

Crystal Growth: Physics, Technology and Modeling

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Lecture 4. Molecular beam epitaxy

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<http://www.unipress.waw.pl/~stach/cg-2024-25>

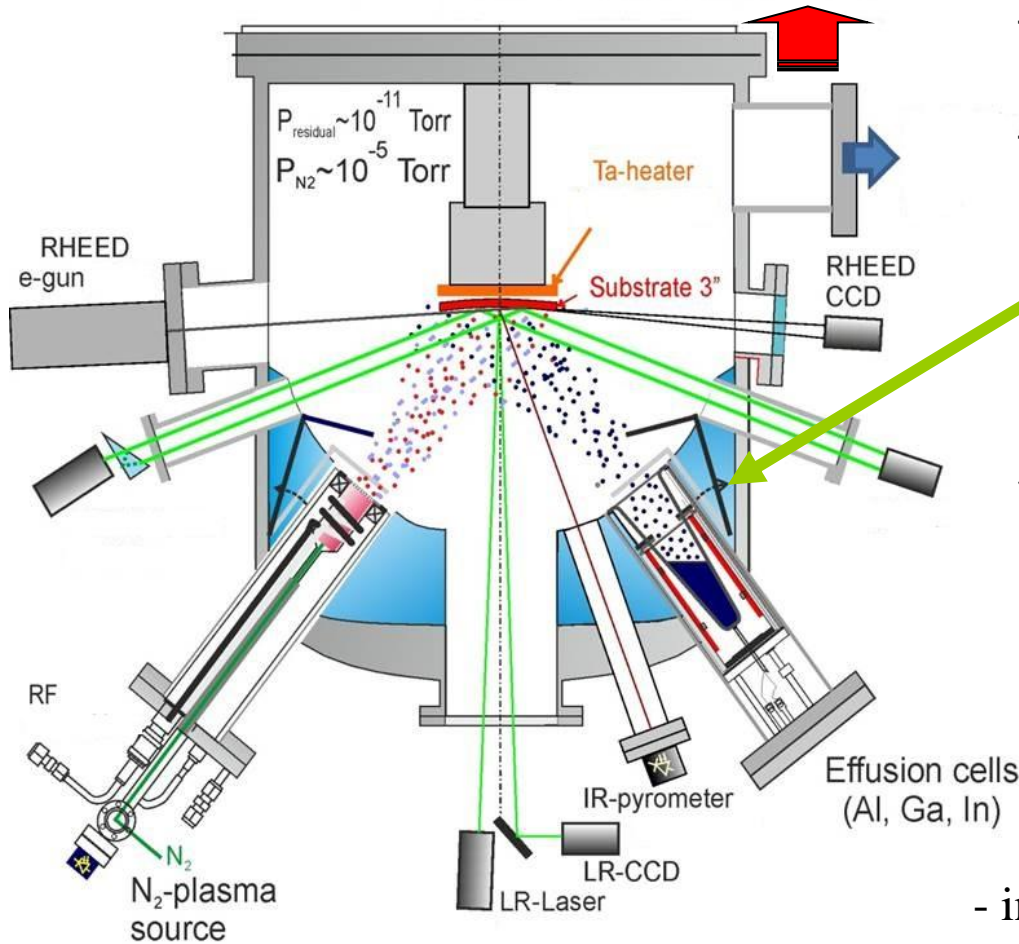
Molecular beam epitaxy - MBE

pol. epitaksja metodą (z) wiązek molekularnych

Outline:

- **idea and physical background of MBE**
- **technical aspect of MBE growth**
- ***in situ* monitoring of MBE growth**
- **examples of MBE grown structures**
 - **low-temperature growth**
 - **superlattices**
 - **quantum dots and nanowires**
- **summary**

pumping system

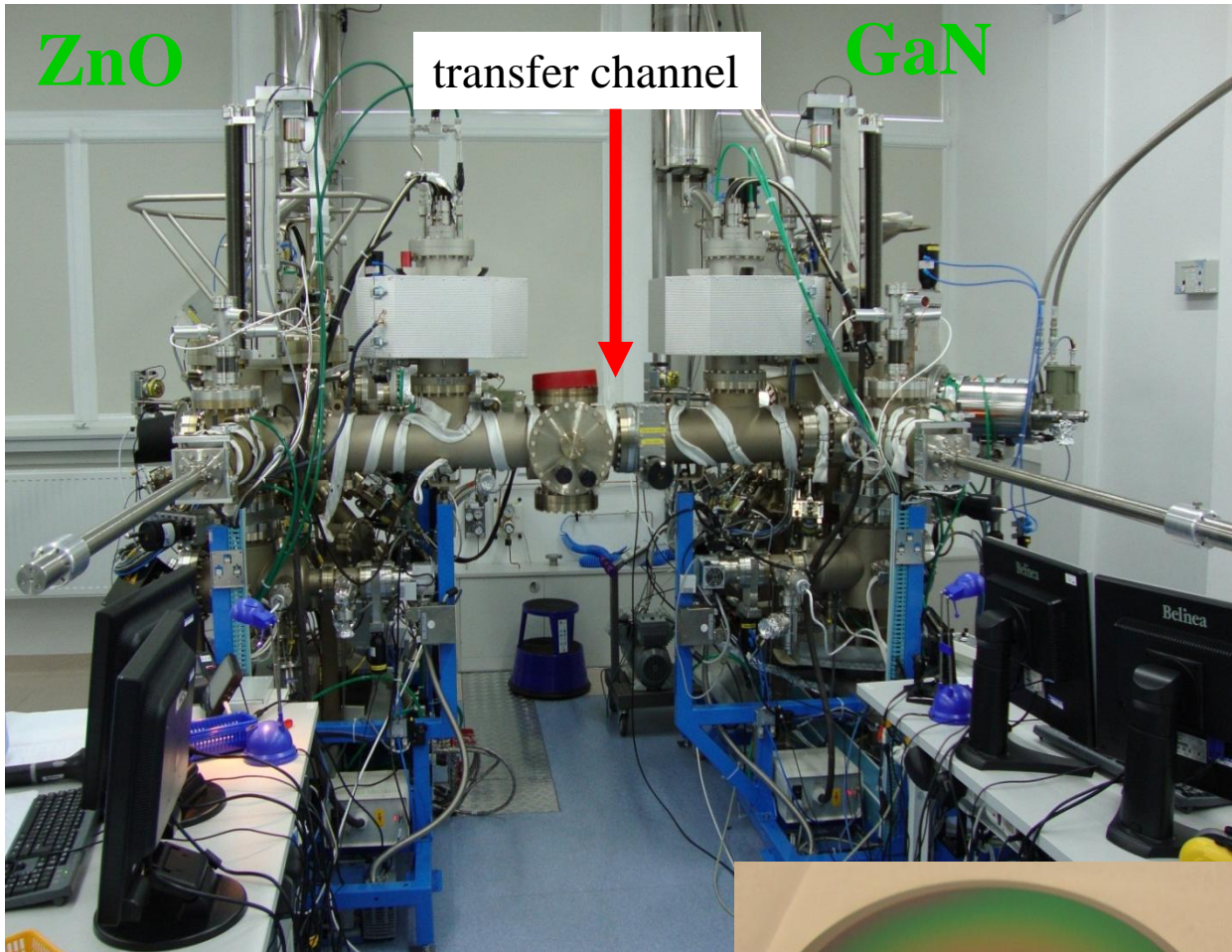


- UHV conditions
 - background pressure $10^{-10} - 10^{-11}$ Tr
- LN₂ filled cryoshroud :
 - additional pumping
 - freezing out atoms on the walls
 - lower „memory effect”
 - thermal separation of the sources

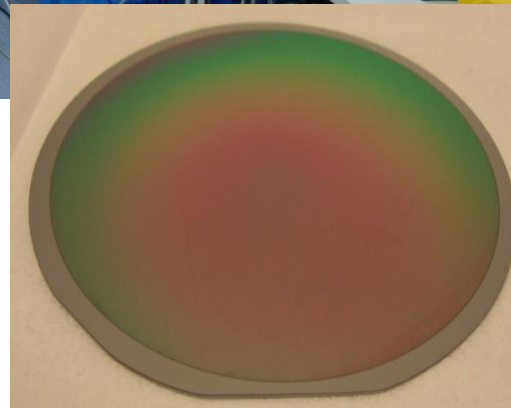
$\sim 10^{-5}$ Tr inside the molecular beam

- independent sources of atoms/molecules;
 - the flux usually controlled by T_{source}
- measurement of the flux – flux monitor
- mechanical shutters to open/close the source
- substrate heated by the heater
 - $T = \sim 200 \text{ }^\circ\text{C} - \sim 1000 \text{ }^\circ\text{C}$
- many *in-situ* diagnostic tools available

Plasma-Assisted MBE (PAMBE) Riber Compact 21



growth on
3" substrates
(4" possible)



TOOLS:

- ▶ optical pyrometer
- ▶ RHEED (k-Space)
- ▶ laser reflectometry
- ▶ LayTec EpiCurve TT (temperature, wafer curvature)
- ▶ line-of-sight quadrupole mass spectrometry (QMS)

SOURCES:

- ▶ Ga x2
- ▶ Al x2
- ▶ In
- ▶ **RF nitrogen source**
- ▶ Si x2
- ▶ Mg
- ▶ Fe

UHV in MBE – meaning how much ?

condition 1: mean free path of atoms > source – substrate distance

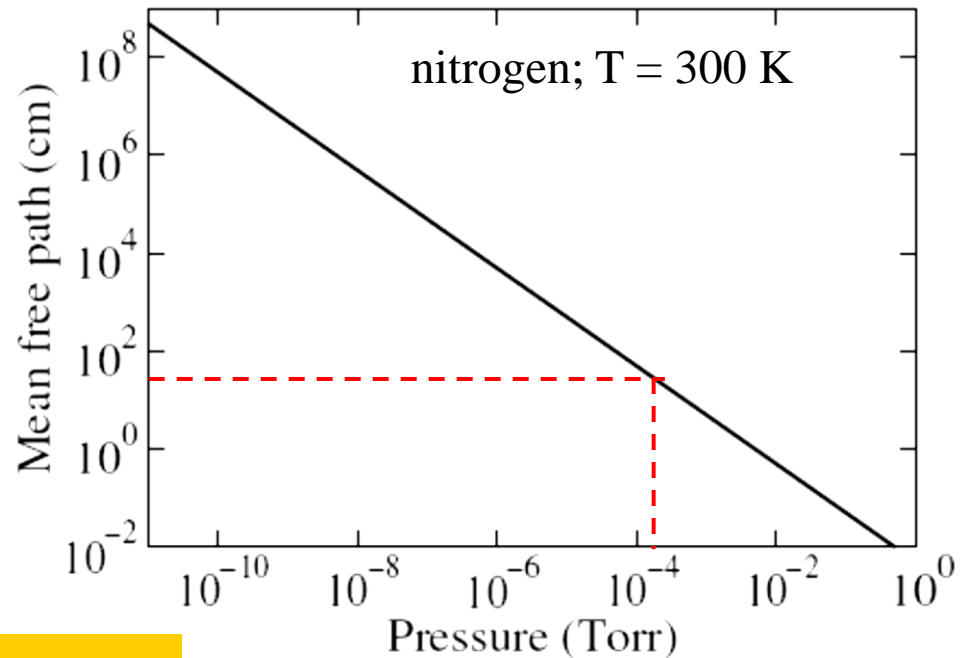
mean free path of atoms λ in gas phase under pressure p

$$\lambda \approx \frac{5 \times 10^{-4}}{p[\text{Tr}]} [\text{cm}]$$

$p = 10^{-4} \text{ Tr} \leftrightarrow \lambda = \sim 50 \text{ cm}$

$p = 10^{-7} \text{ Tr} \leftrightarrow \lambda = \sim 0.5 \text{ km}$

$p = 10^{-11} \text{ Tr} \leftrightarrow \lambda = \sim 5\,000 \text{ km}$



ballistic atom transport (no collisions) in MBE
- no UHV conditions needed

UHV in MBE – meaning how much ?

condition 2: high purity of MBE grown layer

assumption: all particles/atoms arriving stick to the substrate (no desorption)

flux of particles in a gas under pressure p on area of 1 m² in 1 sec.

$$J = \frac{p \cdot \sqrt{N_{Av}}}{\sqrt{2\pi m k_B T}} \quad \text{if } m=28 \text{ (N}_2\text{); } T=300\text{K then } J = 2 \times 2.8 \cdot 10^{22} \times p \left[\frac{\text{atoms}}{\text{m}^2 \text{s}} \right]$$

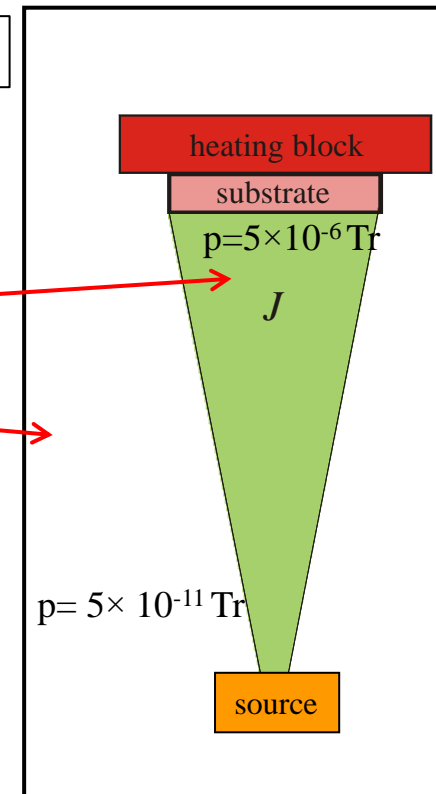
number of lattice places on Si surface $N = 2.8 \cdot 10^{19} [\text{atoms} \cdot \text{m}^{-2}]$

time to cover Si by 1 monolayer (ML) of particles $\tau = \frac{N}{J} = \frac{5 \times 10^{-4}}{p [\text{Pa}]}$

$$p = 5 \times 10^{-6} \text{ Tr} \leftrightarrow \tau = 1 \text{ sec}$$

$$p = 5 \times 10^{-11} \text{ Tr} \leftrightarrow \tau = 10^5 \text{ s} \approx 28 \text{ h}$$

$p = 5 \times 10^{-11} \text{ Tr} \Rightarrow$ 1 impurity atom per 10⁵ Si atoms
impurity concentration in the bulk $\sim 10^{17} \text{ cm}^{-3}$



