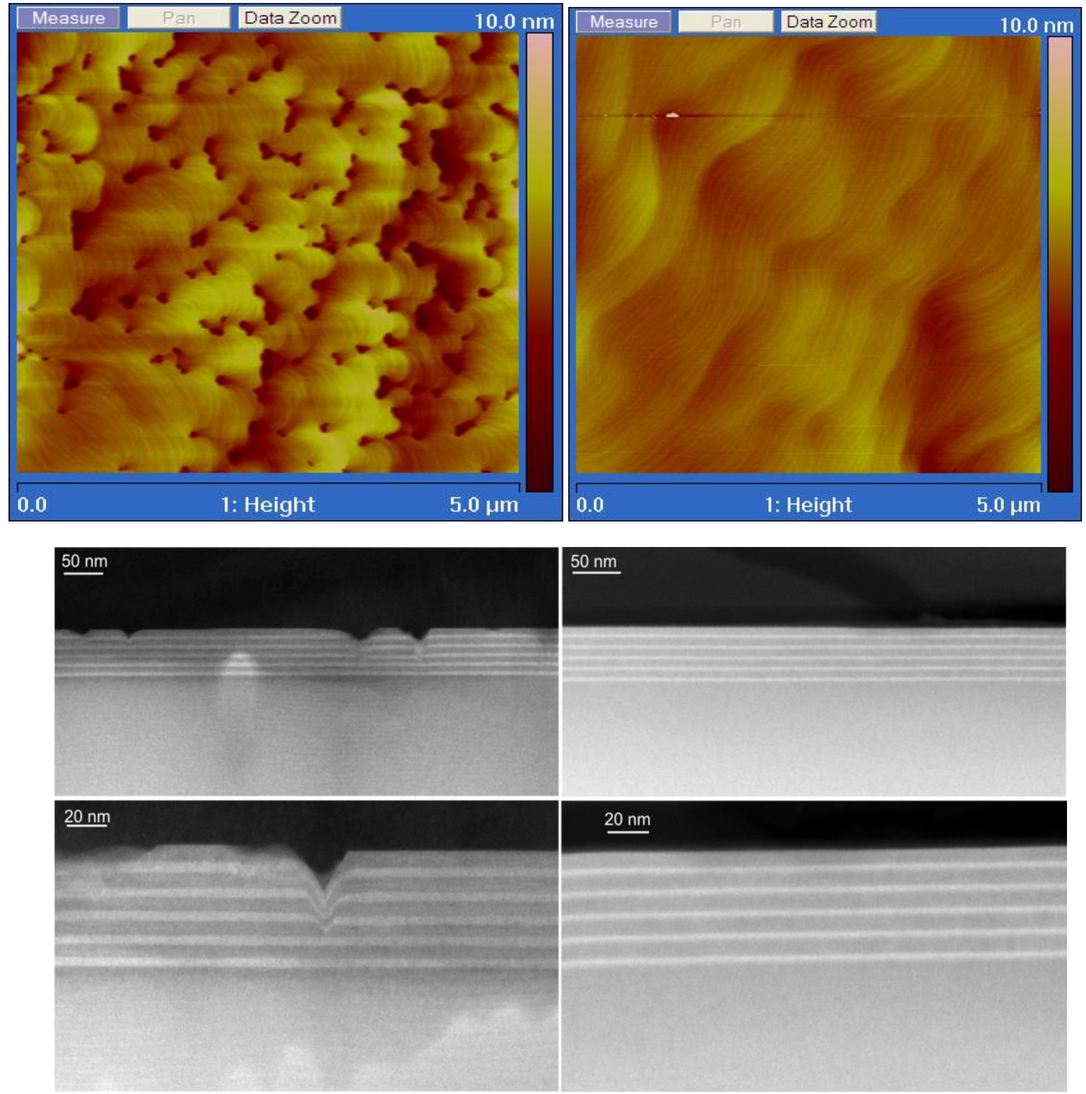
Acceptable for LEDs
Not for LDs

Examples of MOVPE growth: InGaN Quantum Wells

1. Influence of temparature
2. Influence of hydrogen as a carrier gas
3. Homogenization and decomposition

Influence of hydrogen used in the carrier gas



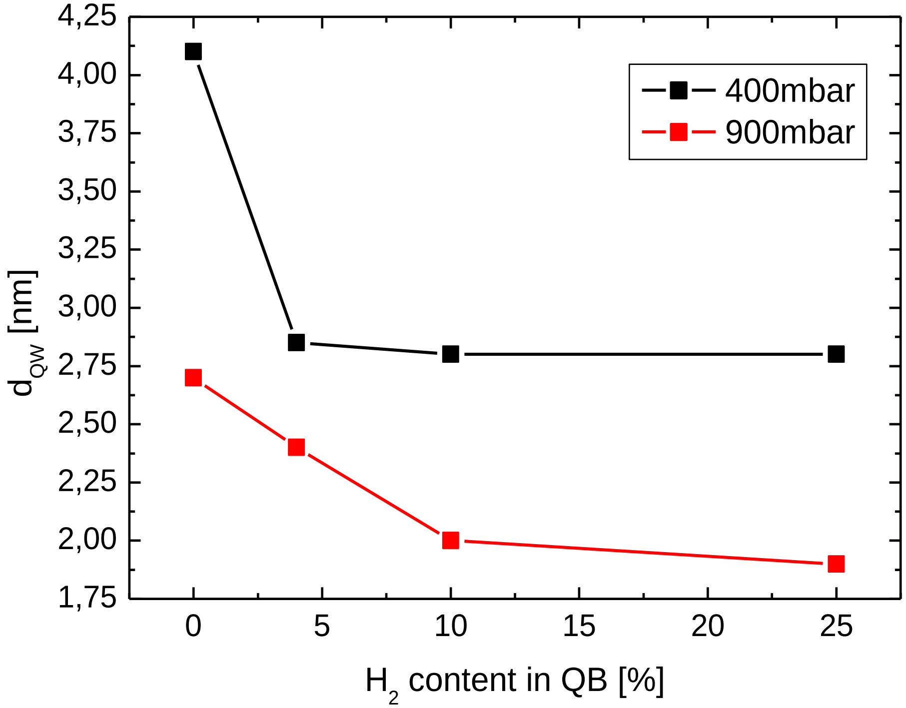


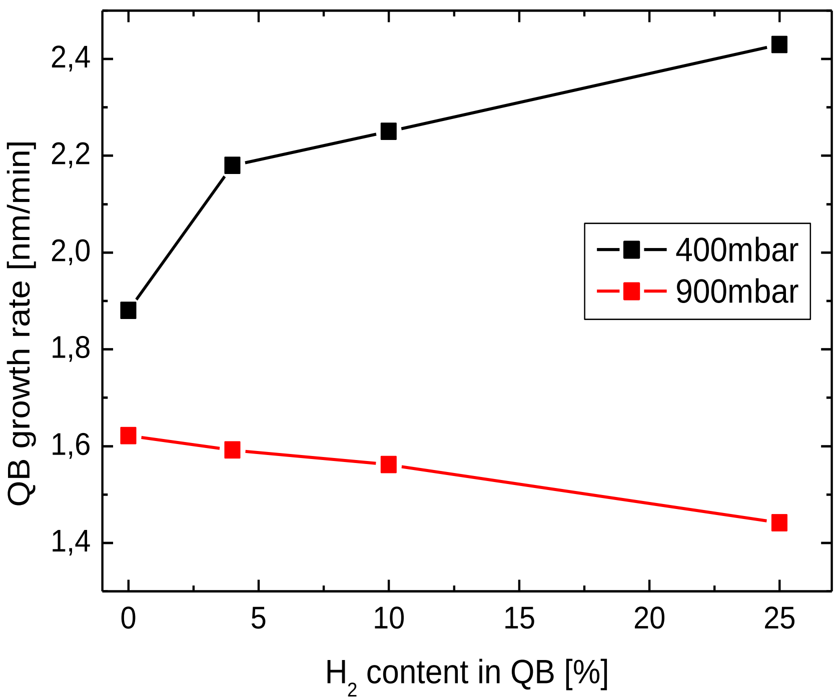
No H2 in N2

2% of H2 in N2

We use hydrogen for growing the QBs- but with special care

Influence of hydrogen used in the carrier gas during GaN barrier growth in CCS

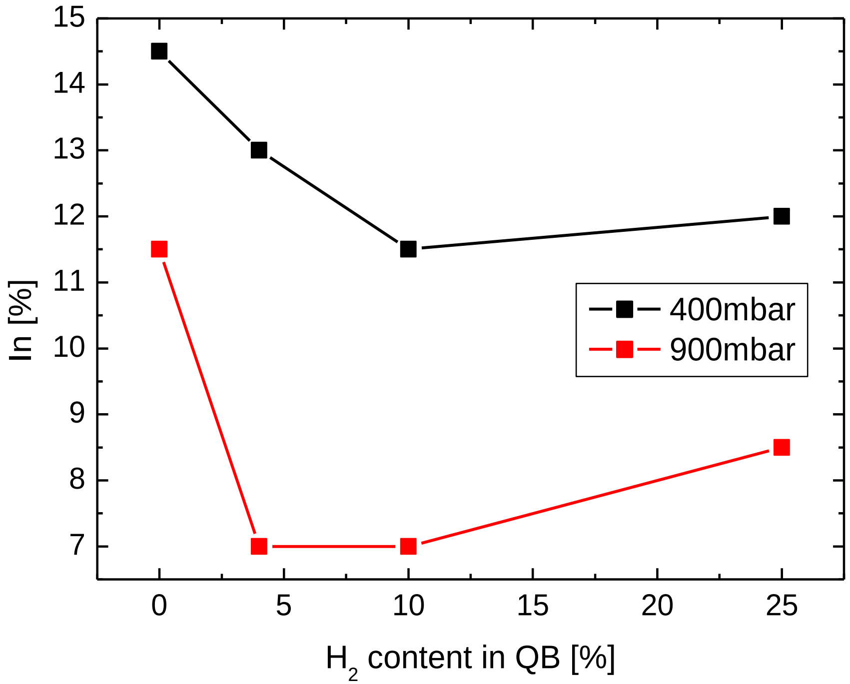


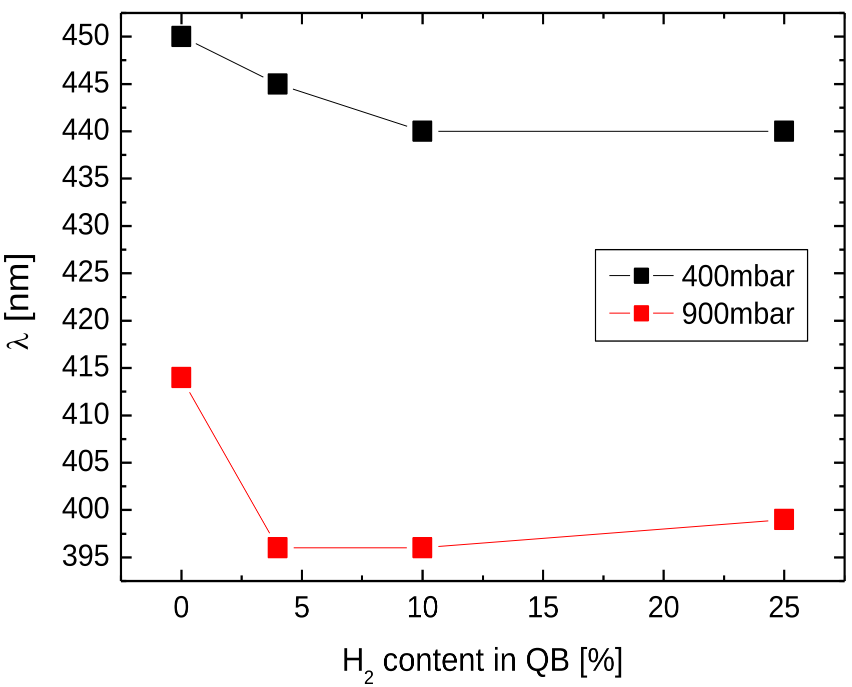


Hydrogen used in QB growth etches off InGaN QW more efficiently at high pressure