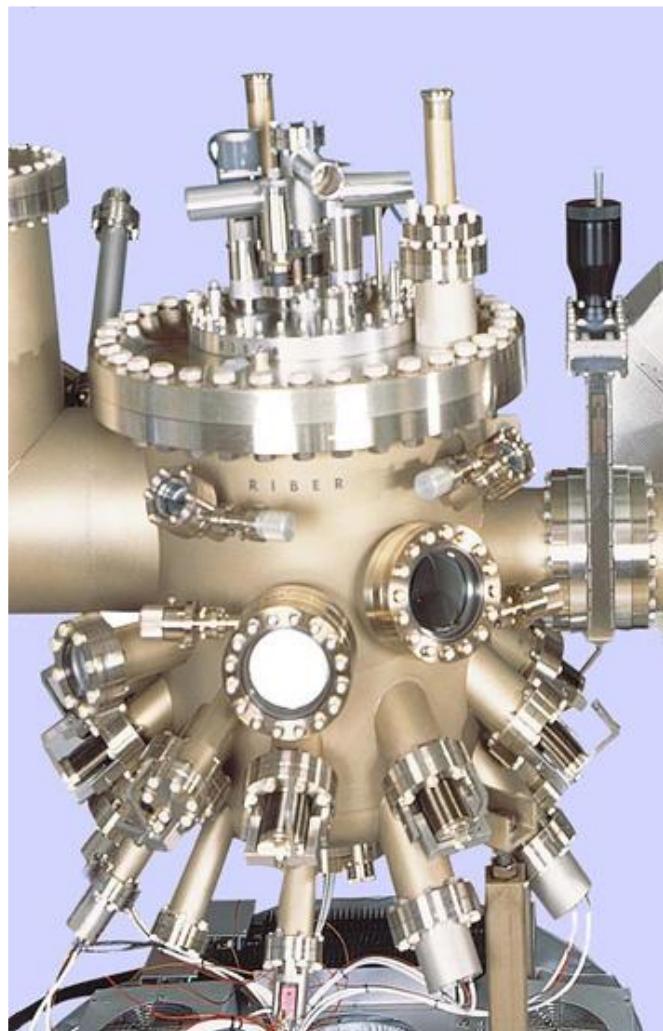
under real conditions: layers more pure ($\sim 10^{14} \text{ cm}^{-3}$ possible) since:
- often sticking coefficient < 1 (desorption important)

Three-chamber configuration of MBE system

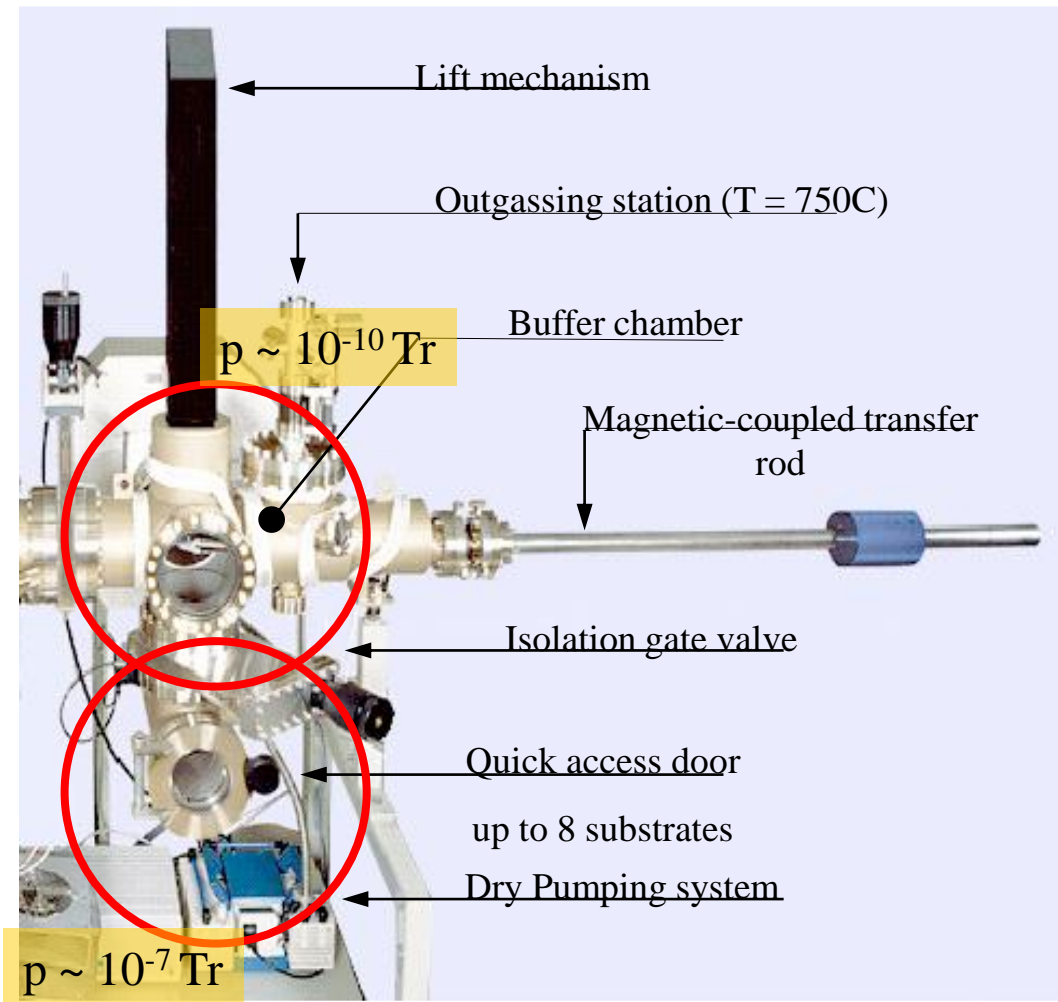
(example - Riber Compact 21)

growth chamber

loading and pre-annealing of the substrate



$p \sim 10^{-11} \text{ Tr}$



each chamber equipped with its own pumping system

Generation of vacuum

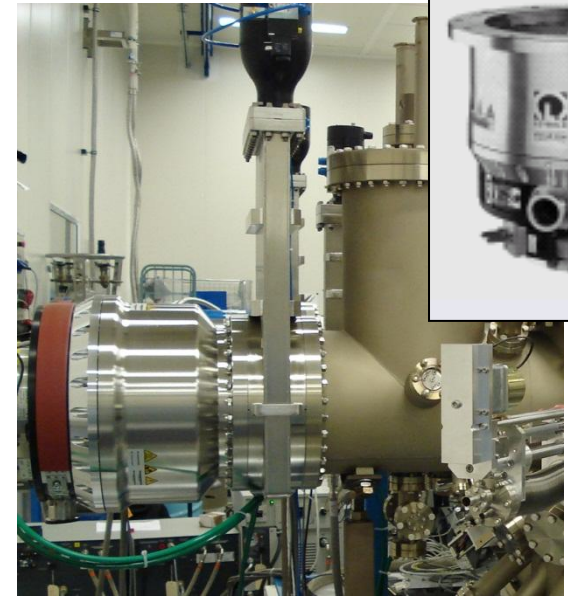
- mechanical pumps - rough pumps (rotary, Scroll, membrane, ..) and turbomolecular (UHV)
- cryopumps
- ion and titanium pumps



pumping speed (N₂) 1200 l/sec



Helix CTI-10; pumping speed (N₂) 3000 l/sec



pumping speed (N₂) 2800 l/sec



long annealing of all chambers at T ~ 150° C after each opening of the system to remove residual gases and water



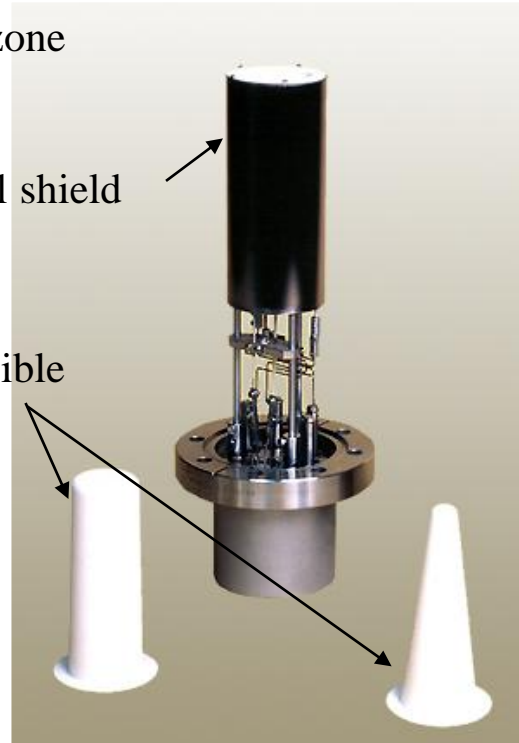
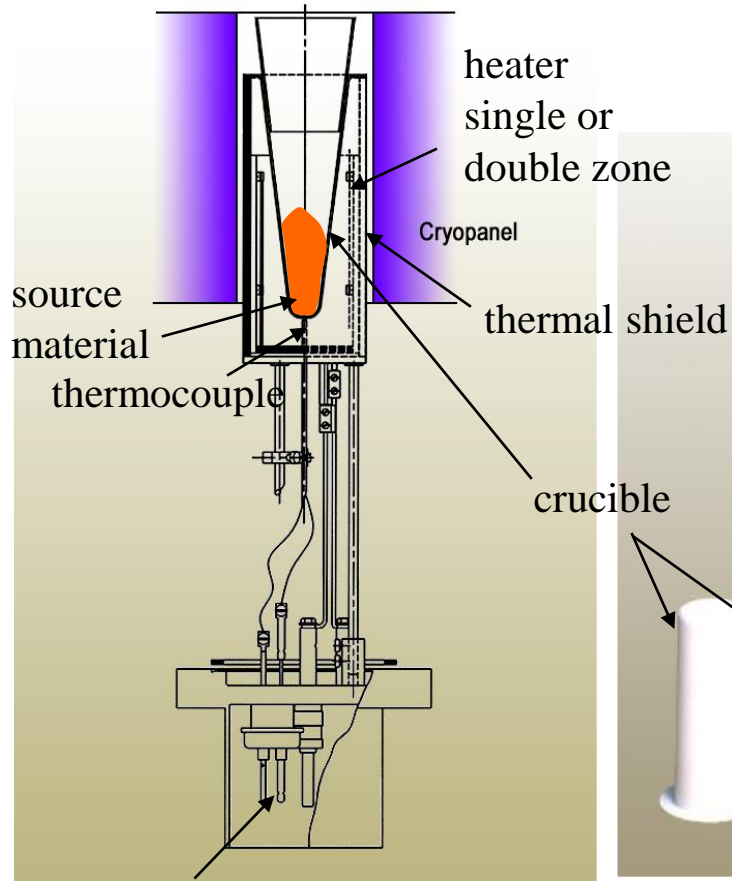
Generation of molecular beams – effusion (Knudsen) cell

shutter



Modern molecular sources:

- many sources in one system (10 in Compact 21 Riber)
- sources centered at the substrate \Rightarrow flux uniformity
- high flux stability;
 - flux drift $< 1\%/day \Rightarrow \Delta T < 1^\circ C @ T \sim 1000^\circ C$
- each cell equipped with its own shutter
- cells thermally separated from each other



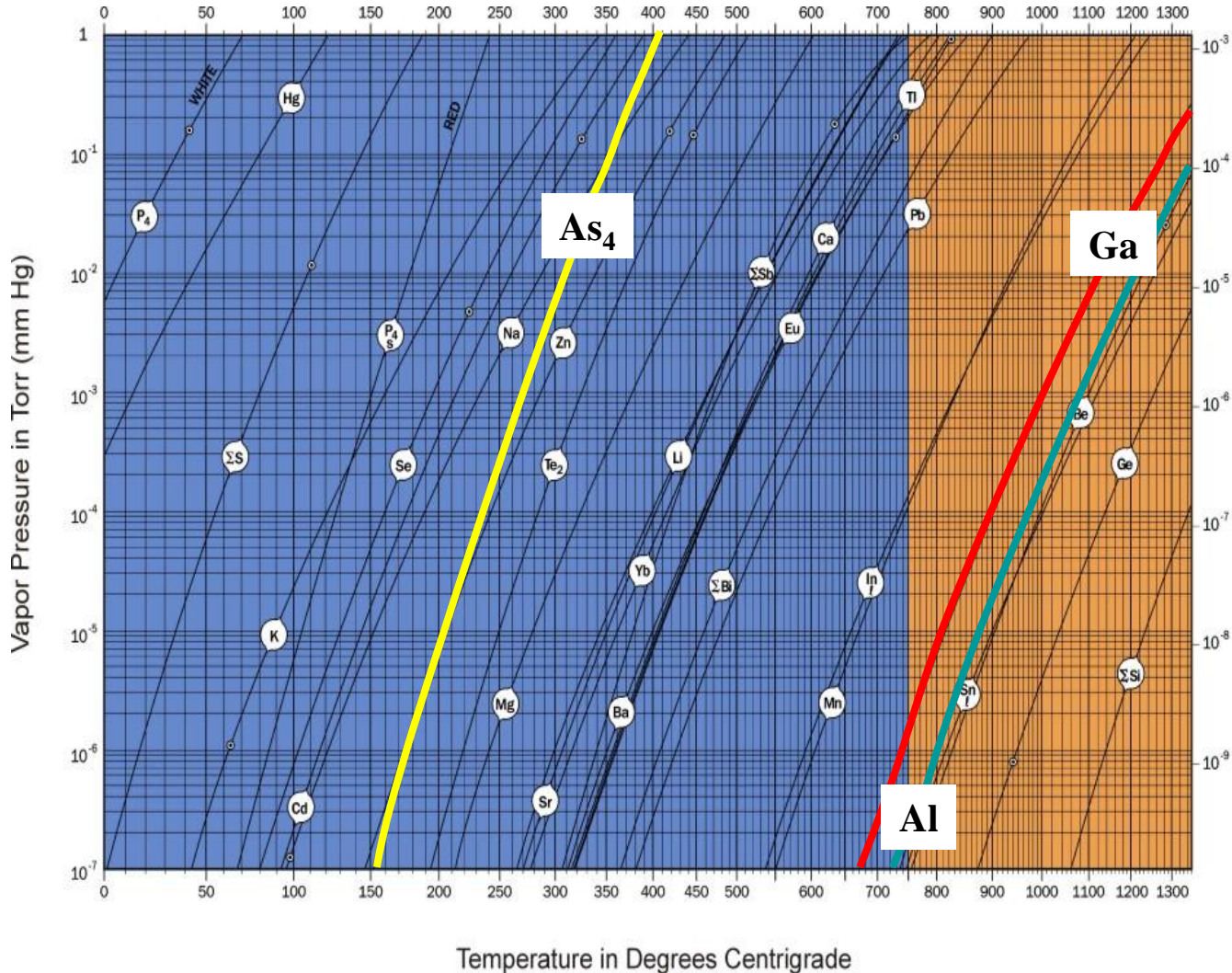
holes for sources and shutters
in cryopanel of Compact 21
Riber system



Generation of molecular beams – effusion (Knudsen) cell

assumption: vapor – liquid/solid equilibrium in the cell

vapor – liquid/solid equilibrium plots for selected elements



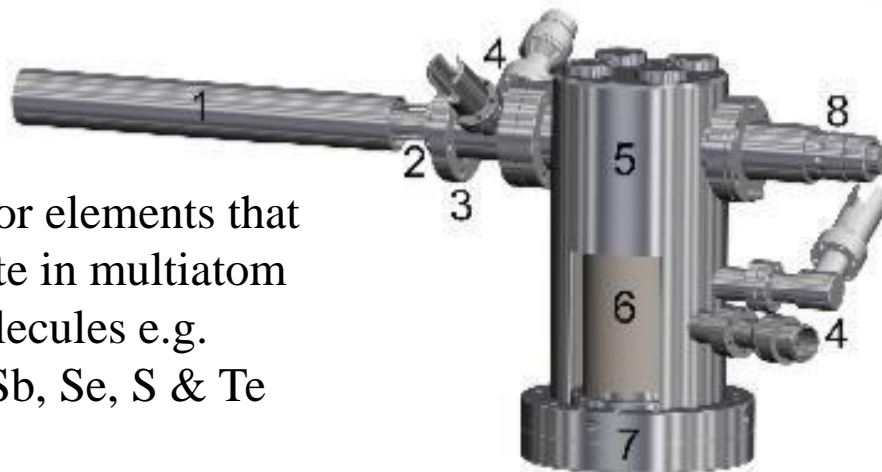
by changing T_{source} the flux from the cell p is controlled

Generation of molecular beams –special sources

valved cracker

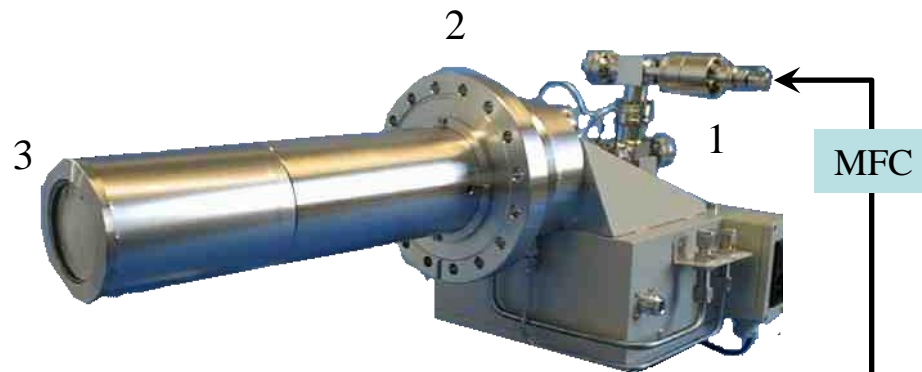
1. Cracking zone $As_4 \rightarrow As_2$
2. connector + needle valve
3. main flange
4. power and TC connection
5. generation of As_4 vapor
6. crucible with solid As

source for elements that sublime in multiatom molecules e.g.
As, P, Sb, Se, S & Te



plasma source

1. inlet of purified gas (MFC)
2. RF cavity
3. exit aperture (plate with small holes)



stable molecules N_2 , O_2 , etc.
excited in the cavity to produce
active gas species

gas injectors

gas sources with needle valve to deliver precursors used in Gas Source MBE (e.g. SiH_4) or metalorganics in MO MBE

growth rate in MBE – example GaAs

growth under As-rich conditions;
 growth rate V_{gr} controlled by Ga flux;
assumption: no Ga desorption

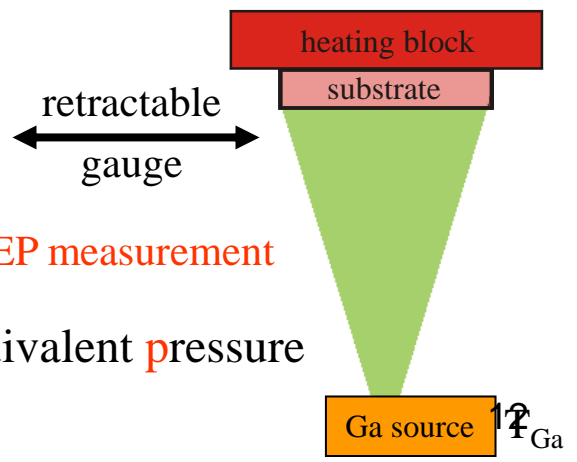
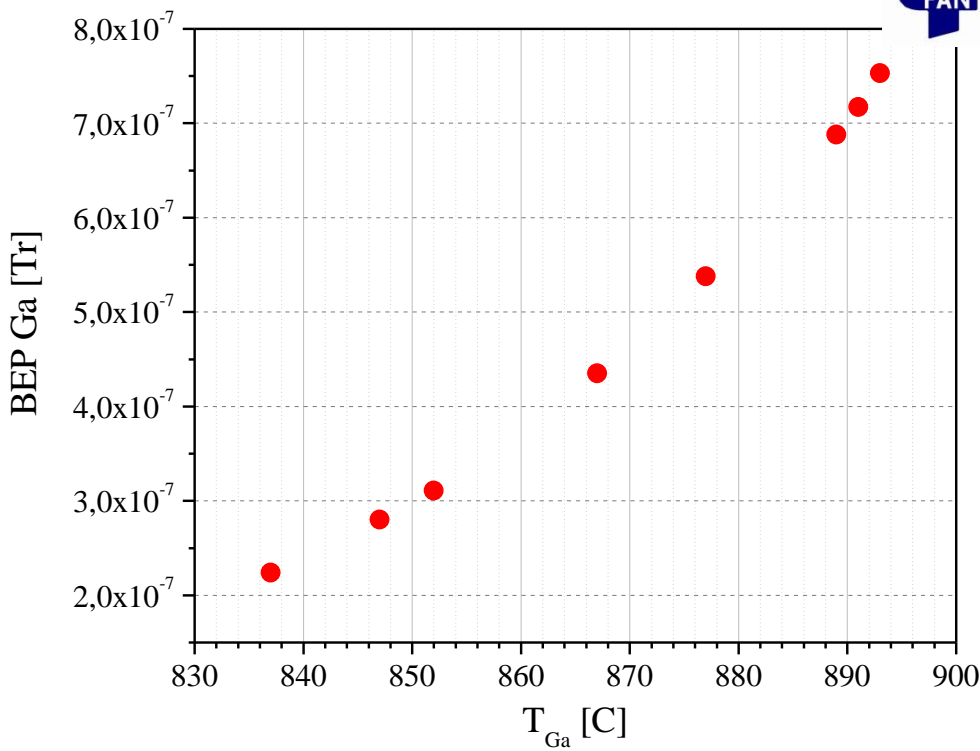
Ga flux $J = 1.18 \times 10^{15} \frac{at}{cm^2 s}$

spec. volume of GaAs $\Omega_0 = 2.27 \times 10^{-23} cm^3$

$$V_{gr} = J\Omega_0$$

$$V_{gr} = 2.67 \text{ \AA/s} = 1 \text{ ML/s} = 0.96 \text{ }\mu\text{m/h}$$

controlled growth of very thin (~1 ML) layers and epitaxial structures



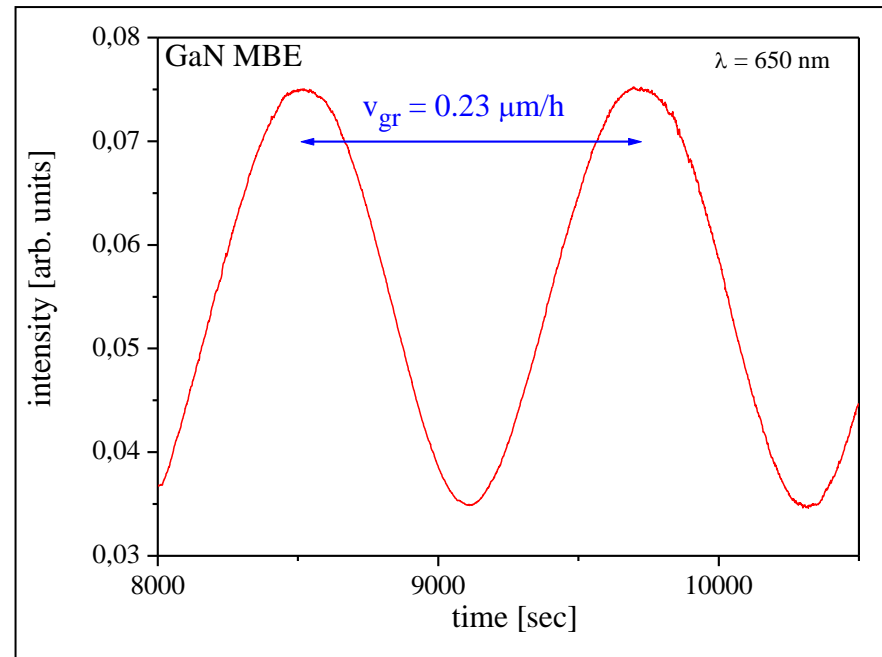
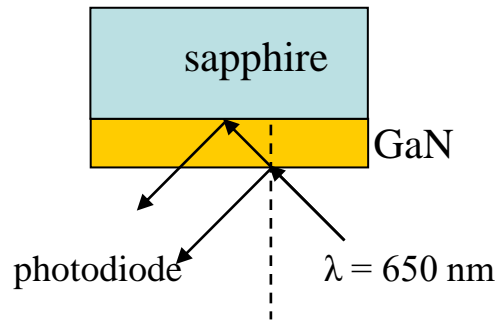
BEP = beam equivalent pressure

in situ growth monitoring

vacuum is transparent for light, electrons, X-rays, ...
wide possibilities to „observe” surface of growing epilayer

laser reflectometry

growth rate, evolution of surface roughness, ...

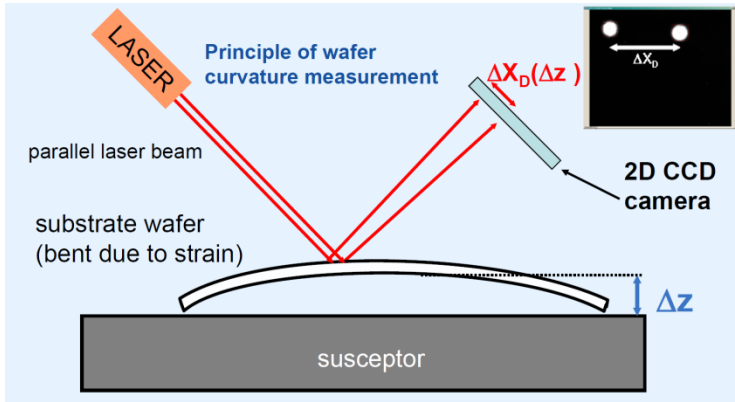


optical pyrometry *measurement of sample T based on the black body emission spectrum*

to be discussed later

ellipsometry

in situ growth monitoring

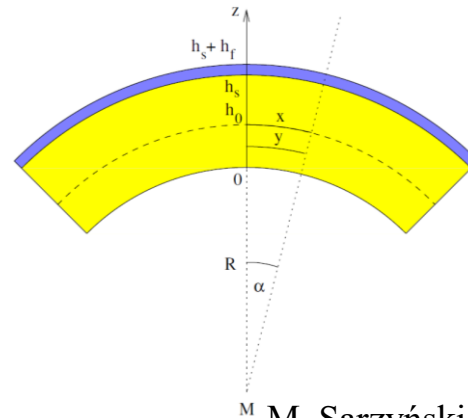


laytec.com

sample curvature

stress and bow evolution in real time

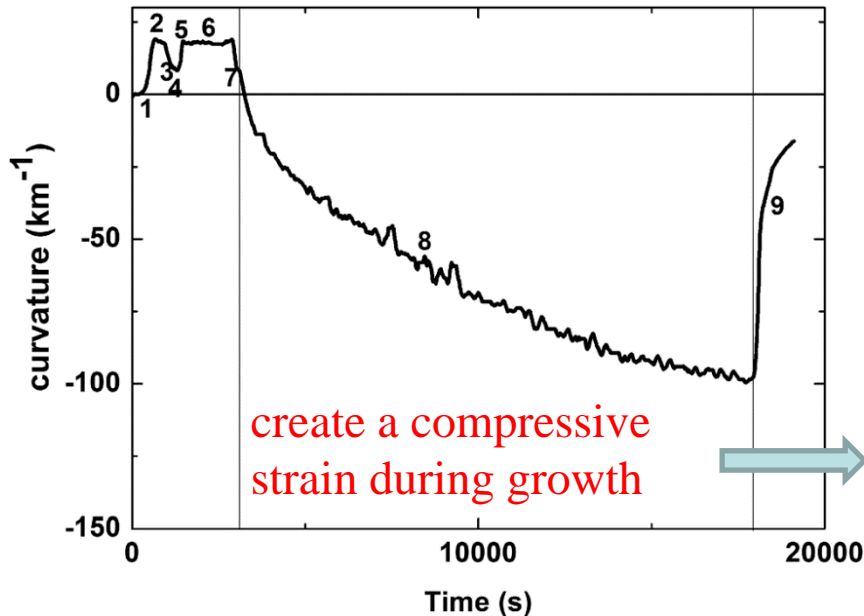
LayTec – EpiCurve; Riber – EZ-Curve, k-Space - kSA MOS...