Hydrogen increases the growth rate of GaN (QB) at low pressure, deacreses at higher pressure.

Influence of hydrogen used in the carrier gas during GaN barrier growth in CCS





Hydrogen used in QB growth etches In off InGaN QW more efficiently at high pressure

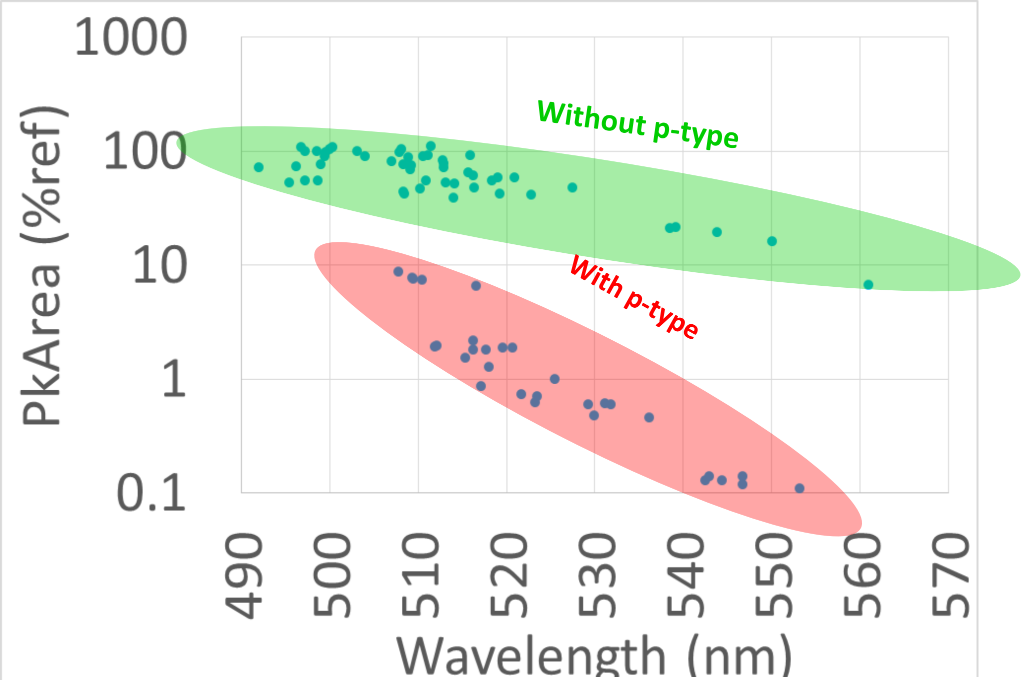
Growth temperatures of the laser diode structure

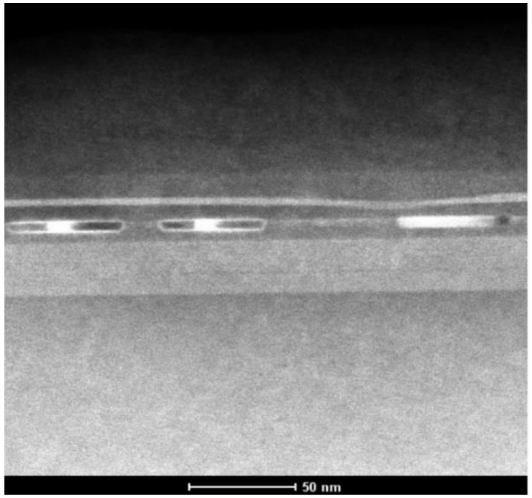
GaN:Mg	
AlGaN:Mg	
GaN:Mg	**T_{Gr} = XXXX^{o}C**
EBL - AlGaN:Mg	
Cap – GaN	
Cap – GaN	
QW - InGaN	**T_{Gr} QW $< 740^{o}$C**
QB – GaN	**T_{Gr} QB $> 740^{o}$C**
InGaN	For In content > 15%
GaN	
AlGaN:Si	**T_{Gr} = 1000^{o}C**
GaN:Si	
Substrate	

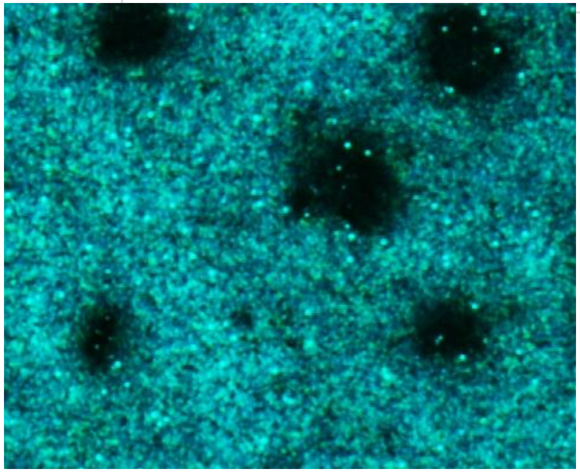
To obtain good p-type XXXX should be as high as possible (above 900^{o}C).