$$1/R \propto \frac{\epsilon \times h_e}{h_s^2}$$

Stoney formula

M. Sarzyński PhD 2007

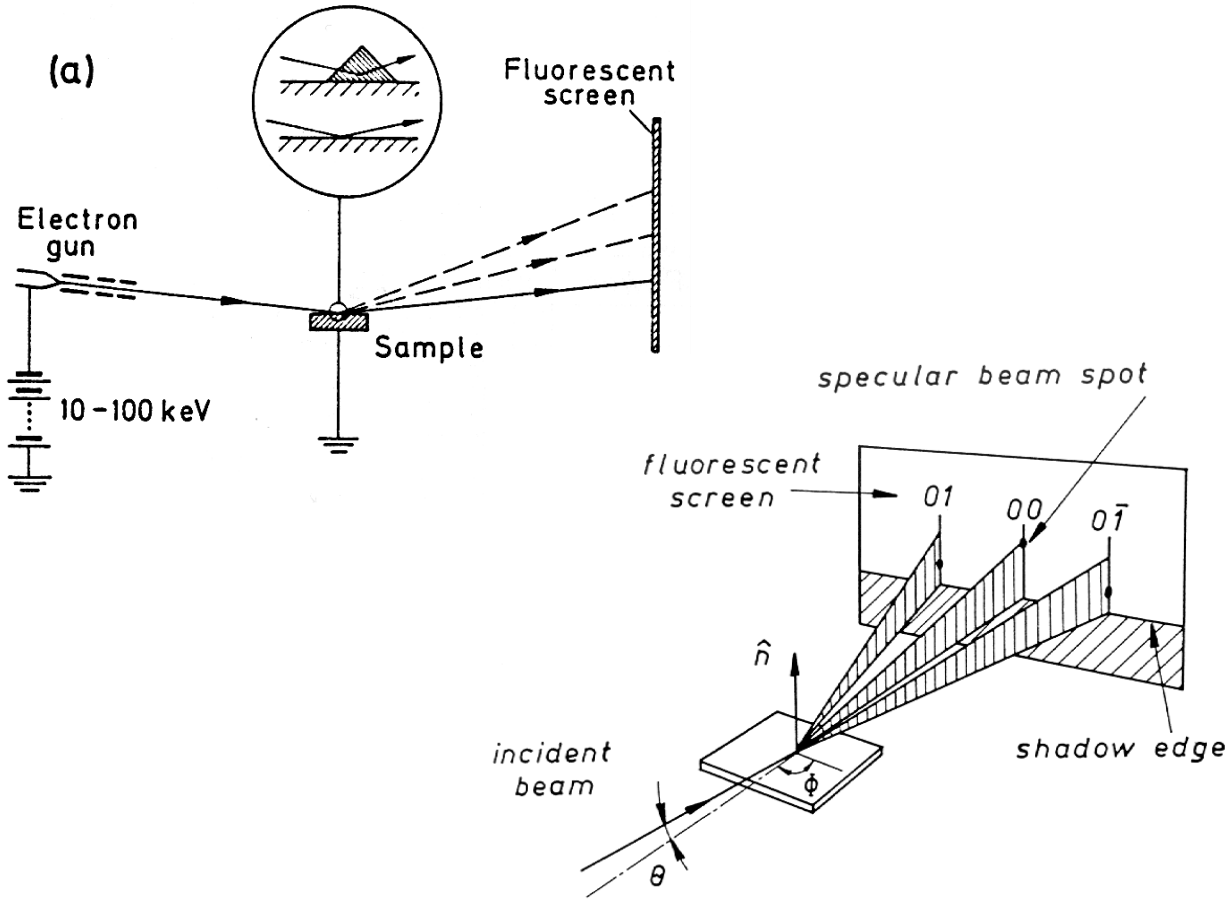


PAMBE growth of GaN/AlGaN on Si(111): mismatch of thermal contraction leads to tensile strain and cracking upon cooling

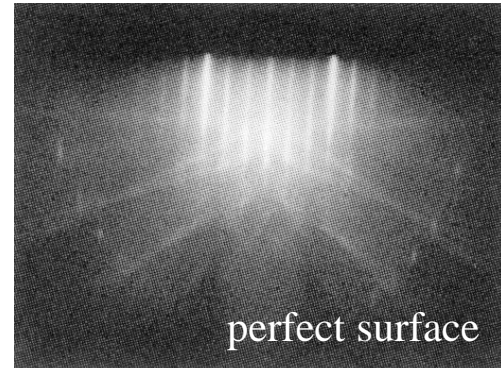
thick GaN buffers can be grown on Si w/o cracks

in situ growth monitoring - reflection high energy electron diffraction (RHEED)

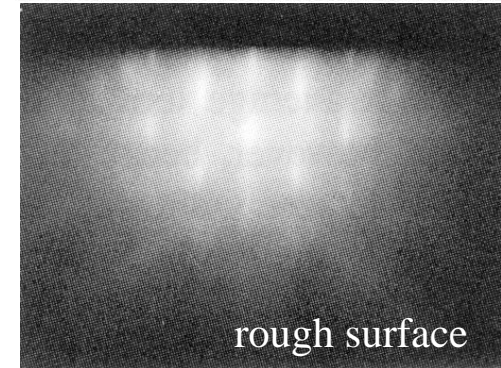
- probing of surface by diffraction of electron beam striking the sample at a very small angle relative to the sample surface ($1^\circ - 3^\circ$)
- electron energy 5 – 20 keV; wavelength $\sim 0.1 \text{ \AA}$
- ideal 2D surface – set of parallel streaks



Si(001) RHEED patterns sputter-cleaned surface



perfect surface

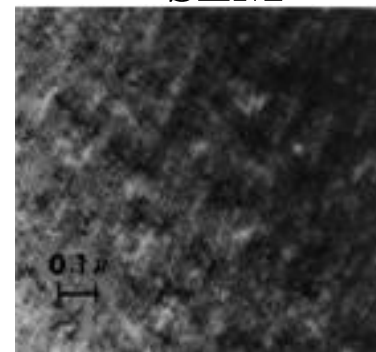
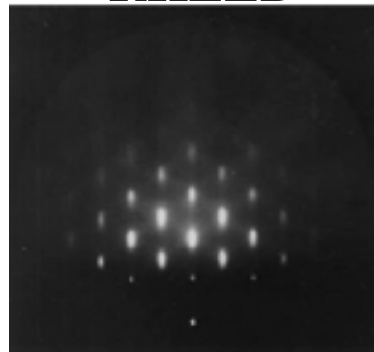


rough surface

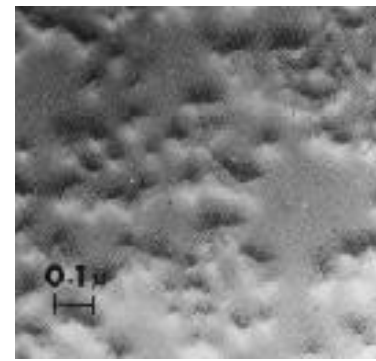
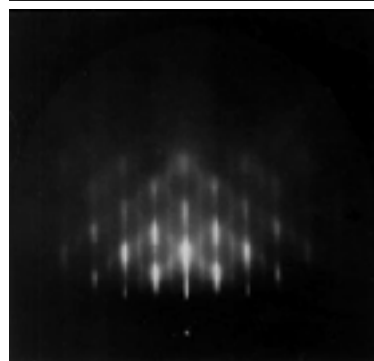
GaAs substrate after oxide desorption

RHEED

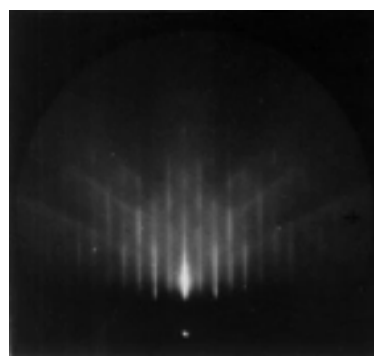
SEM



+ MBE growth of 15 nm GaAs

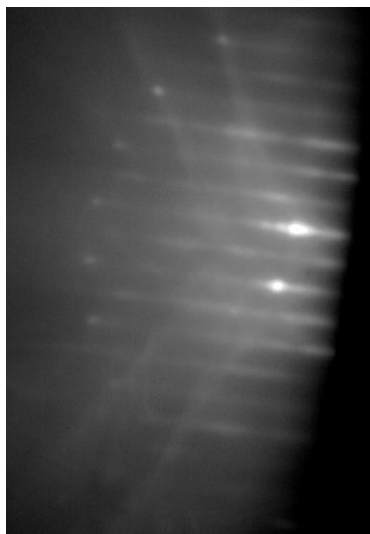


+ MBE growth of 1 μm GaAs

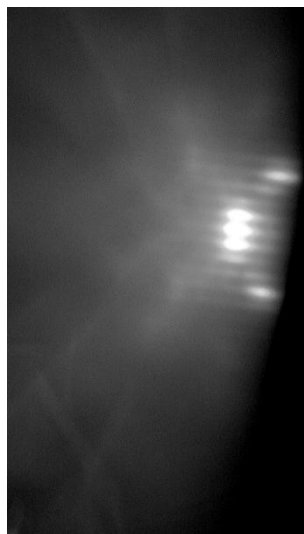


in situ growth monitoring – RHEED – (2x4) GaAs surface reconstruction

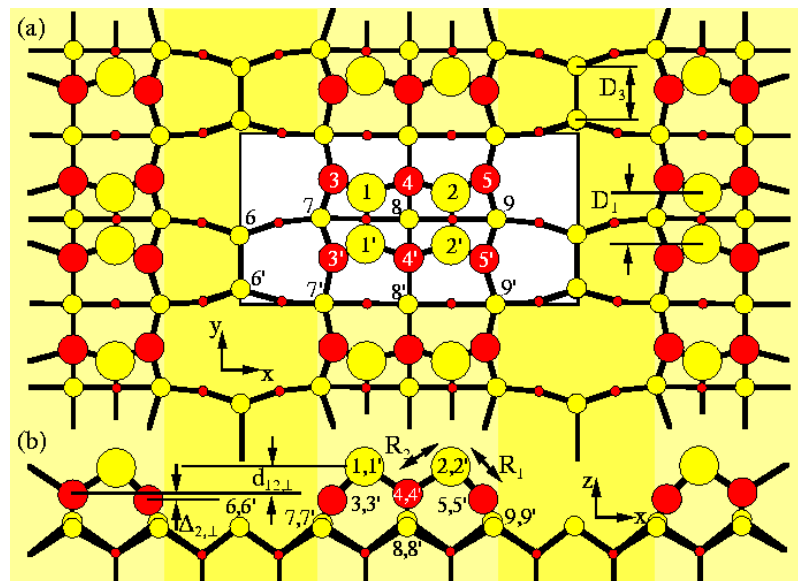
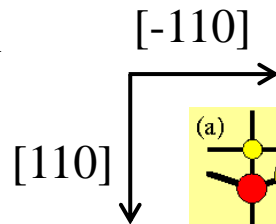
RHEED pattern depends on the azimuth



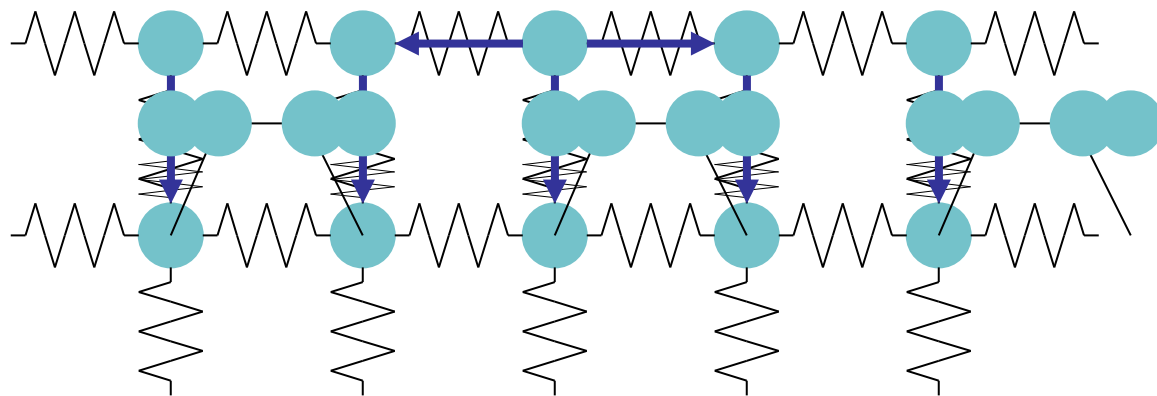
azimuth [110] (2x)



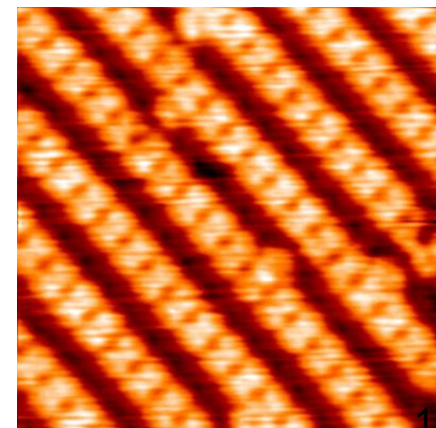
azimuth [-110] (4x)



surface reconstruction – change of periodicity

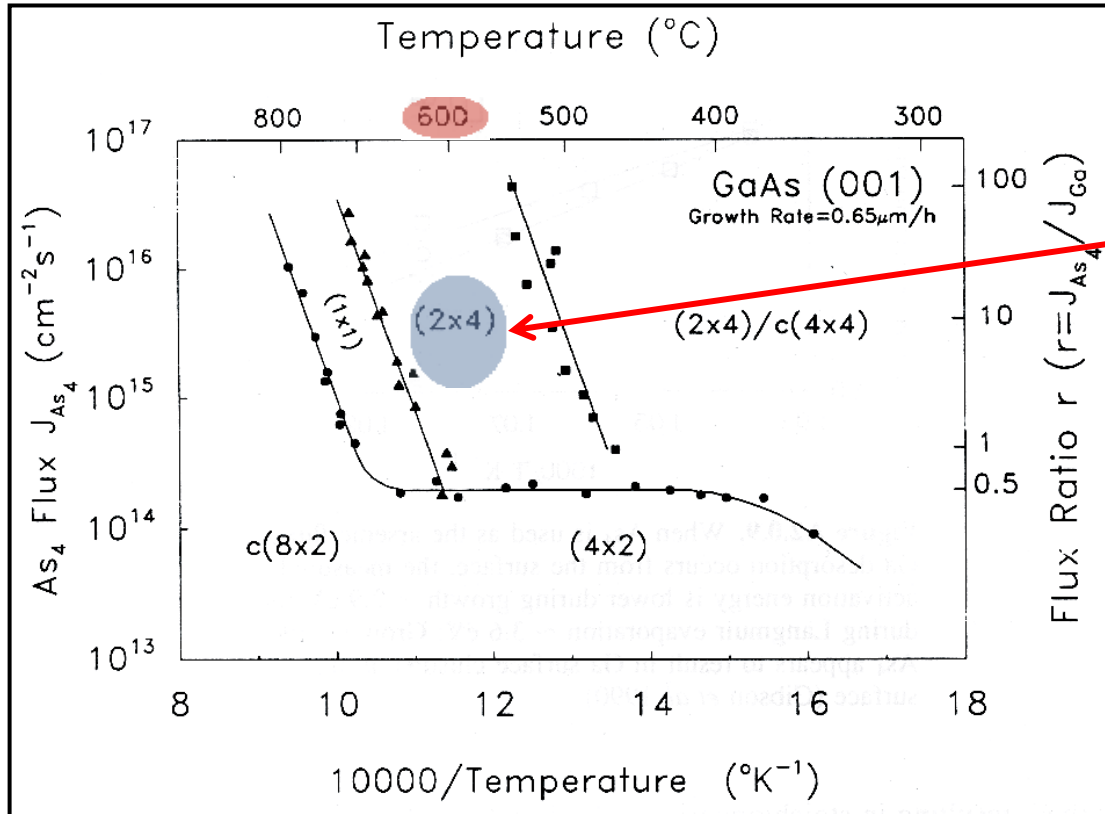


GaAs(001) - STM



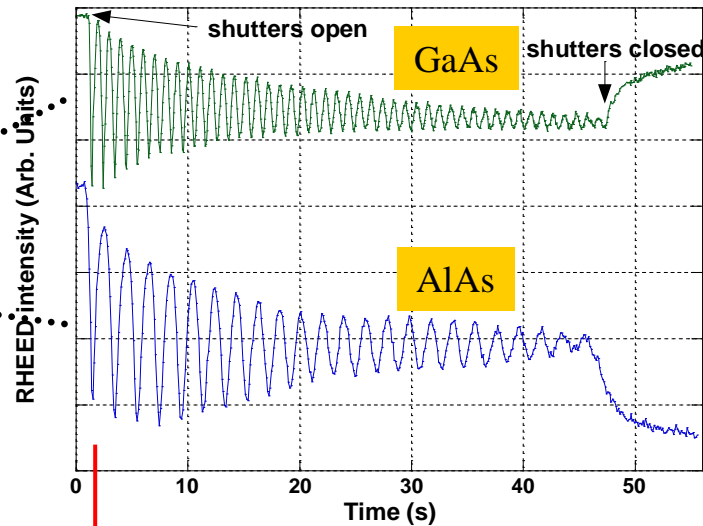
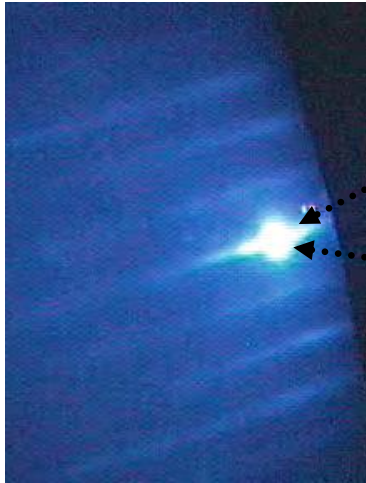
RHEED – surface phase diagram of GaAs

various surface reconstructions depending on T, As/Ga ratio, ...



- **As-stable (2 × 4):** typical GaAs growth conditions by MBE
- surface reconstruction strongly depends on surface temperature – RHEED as surface thermometer

RHEED – growth rate measurements (RHEED oscillations)



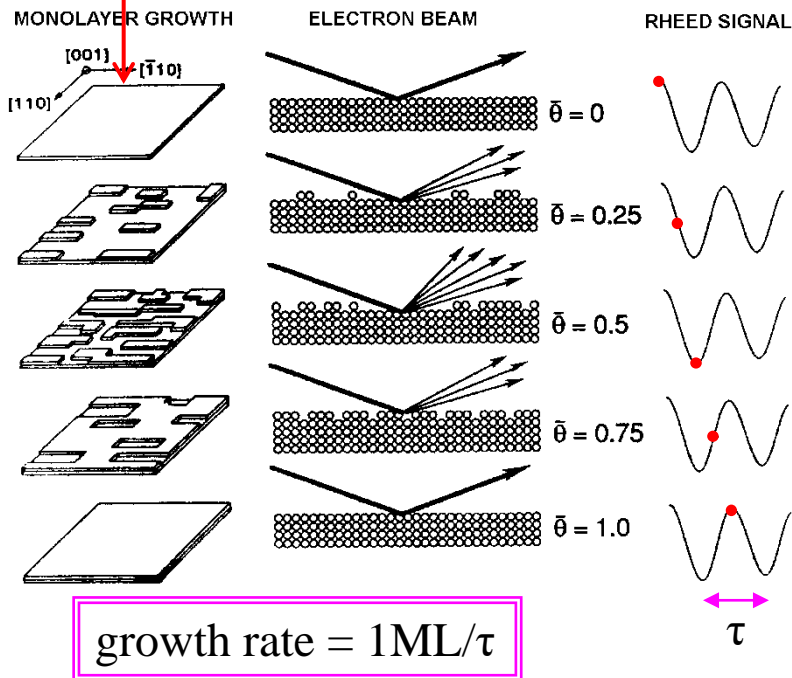
Ga shutter closed:

GaAs: signal recovery \Rightarrow high surface mobility of Ga adatoms, smoothing of the surface

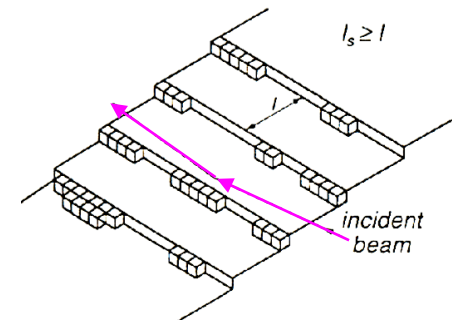
Al shutter closed:

AlAs: signal damping \Rightarrow no surface smoothing; low surface mobility of Al adatoms

start of growth



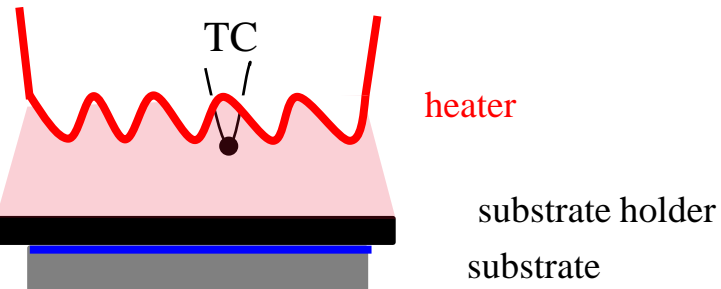
- RHEED oscillations – observation of periodic change of surface roughness as the layer grows
- required: 2D nucleation – layer by layer growth
- no RHEED oscillations for step flow growth



- V element-rich conditions

substrate temperature measurement in MBE

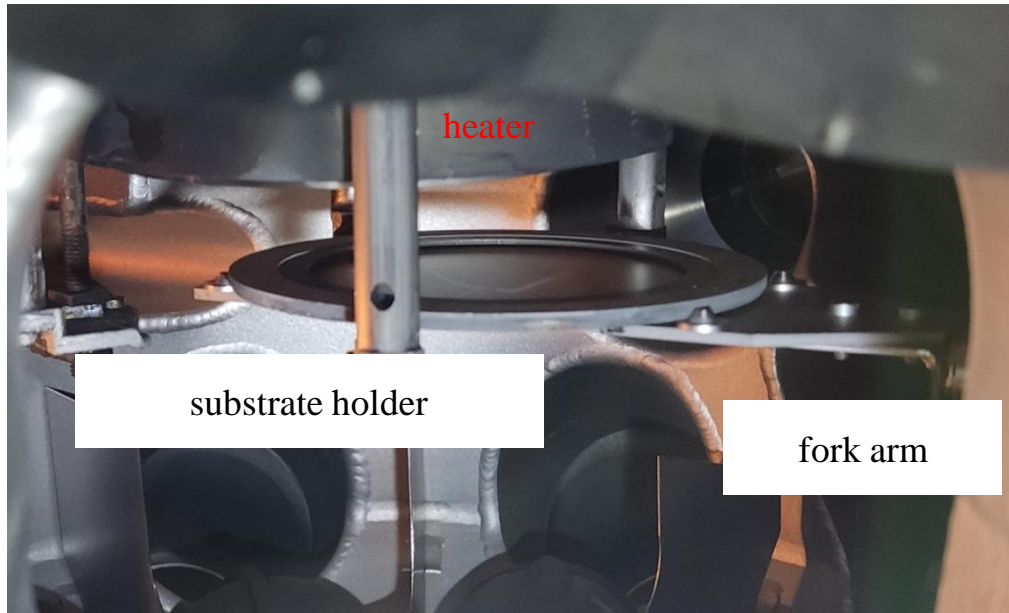
under vacuum heat transport by radiation only



for low T growth substrate glued by In or In/Ga to improve thermal contact of the substrate and the holder

often TC reading used as the measure of the substrate surface T

- TC measures T of the heater !!!
- (TC has no thermal contact with the substrate)
- even if reproducible (?) then for the particular system only
- T values useless for others



Do not do that !!!!

ethical guidelines in science require that your data are presented in the way allowing others to repeat (verify) your experiment !!!!

substrate temperature measurement in MBE

optical pyrometry in IR $\lambda = 1 - 3 \mu\text{m}$ measurement of sample T based on the black body emission spectrum

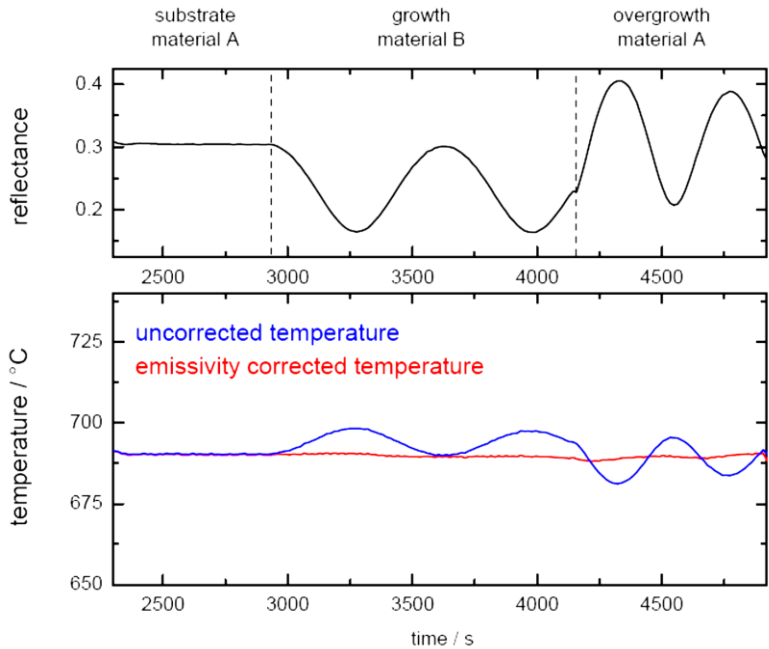
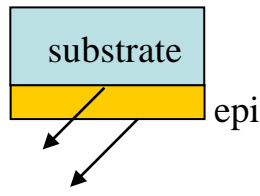
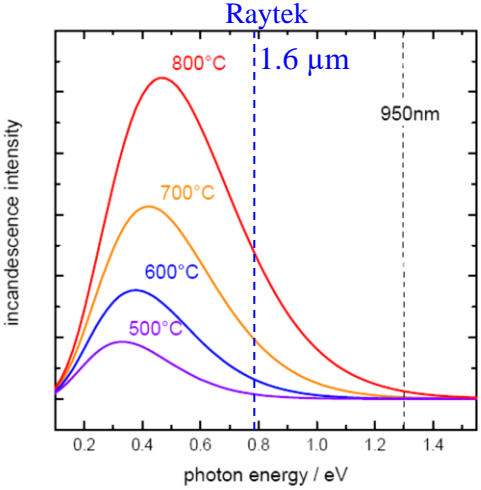


Fig. 1: Emission of a black body according to Planck's law: the emitted radiation power increases together with the temperature, the maximum of the curves shifts to higher photon energies.

IR light interference leads to oscillations of the signal; thus to artificial oscillations of T reading

emissivity corrected pyrometry – simultaneous measurement of pyrometry signal and light reflection; allows to eliminate interference-induced oscillations

- still the problem exists if the viewport transmission changes during a long growth campaign

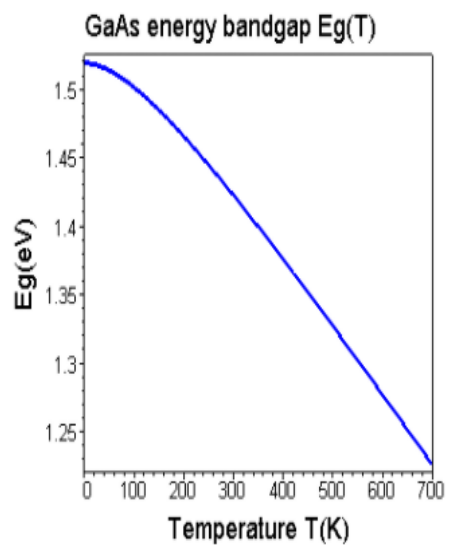
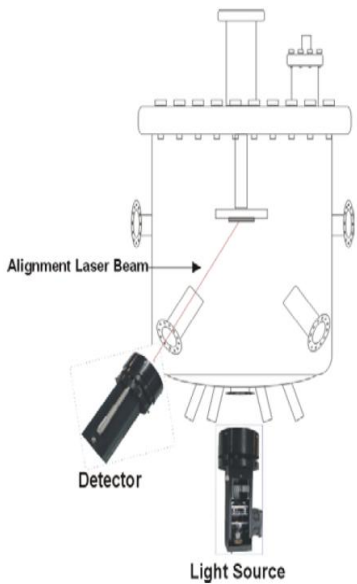


- T of which part pyrometry measures if the substrate is transparent (e.g. sapphire, GaN, ZnO,) ?

T of the substrate holder !!!!!

it may differ from the substrate surface T by tens of °C

substrate temperature measurement in MBE

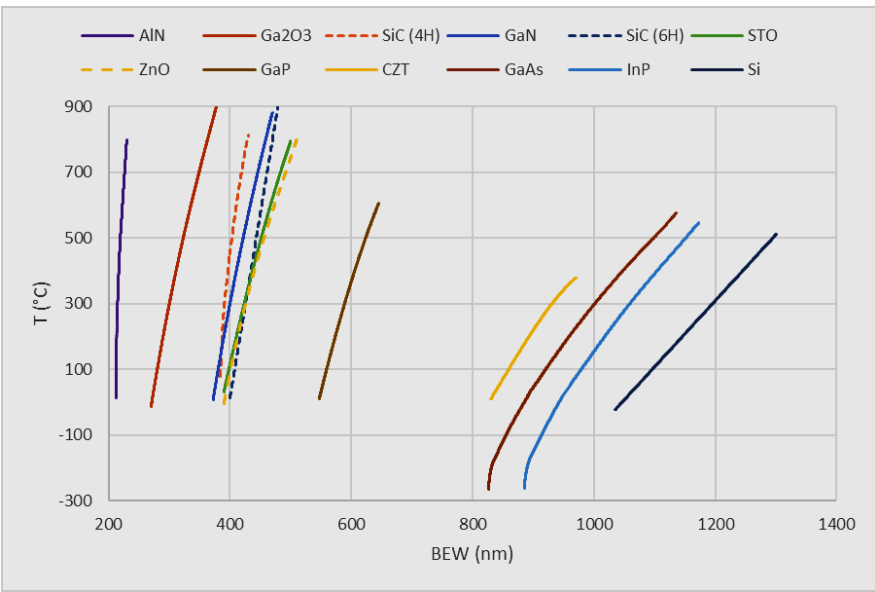


BandiT – k-Space Inc.