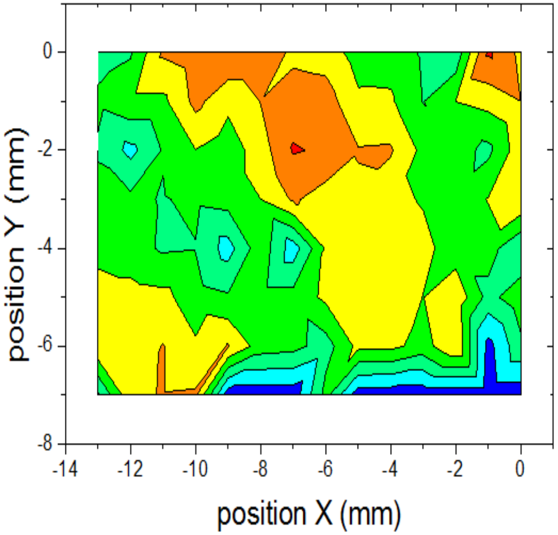
At high temperatures, InGaN QWs change

InGaN MQW decomposition





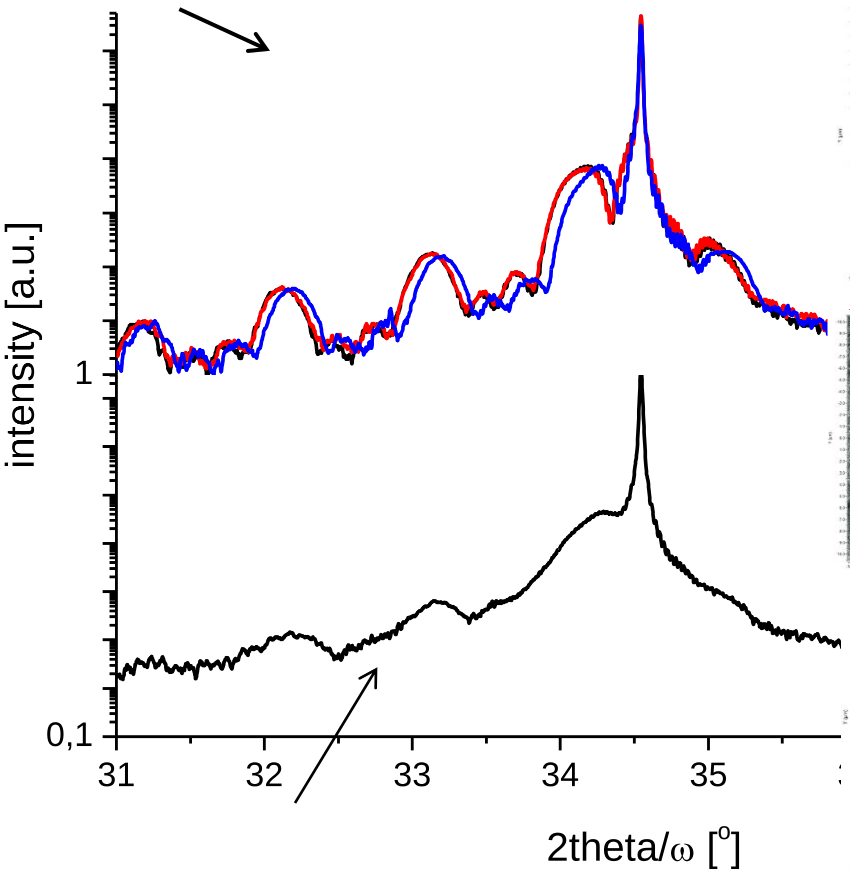




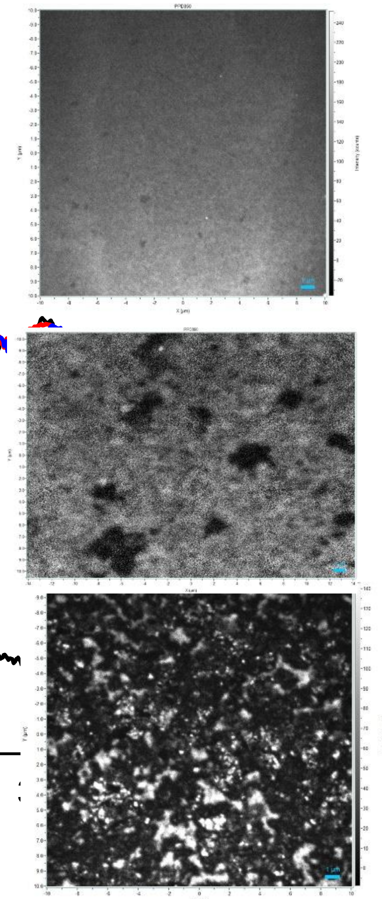
TEM: nanometer scale, Fluorescent microscopy: micrometers, MicroPL: milimieters

Changes of the MQWs during p-type growth at high temperature: samples on sapphire

HR-XRD measurements of LED structures:



CL



XRD:

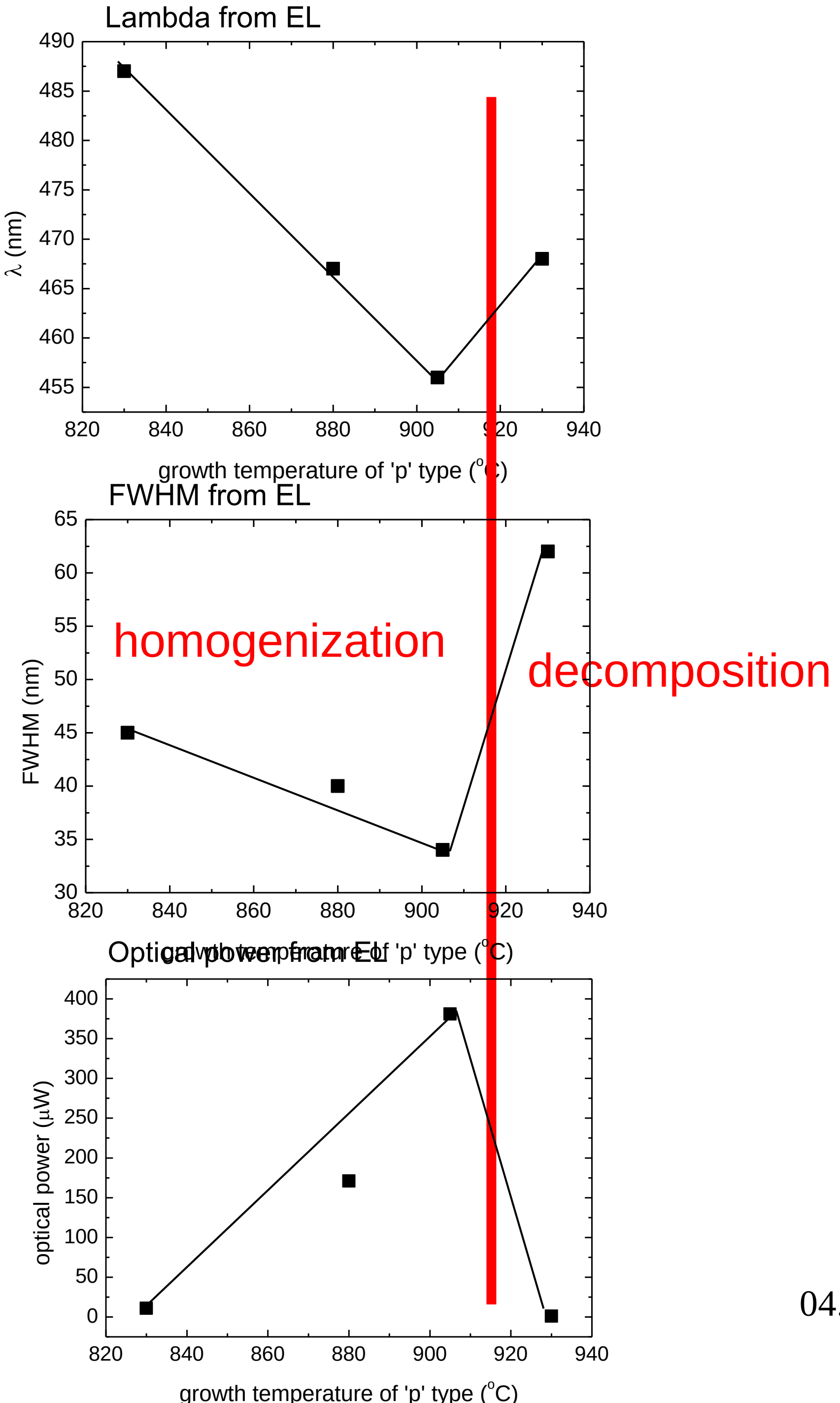
T=830°C In – 18%

T=880°C In – 18%

T=905°C In – 16%

T=930°C In – ?