- measurement of the **substrate optical absorption edge (not the light intensity)**, the absolute temperature of the wafer can be determined. This absorption edge, which is directly proportional to the band gap of the material, is temperature dependent.

$$E_g(T) = 1.519 - 5.408 \cdot 10^{-4} T^2 / (T + 204)$$

J. S. Blakemore J. Appl. Phys. 53 (1982)



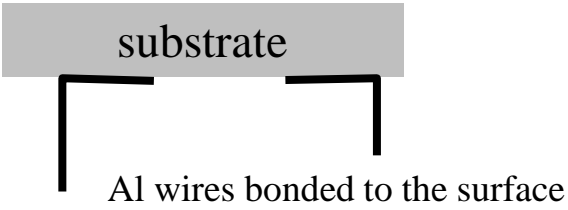
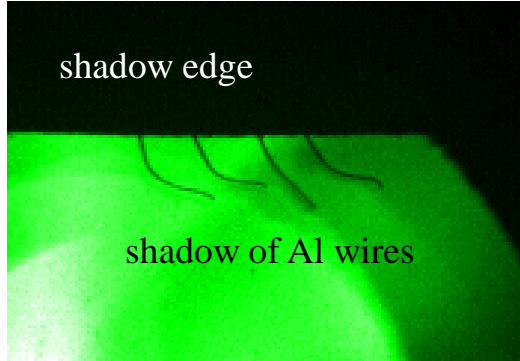
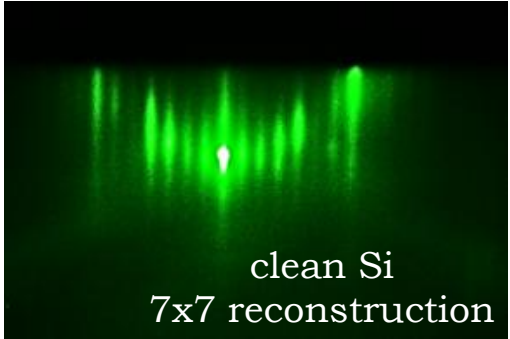
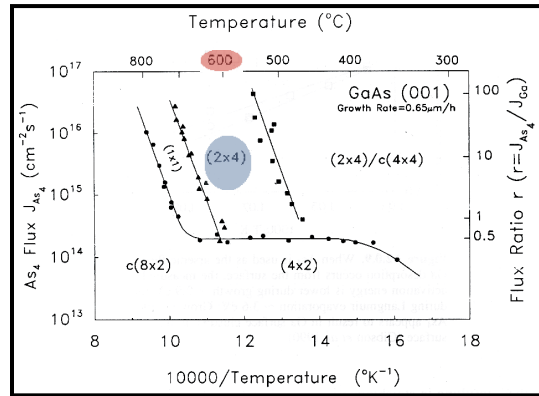
- immune to changing viewport transmission, stray light, and signal contribution from substrate or source heaters, all sources of measurement error for pyrometers
- requires that your substrates are carefully calibrated (factory calibration files)
- different substrate doping can change the result
- does it measure substrate *surface* T ?

substrate temperature measurement in MBE

rely on surface phenomena if possible

some examples:

1. oxide desorption from GaAs (RHEED observation) $\sim 560^\circ\text{C}$
2. change of surface reconstruction (RHEED pattern) of GaAs
3. change of surface reconstruction of Si(111) $7\times 7 \rightarrow 1\times 1$ @ 830°C
4. melting point of suitable metal (Al 660.3°C)
5. desorption rate of deposited metal (e.g. Ga from GaN surface)
6. others, depending on your material system



Try to calibrate surface T as precisely as possible. For you the run to run T reproducibility is the most important. In some cases differences in T readings between various MBE systems as large as $\pm 50^\circ\text{C}$ are standard. It is crucial to describe in details the way of substrate T measurement in your experiment !!!

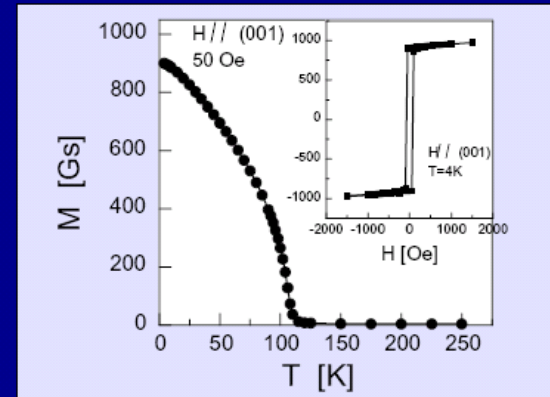
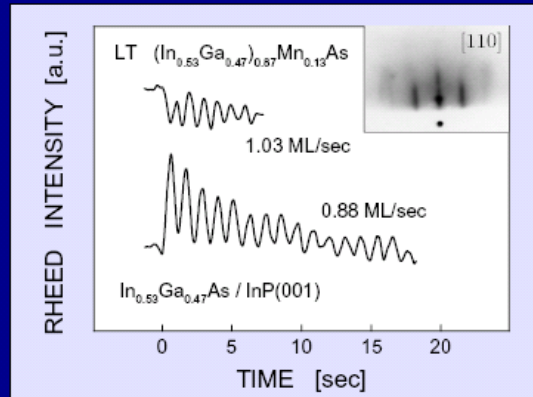
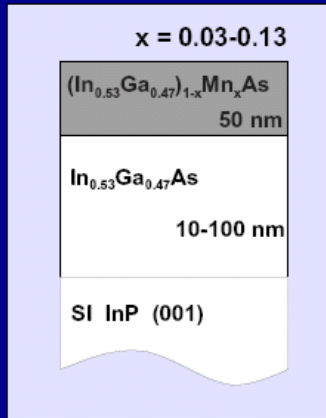
Application of MBE: incorporation of Mn above the solubility limit in III-V's

nonequilibrium growth in MBE \Rightarrow possibility to grow (Ga, In)As layers with high concentration of randomly distributed Mn

FERROMAGNETIC $(\text{In}_{0.53}\text{Ga}_{0.47})_{1-x}\text{Mn}_x\text{As}$ GROWN ON InP(001)



CURIE TEMPERATURE $T_c \sim 100\text{-}110\text{ K}$

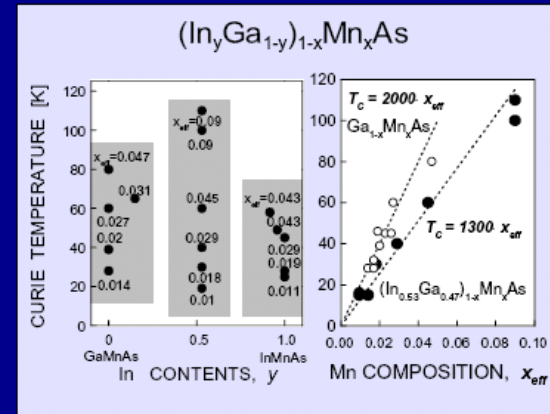
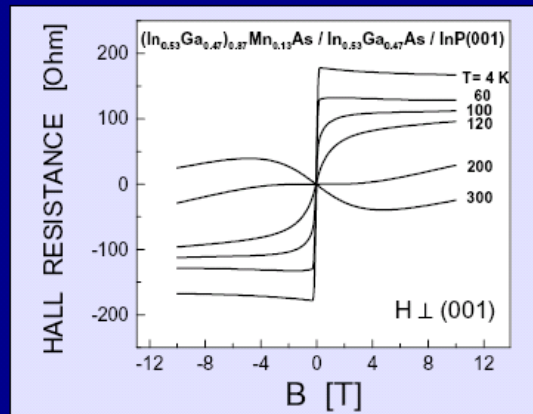


near-lattice matched to InP (001) substrate

$T_s = 200\text{-}260\text{ }^\circ\text{C}$

- optimized vs. Mn

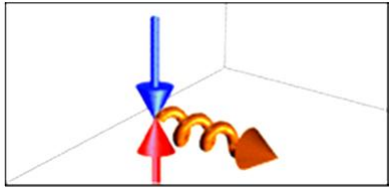
easy axis in-plane



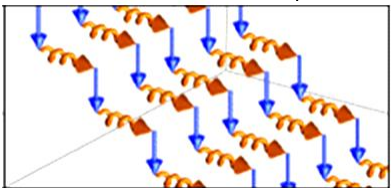
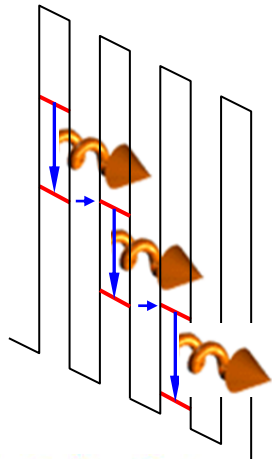
Application of MBE: superlattices in optical devices

cascade IR laser GaAs/AlGaAs (~9μm)

conventional laser



cascade laser



„cascade of electrons”

multiple photon emission from one electron

www.bell-labs.com/org/physicalsciences/projects/qcl/qcl2.html

GaAs:Si	1,0μm	$n = 6,0 \cdot 10^{18}$
GaAs:Si	3,5μm	$n = 4,0 \cdot 10^{16}$

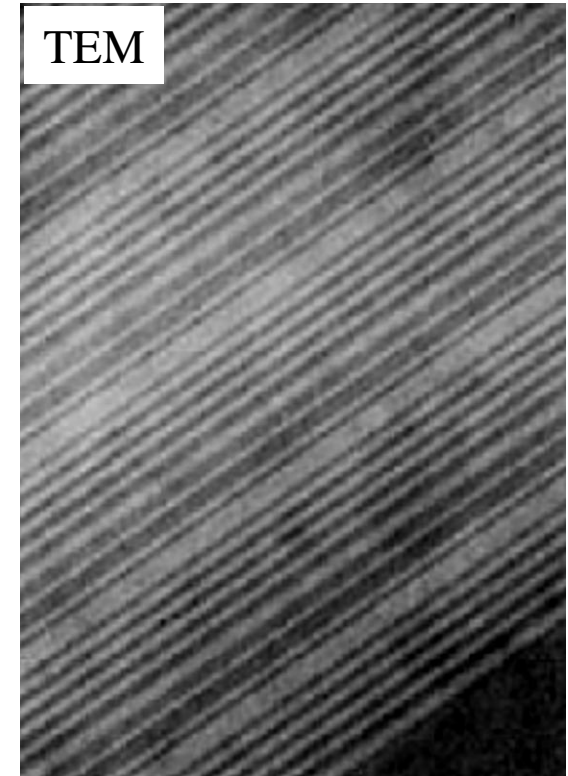
Active region
1,62μm
(45nm x 36)

AR = 36 x this segment:

GaAs:Si	3,5μm	$n = 4,0 \cdot 10^{16}$
GaAs:Si	1,0μm	$n = 6,0 \cdot 10^{18}$
GaAs:Si	1,0μm	$n = 2,0 \cdot 10^{18}$
GaAs:Si	450μm	$n = 2,0 \cdot 10^{18}$

- GaAs
- AlGaAs x(Al)=45%
- Active Region (AR)
- substrat

2,8 -underline means Si doped



ITE Warszawa - Kosiel et al. EuroMBE 2009, Zakopane

MBE allows growth of complicated set of superthin epilayers with very high crystallographic quality

Application of MBE: modulation doping (δ -doping)

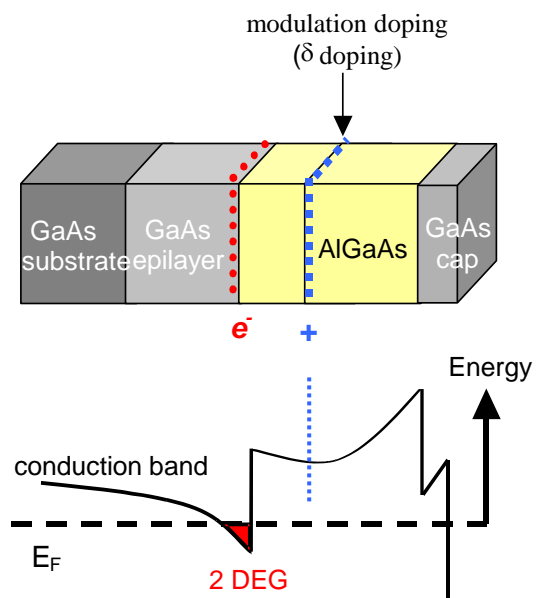
problem: doping required for high electrical conductivity

BUT

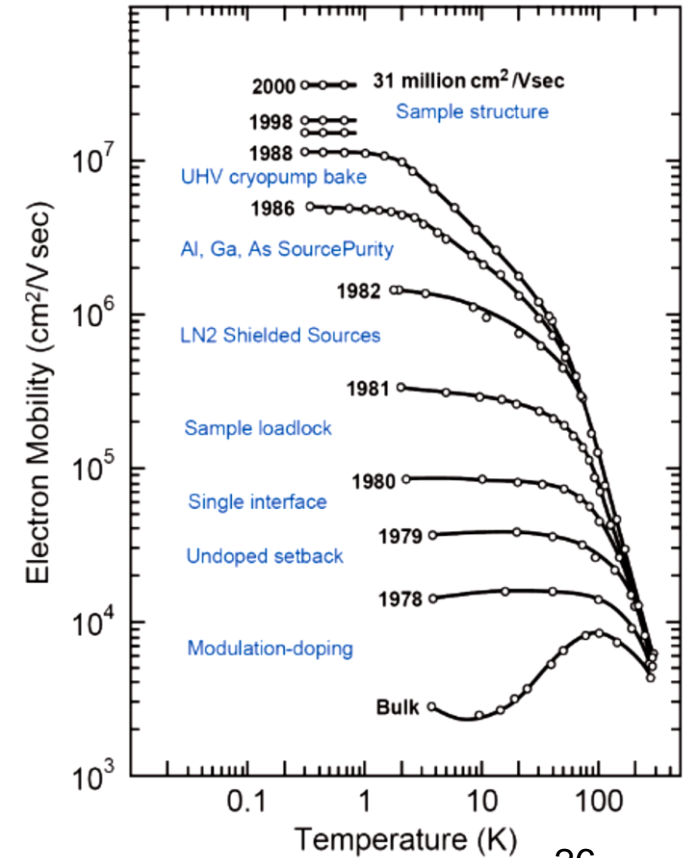
impurities scatter charge carriers \Rightarrow lower mobility at low T

solution: separate in space source of charge carriers (dopants) from the electrical conductivity channel

70ties, Art Gossard i Horst Störmer z Bell Labs.