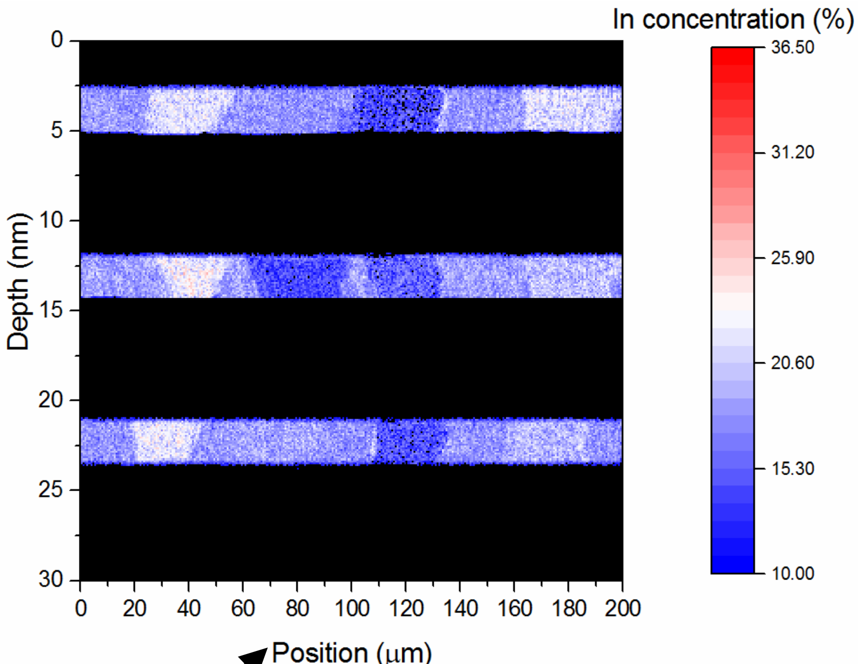
At high temperature satellite peaks becomes much more broader and disappear : QWs degradation occurs



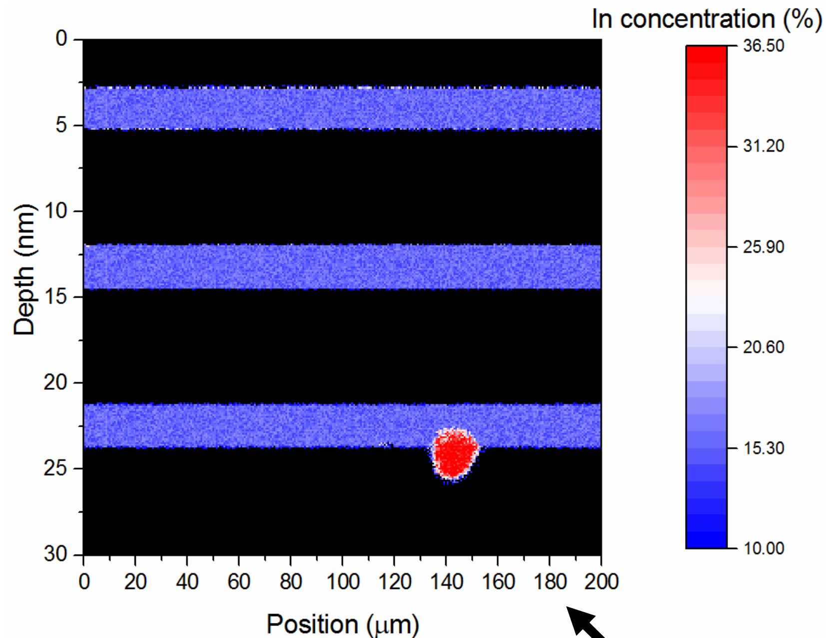
For samples on sapphire (high dislocation density) we are able to homogenize InGaN QWs by growing p-type at high temperature.

If this temperature is too high, we deal with a catastrophic damage

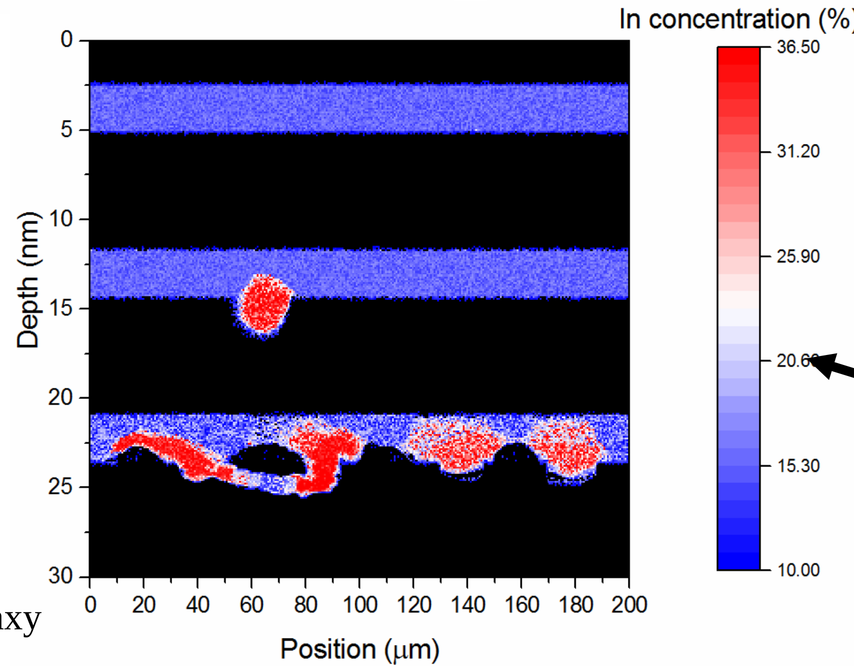
SIMS data of In-content



as grown

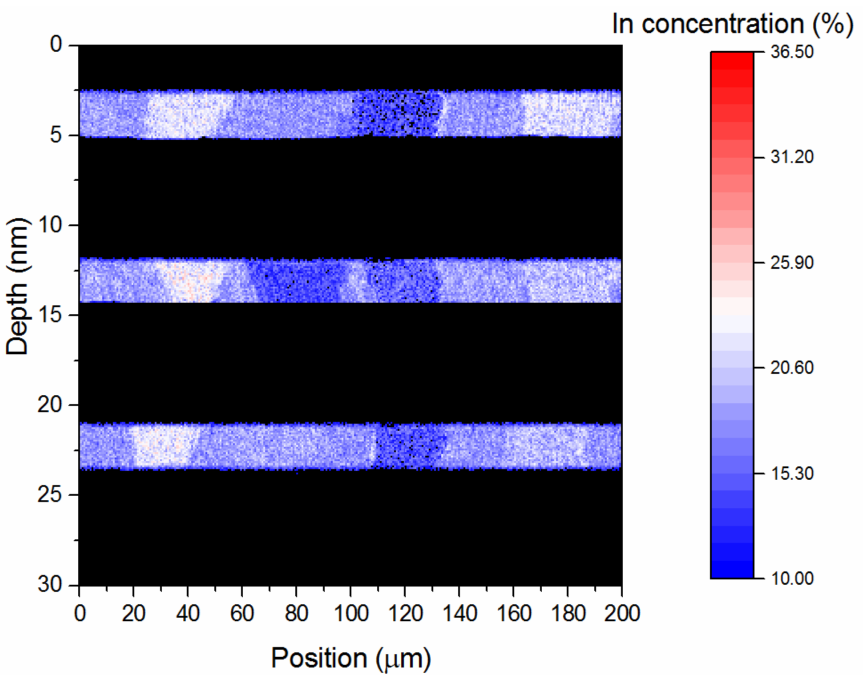


Homogenized 905ºC

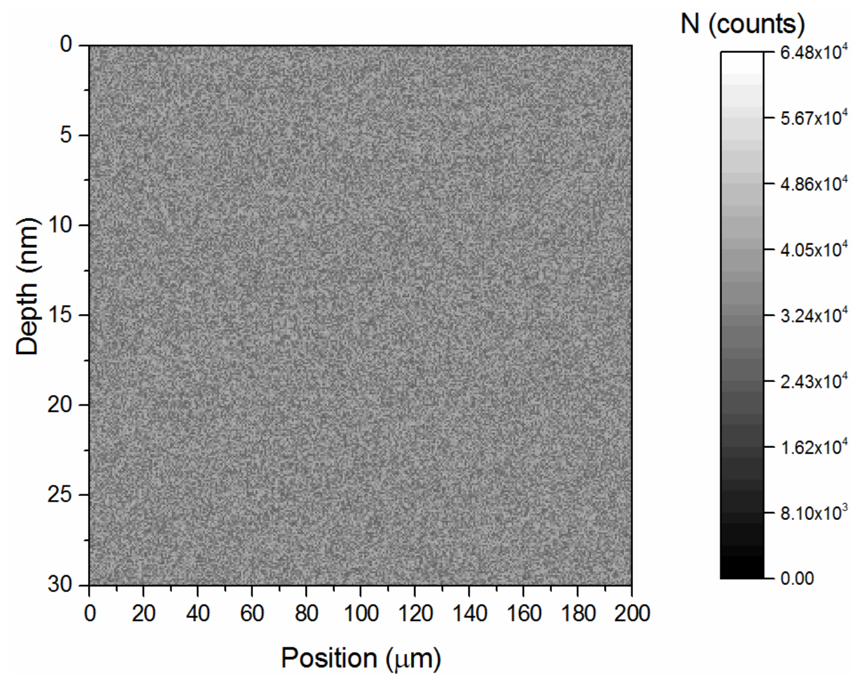


Decomposed 930ºC

as grown

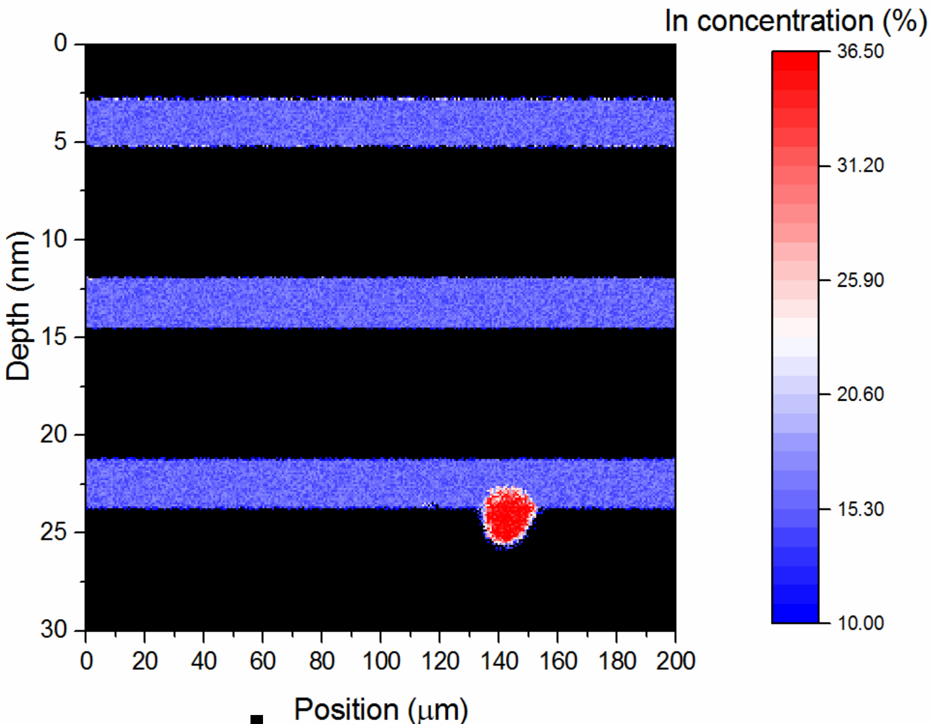

In concentration (%)
36.50
31.20
25.90
20.60
15.30
10.00
Depth (nm)
0
5
10
15
20
25
30
Position (μm)
0
20
40
60
80
100
120
140
160
180
200

In

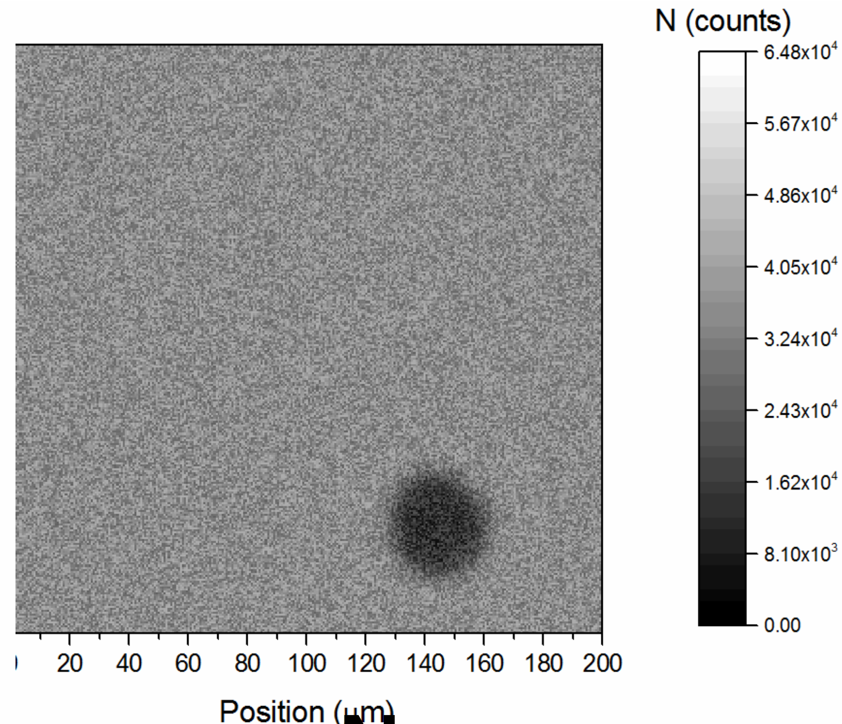

N (counts)
6.48x10^4
5.67x10^4
4.86x10^4
4.05x10^4
3.24x10^4
2.43x10^4
1.62x10^4
8.10x10^3
0.00
Depth (nm)
0
5
10
15
20
25
30
Position (μm)
0
20
40
60
80
100
120
140
160
180
200