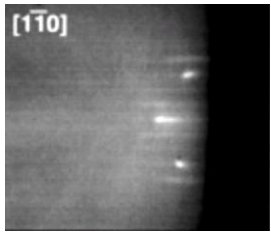
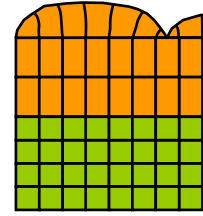
transfer of charge carriers into the 2D channel and their separation from dopants \Rightarrow higher μ



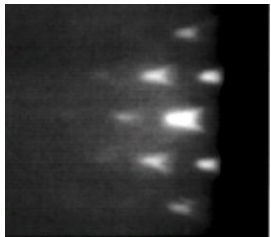
Application of MBE: self-organized quantum dots (QD)

InAs/(001) GaAs
azimuth [1-10]

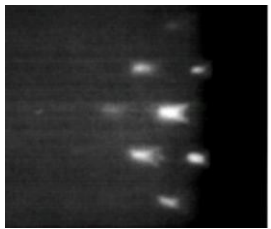
lecture 17.02.2025 - surface deformation as the way to relax lattice mismatch strain
7% lattice mismatch in the InAs/GaAs system



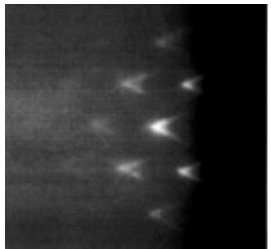
1 ML InAs



2 MLs InAs



3 MLs InAs



30 MLs InAs

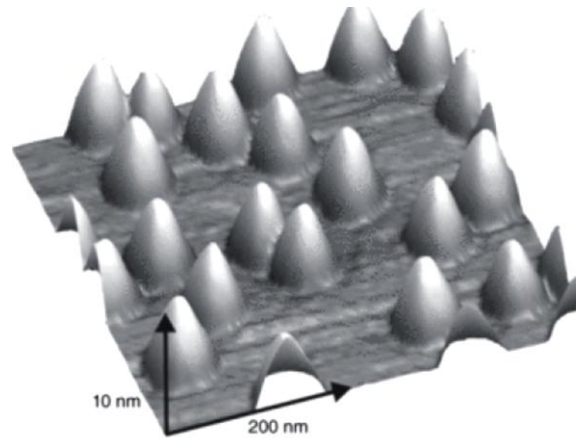
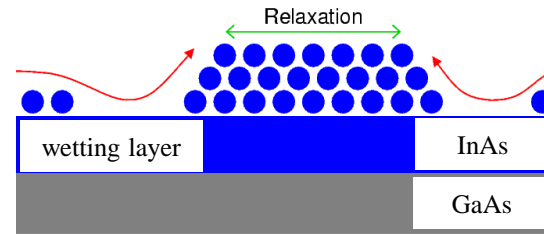
growth 3D

growth modes:

Frank-van der Merwe (layer-by-layer)

Stranski-Krastanov (layer + island)

Volmer-Weber (island)



InAs QDs on GaAs:

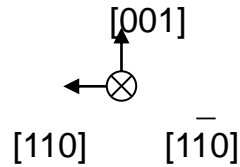
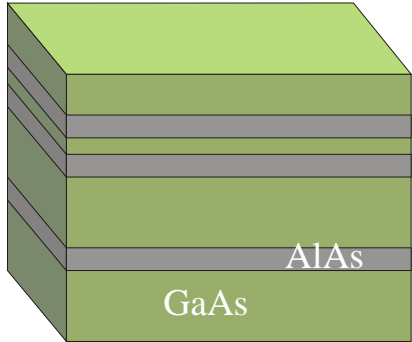
- dislocation free
- diameter ~20nm
- height a few nm
- broad distribution of dimensions
- random positions on the substrate

(self-organization)

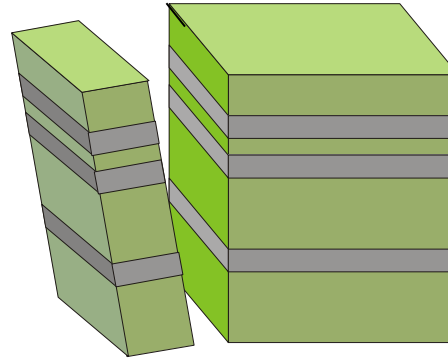
Application of MBE: organized quantum dots (QD)

E. Uccelli et al. EuroMBE 2009, Zakopane

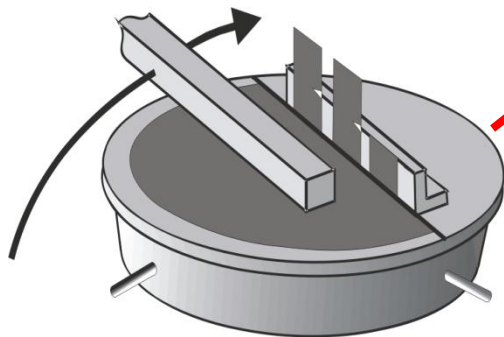
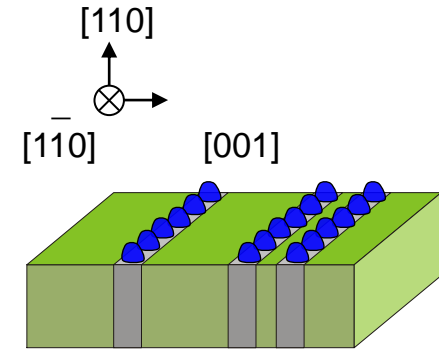
1. growth of AlAs/GaAs (001) layers



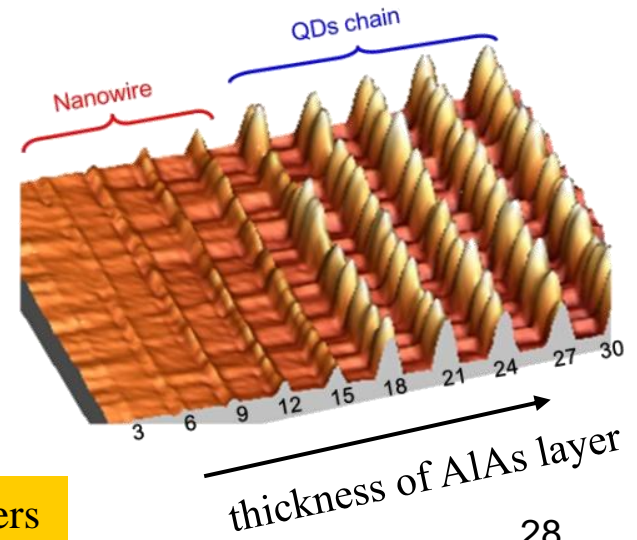
2. *In situ* cleavage: (110) flat surface



3. growth of InAs on the cleaved (110) surface



shutter blade cuts the wafer

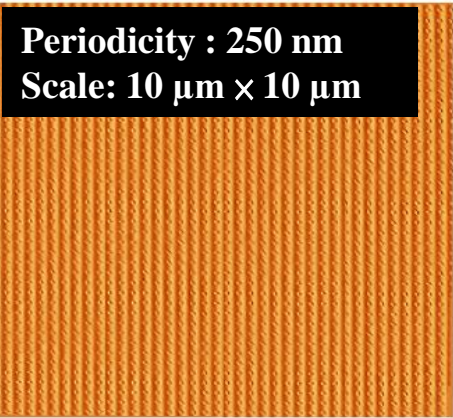
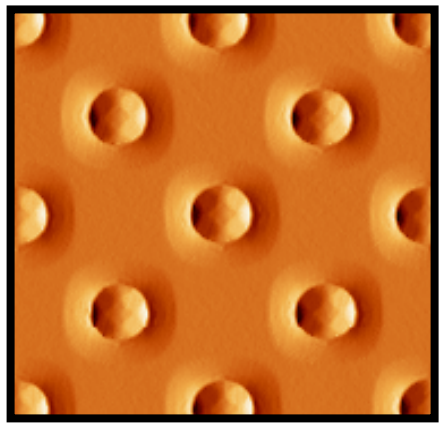


- in-plane nanowires for thin AlAs layers
- QDs for thicker AlAs layers

Application of MBE: organized quantum dots (QD)

- advantages of ordering:
- better uniformity of QDs' dimensions (more uniform light emission λ)
 - single QD can be addressed
 - ...

G. Chen (EuroMBE 2009, Zakopane)
E-beam lithography + RIE



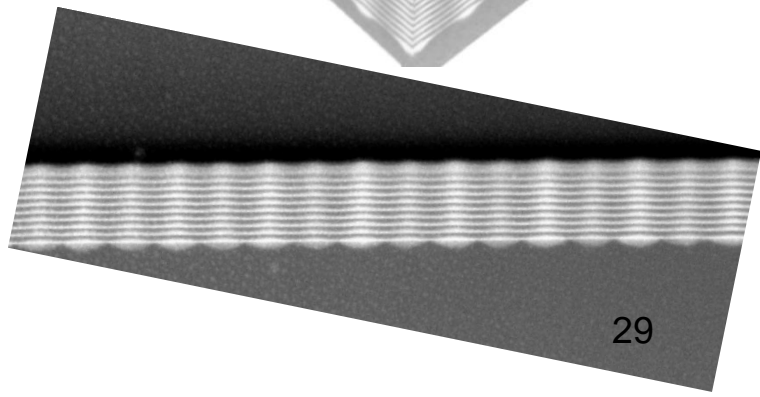
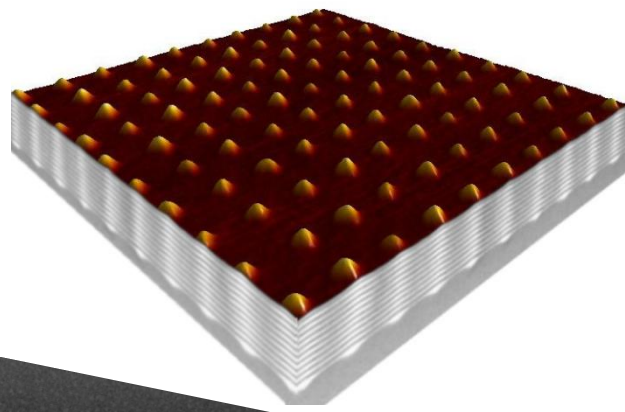
Periodicity : 250 nm
Scale: 10 μm \times 10 μm

Ge QDs on Si substrate

- lithography of the substrate (e-beam or X-Ray)
 - etching of the pattern (RIE)
 - MBE growth of QDs
-
- positions of QDs in the next layer reflects their distribution in the layer underneath (coupling via the strain field)

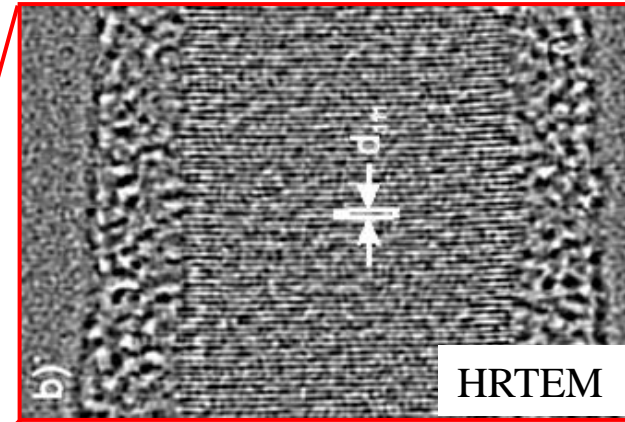
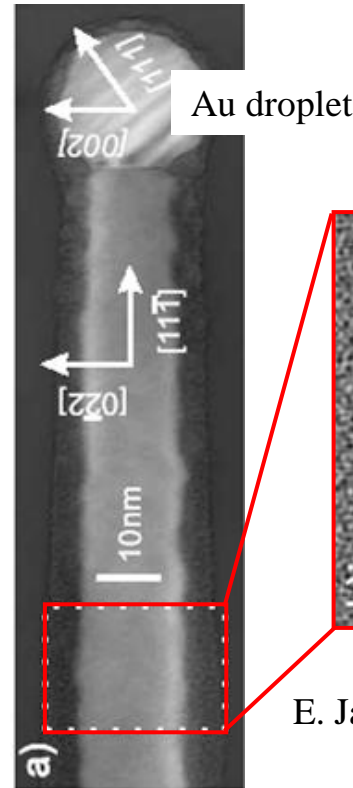
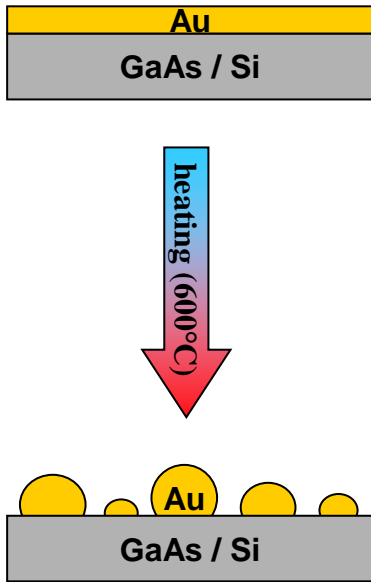
G. Mussler (EuroMBE 2009, Zakopane)
X-ray lithography + RIE

crystal of Ge QDs



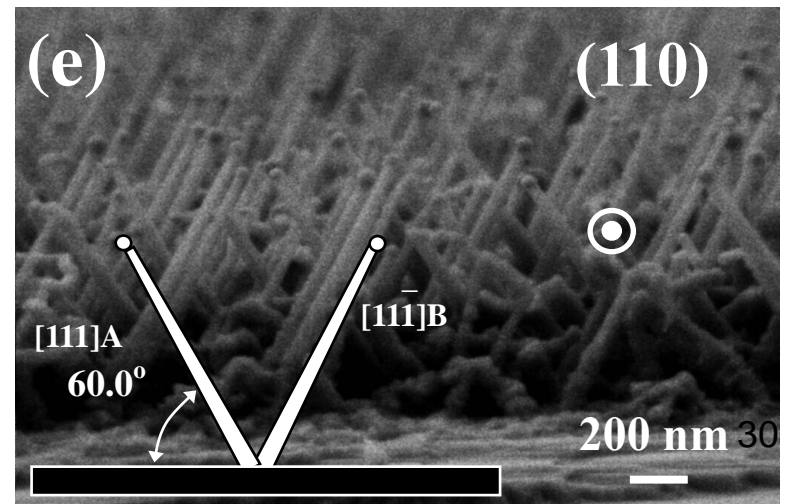
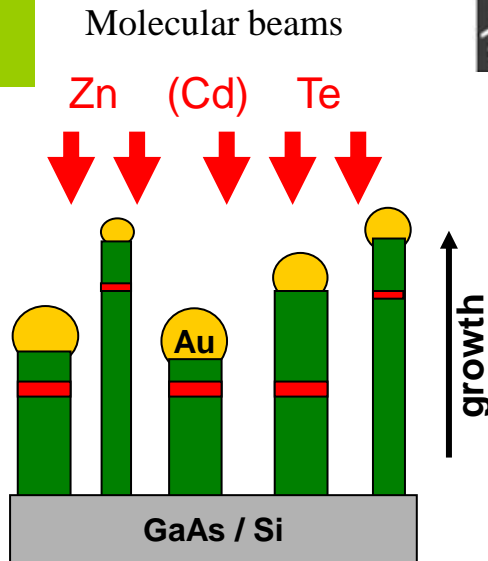
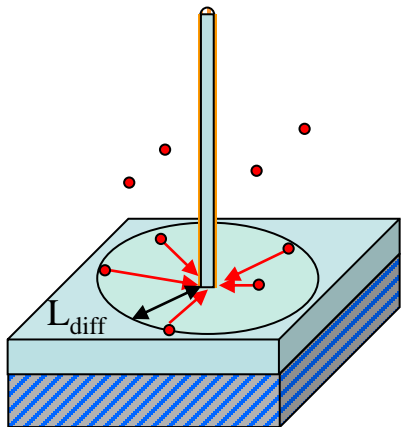
Application of MBE: self-organized nanowires (NWs)