N

homogenozed

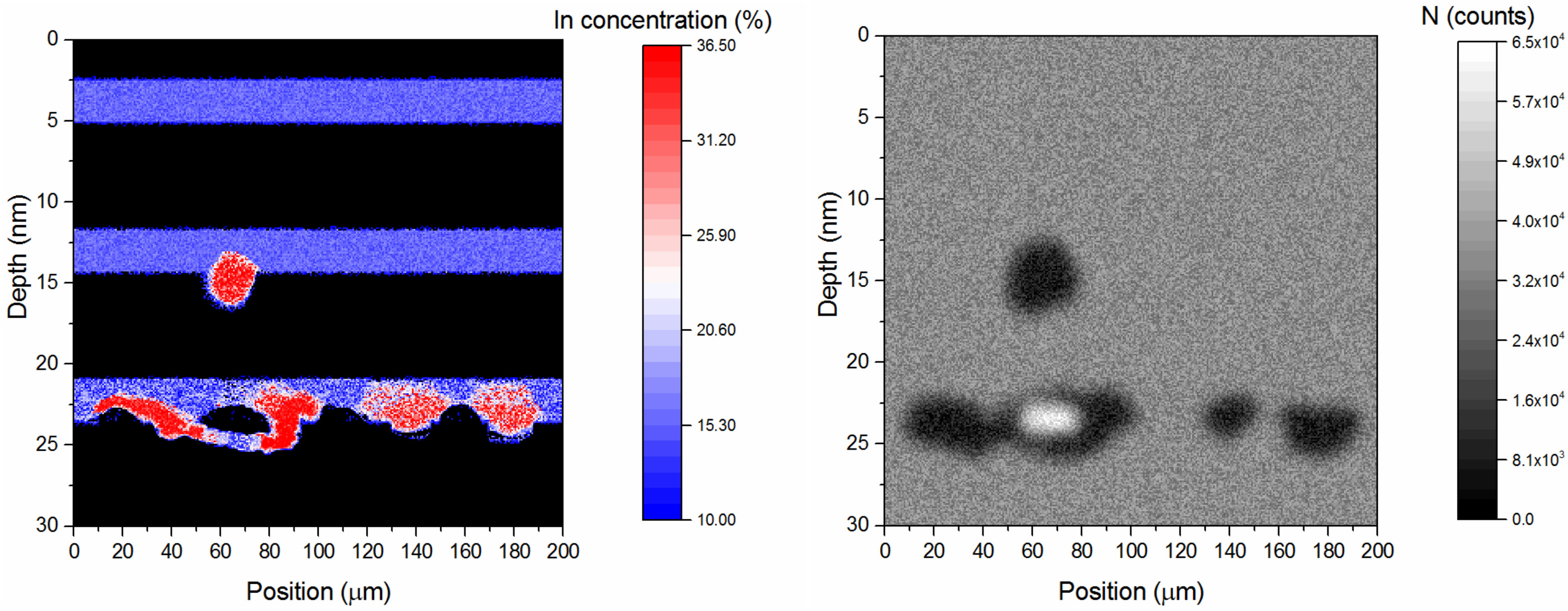


In



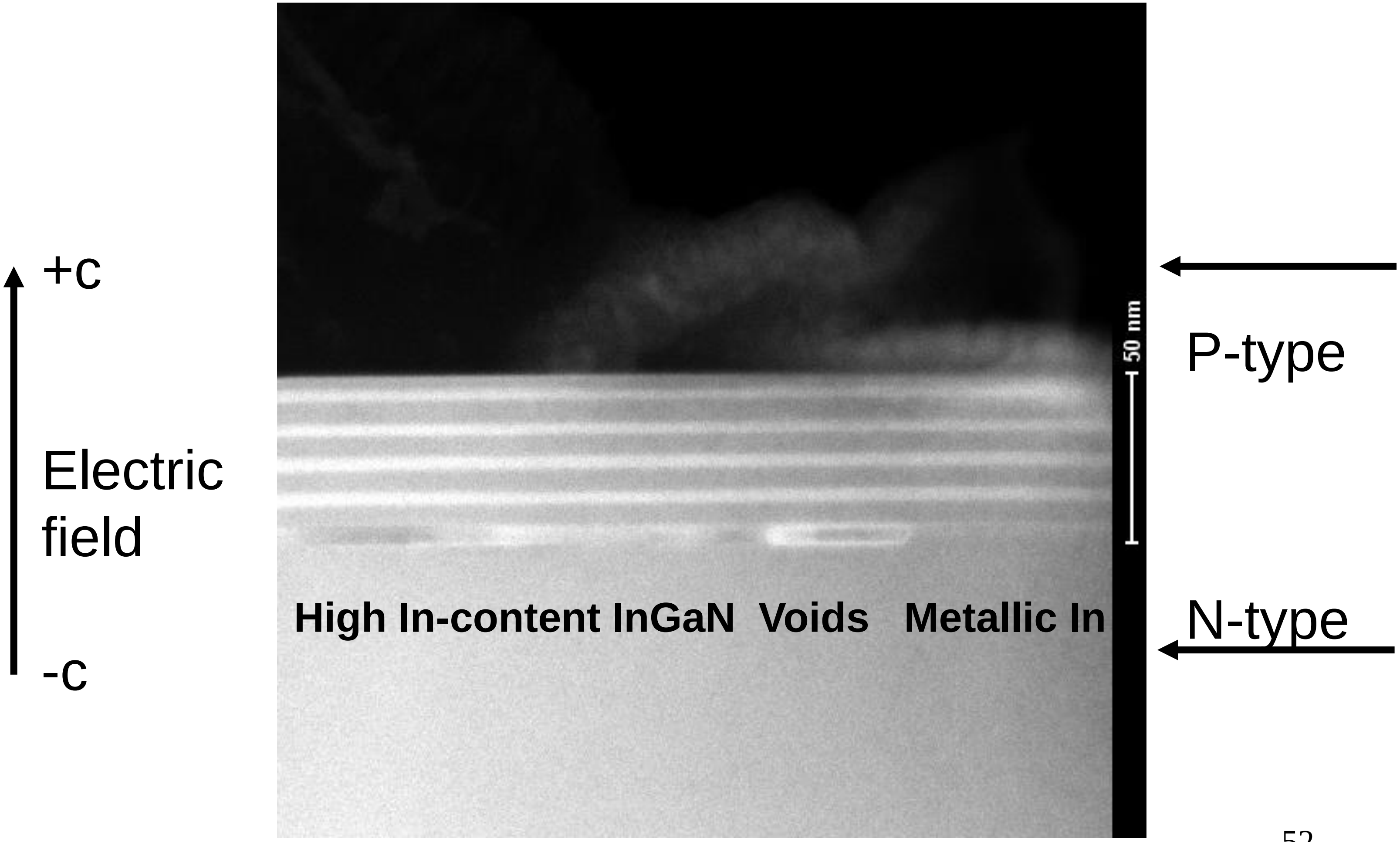
N

decomposed



In **N**

Why does the decomposition start from the first QW?



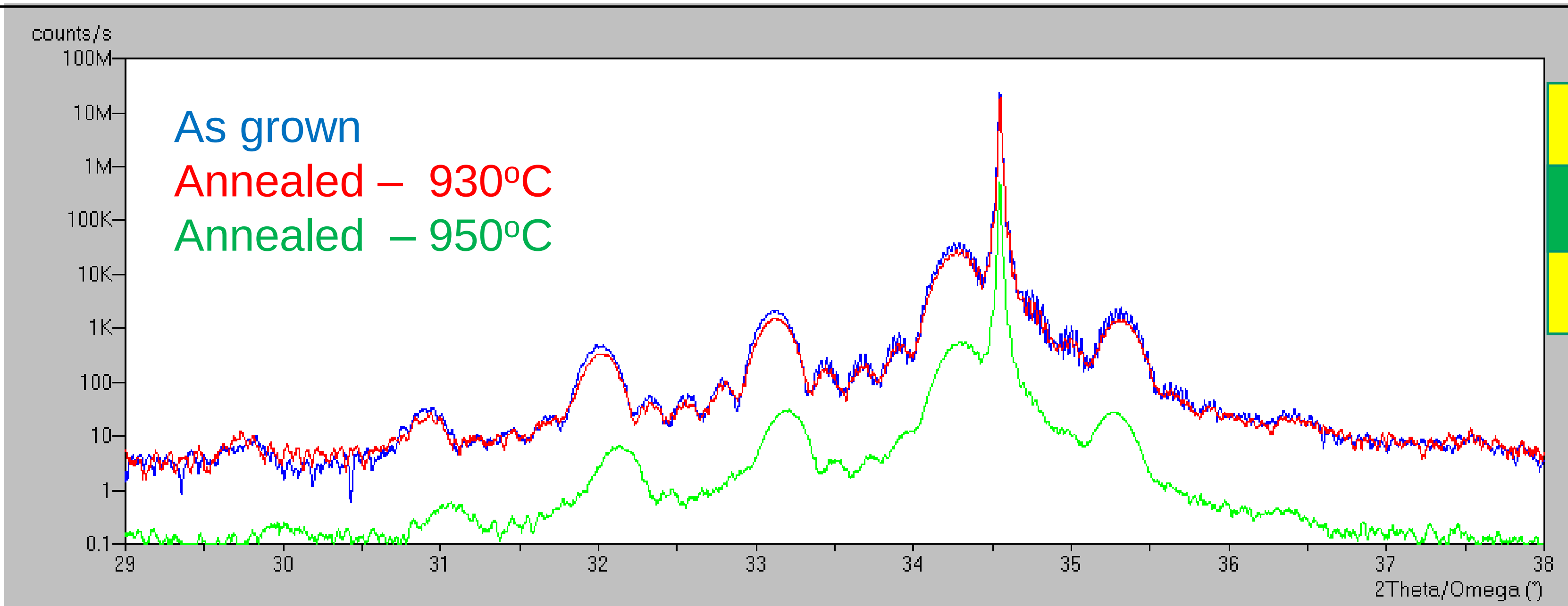
Diffusion of vacancies?

In GaN:Si we have more Ga-vacancies

In GaN:Mg we have more N-vacancies

Diiferent doping below and above the InGaN/GaN MQWs

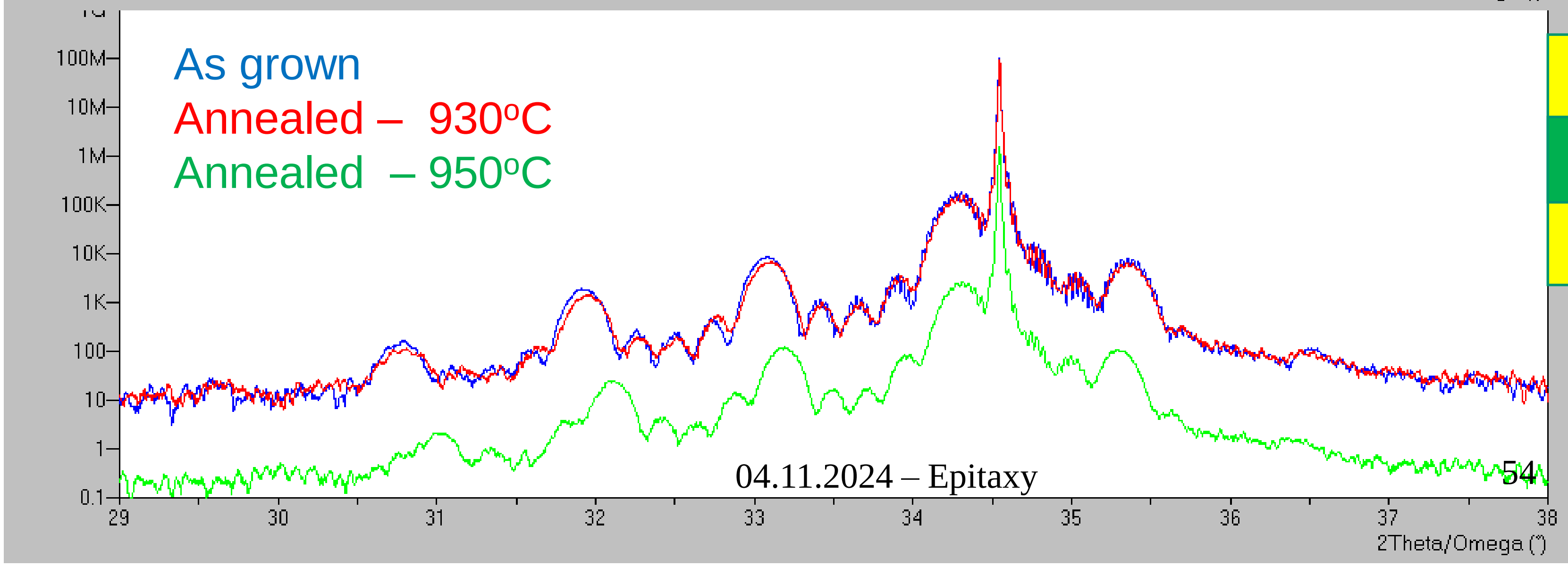
5QWs - (460 nm) - 17%In



GaN:**Si**

5QWs

GaN:**Si**



GaN:**Mg**

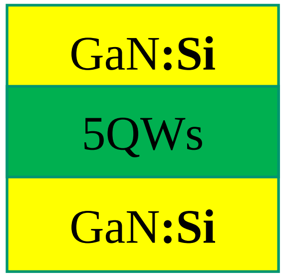
5QWs

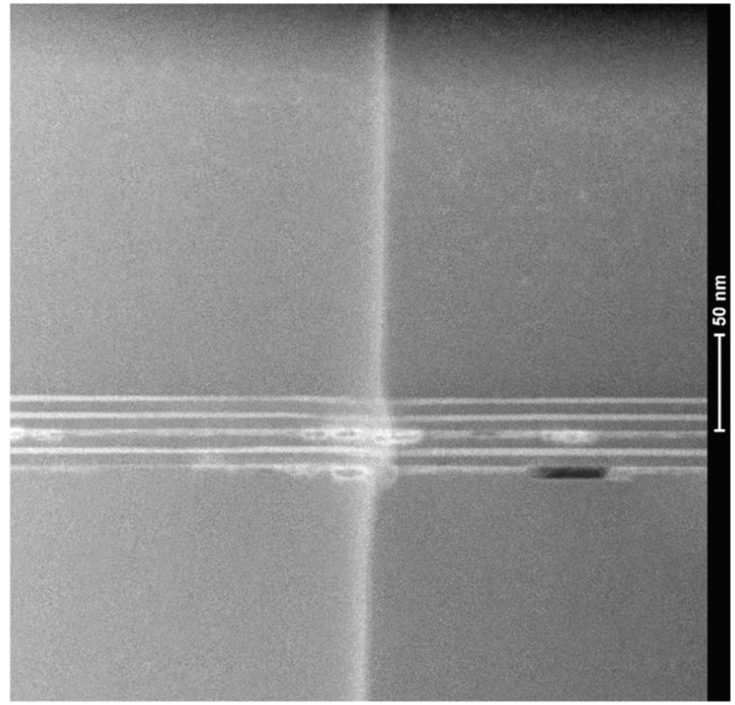
GaN:**Mg**

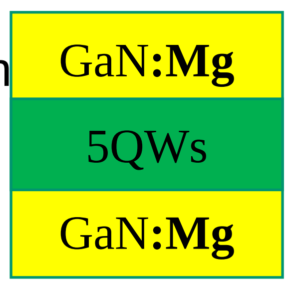
After annealing at 950°C

No decomposition from the top

Electric field driven diffusion? Yes!

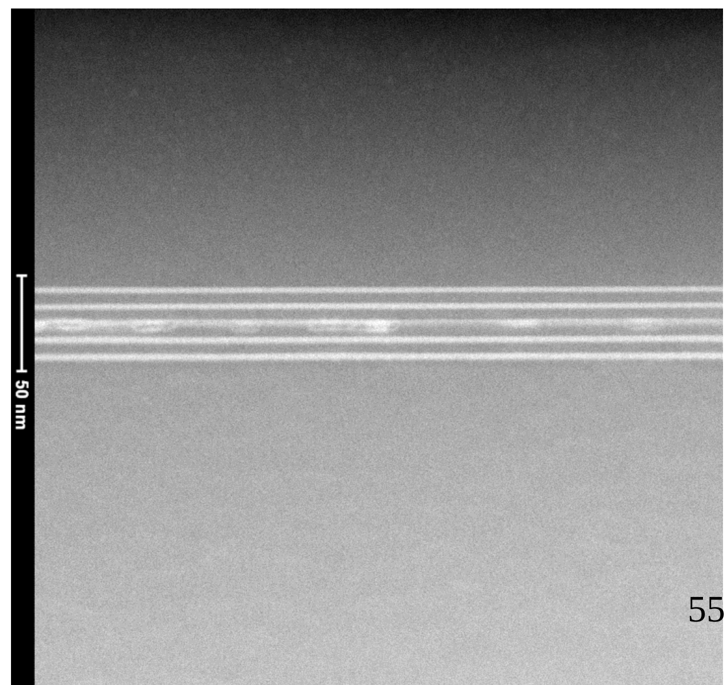






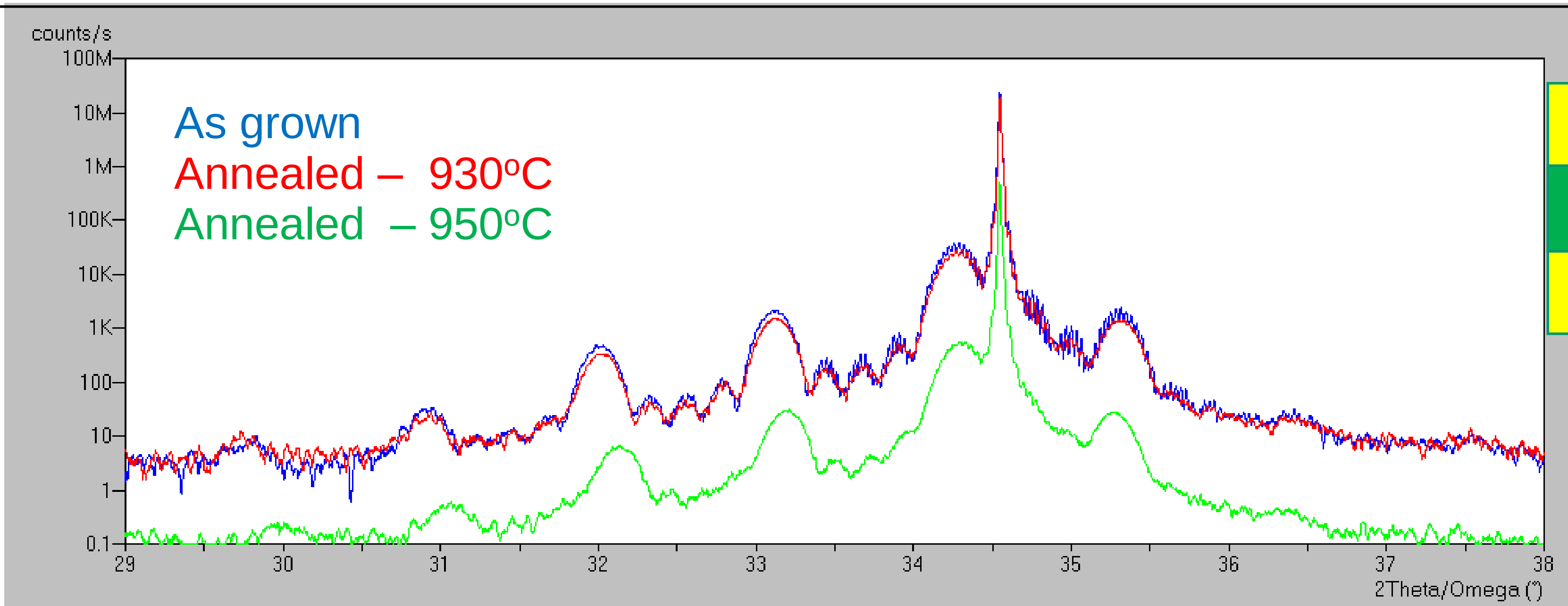
No decomposition above and below GaN:Mg

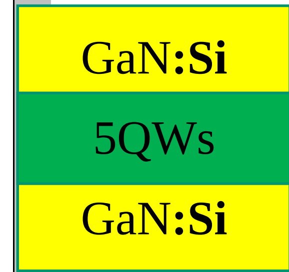
Ga-vacancies? Yes!

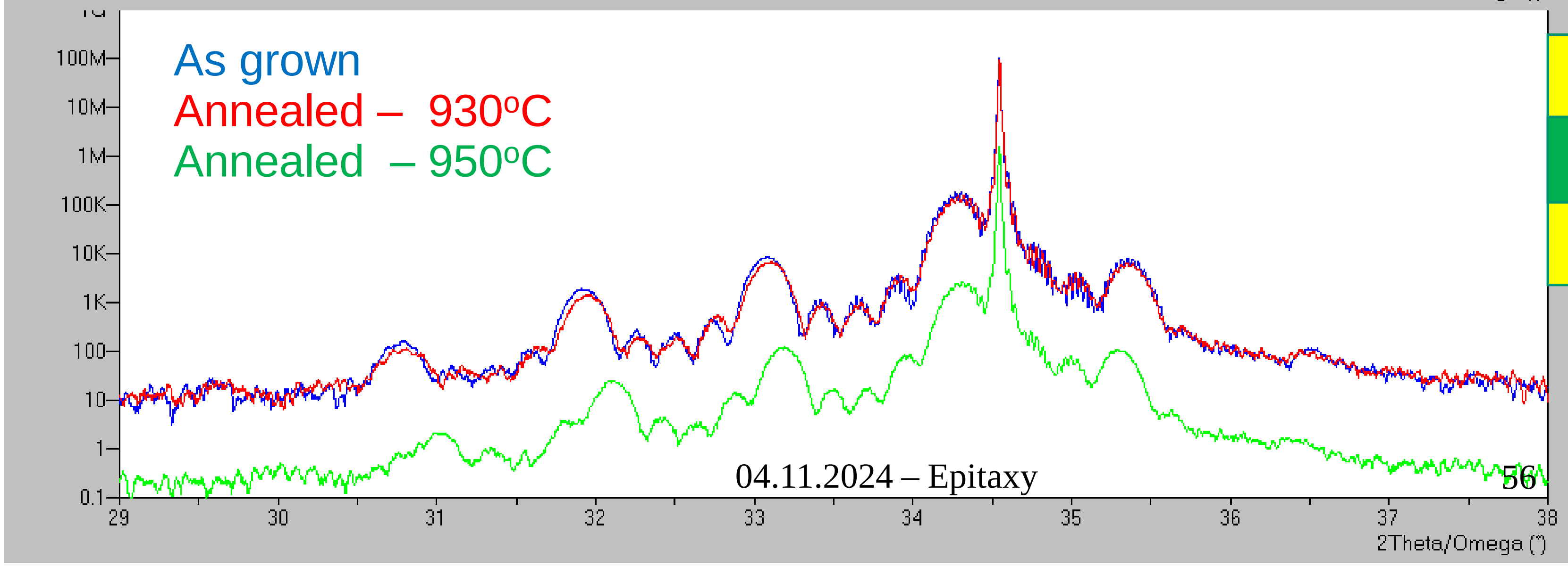


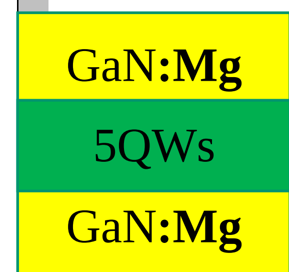
Diiferent doping below and above the InGaN/GaN MQWs

5QWs - (460 nm) - 17%In





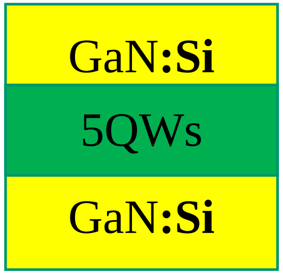


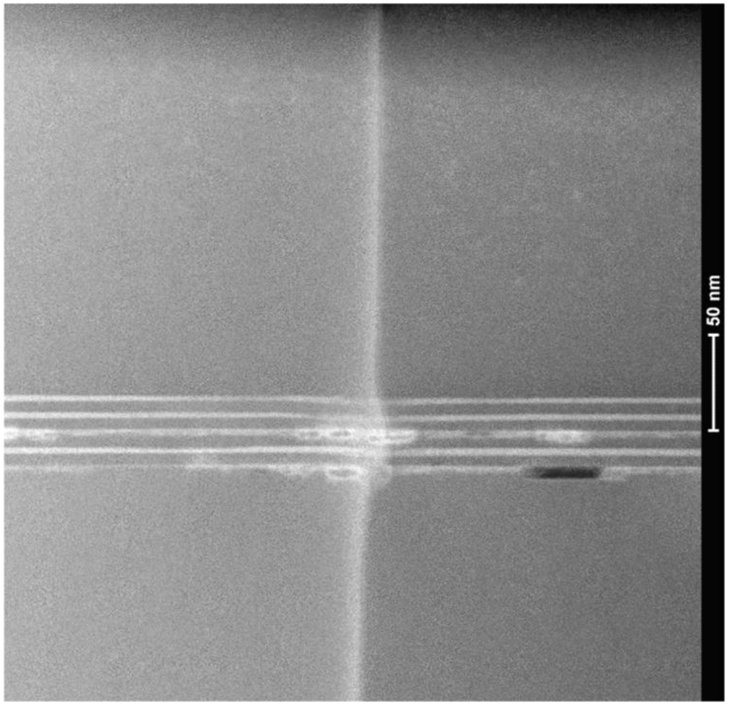


After annealing at 950^{o}C

No decomposition from the top

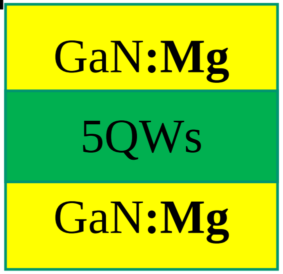
Electric field driven diffusion? Yes!

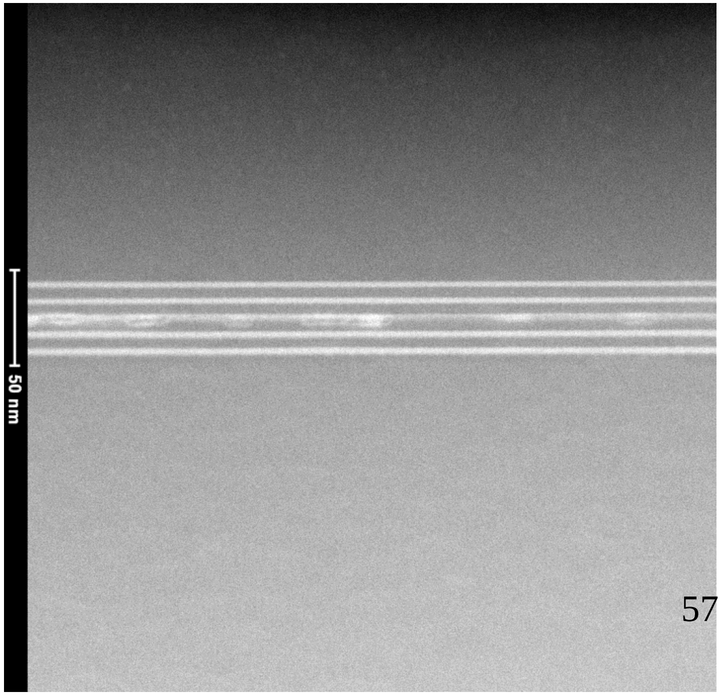




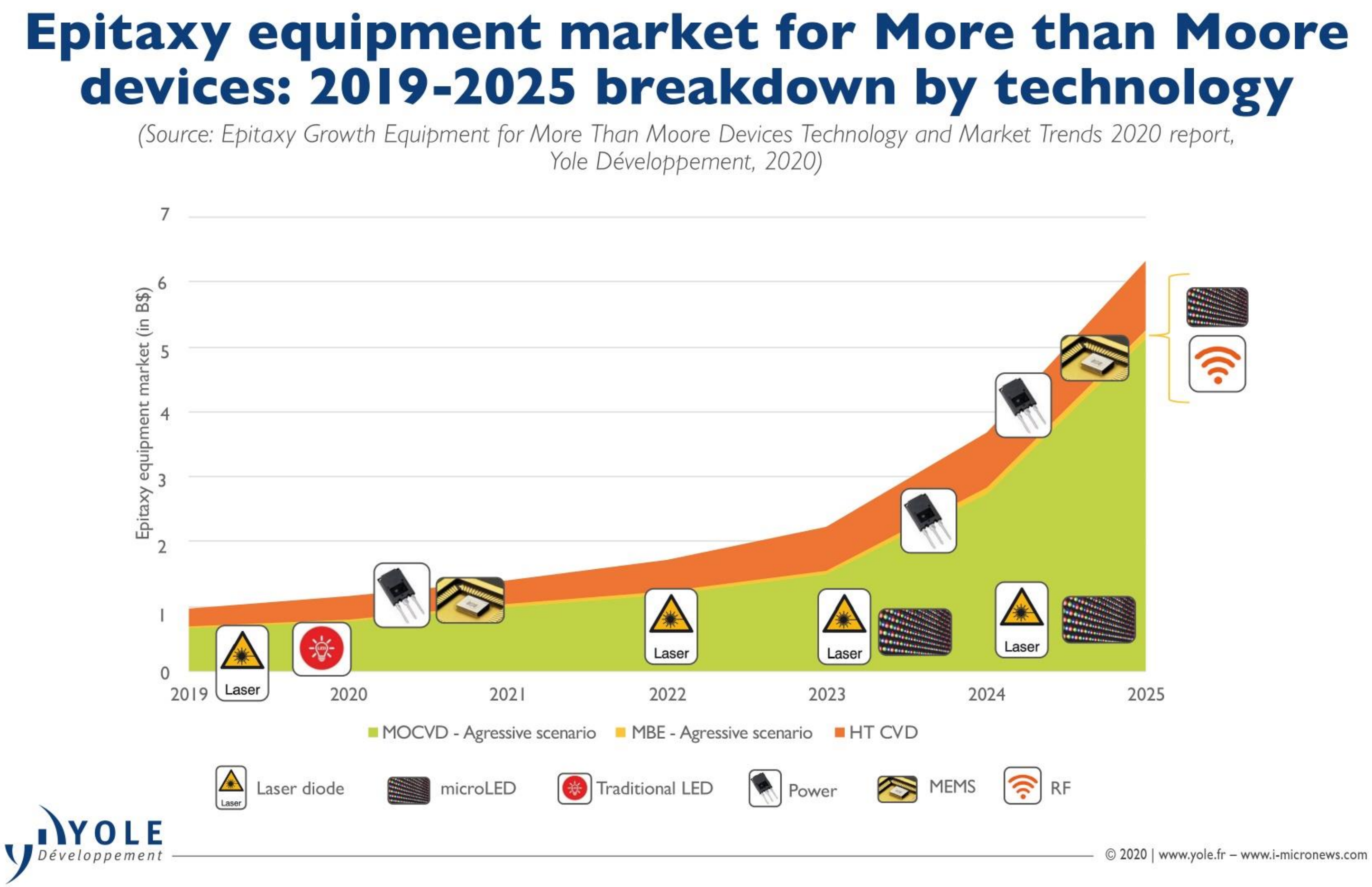
No decomposition above and below GaN:Mg

Ga-vacancies? Yes!





Thank you for your attention!



About 500 epitaxial systems installed every year, 2000 in 2025.