ZnTe NWs on GaAs



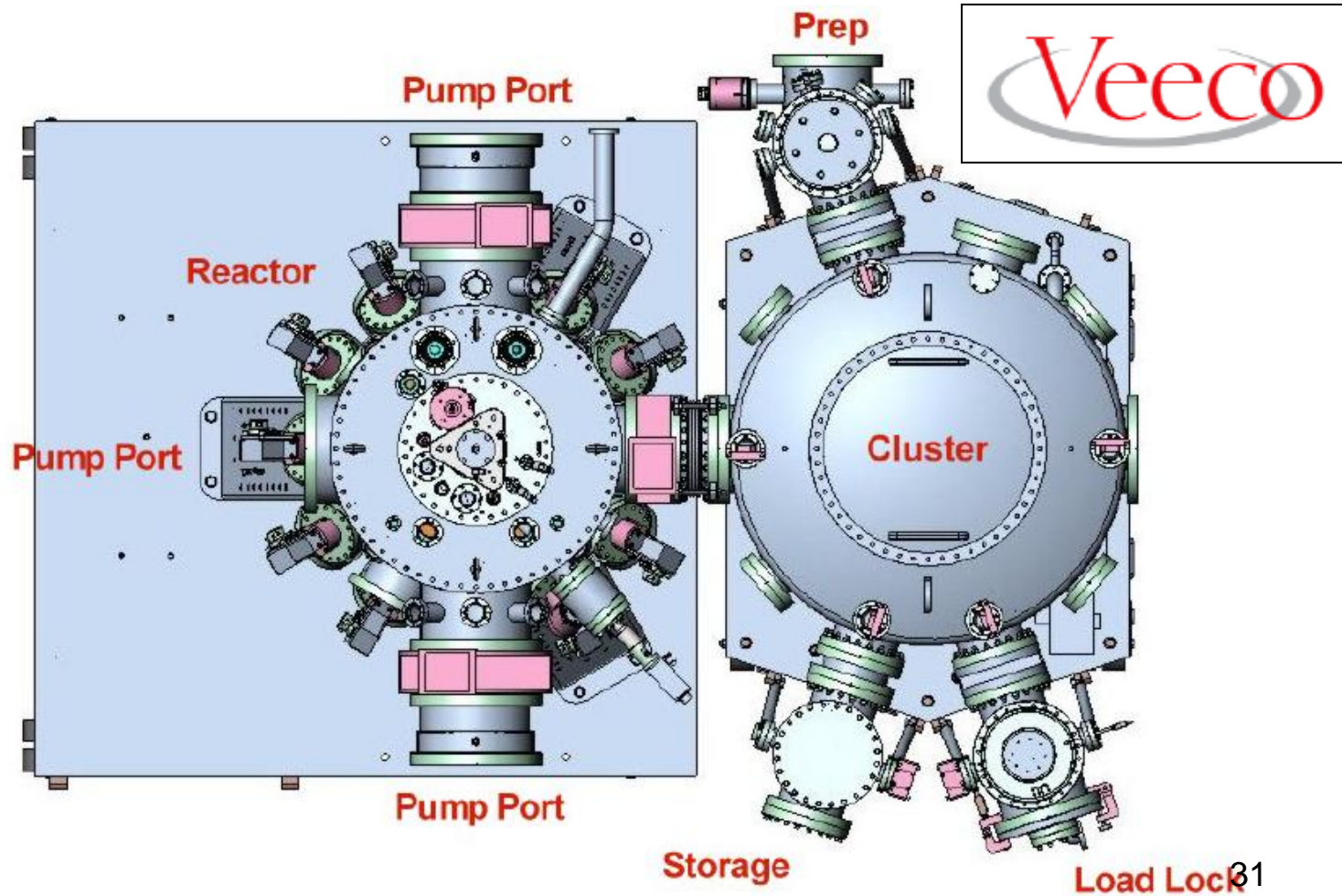
E. Janik, *et al.* APL **89**, 133114 (2006)

growth mode:
vapor – liquid – solid VLS



New generation of MBE systems - clusters

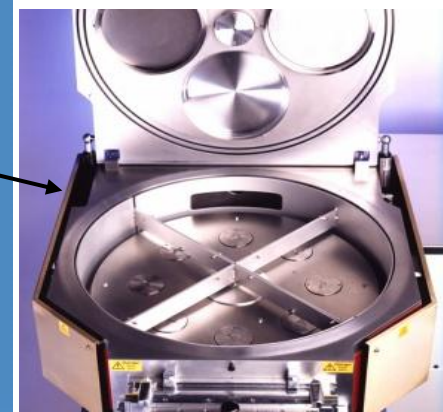
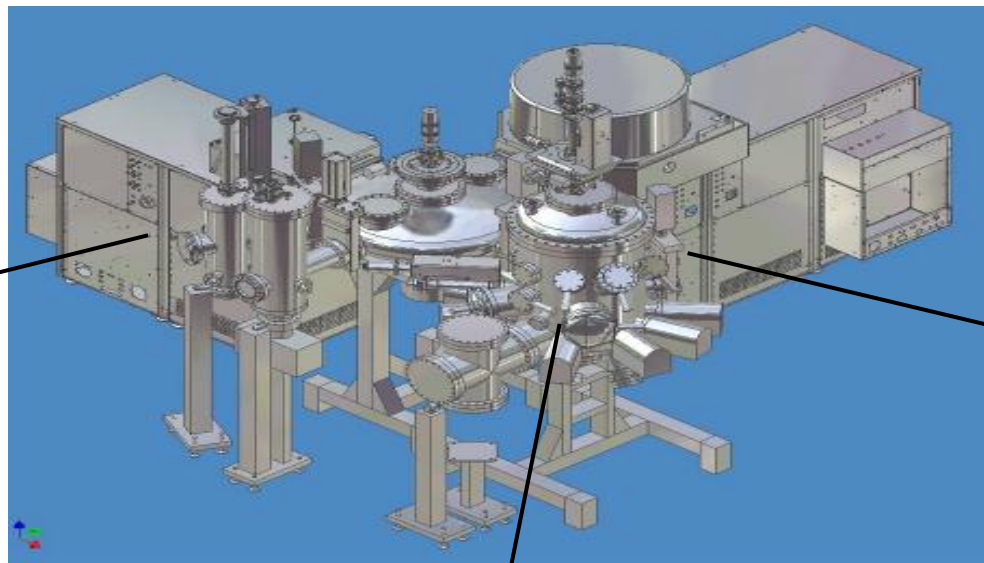
- simultaneous growth on large substrates; multi-wafer substrate holders; to increase yield
- min 12 source ports + extra ports for surface analysis tools
- cluster design – independent growth and surface preparation (plasma etching, metallization, ...) chambers
- additional chambers for surface analysis (STM, ...); sample transfer under UHV



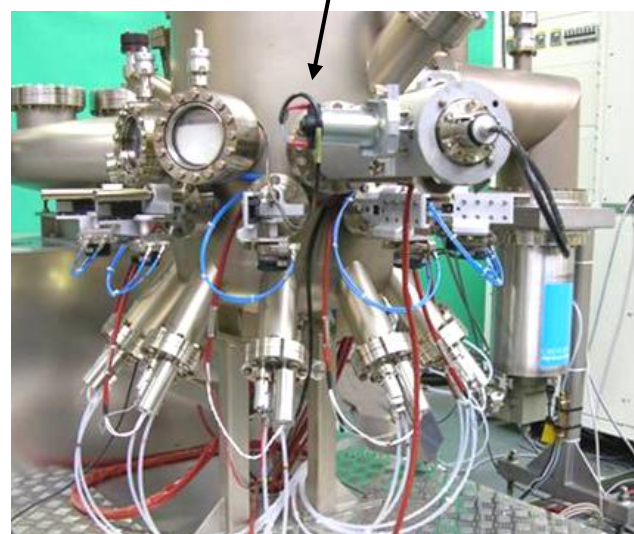
New generation of MBE systems - clusters



Etch Module (ICP)
for Clusterlab 600



Deposition Module
(RF Magnetron Sputter)
for Clusterlab 600



Epitaxial Growth Module
(MBE V60) for
Clusterlab600

Summary

advantages of MBE:

- high purity of epilayers
- precise growth control
- perfect for fabrication of low-dimensional structures; sharp interfaces
- wide range of *in situ* growth monitoring tools; important for R&D
- large variety of materials/compounds that could be grown
- strongly nonequilibrium growth technique; solubility limit can be overcome

disadvantages of MBE:

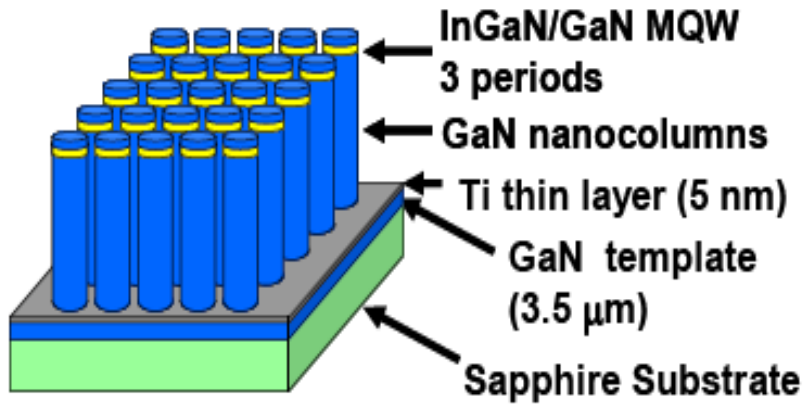
- measurement of REAL substrate surface temperature difficult
- high costs (purchase, installation and every-day use)
- high failure rate (typical for complicated UHV systems)
 - Most Broken Equipment
 - Multi Bucks Evaporator
- selective area growth (SAG) difficult

For further reading

- 1) *M.A. Herman, H. Sitter "Molecular Beam Epitaxy, Fundamentals and Current Status", Springer, 1996*
- 2) *ed. A. Cho "Molecular Beam Epitaxy", AIP, 1994*
- 3) *review papers by T. Foxon; B.A. Joyce; etc.*

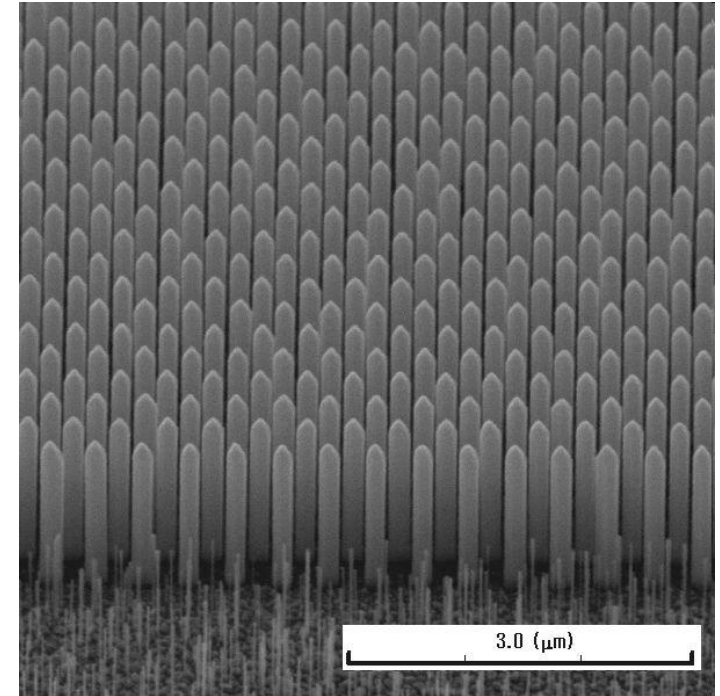
Przykładowe wykorzystanie MBE: uporządkowane NWs (białe nanoLEDs)

H. Sekiguchi et al., IWNS 2008 Montreux, Switzerland

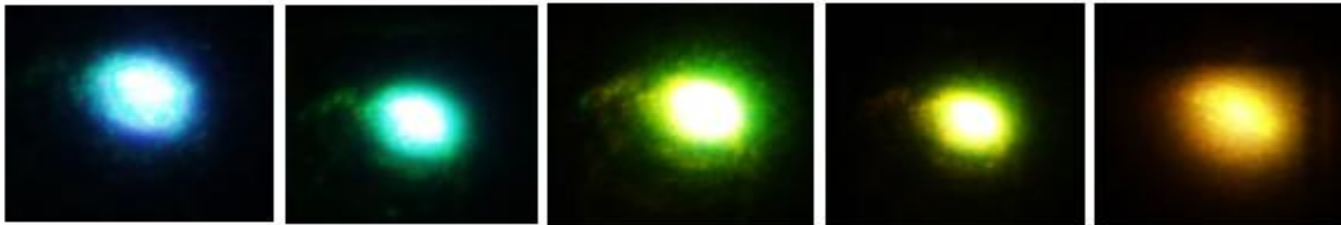


nanodziurki o różnych średnicach w masce Ti

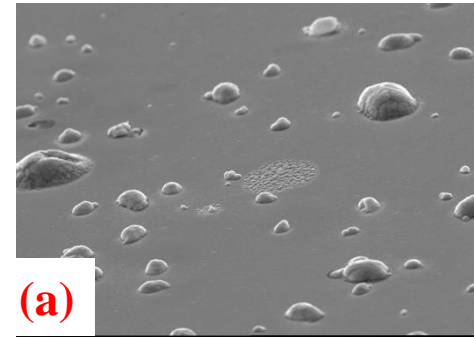
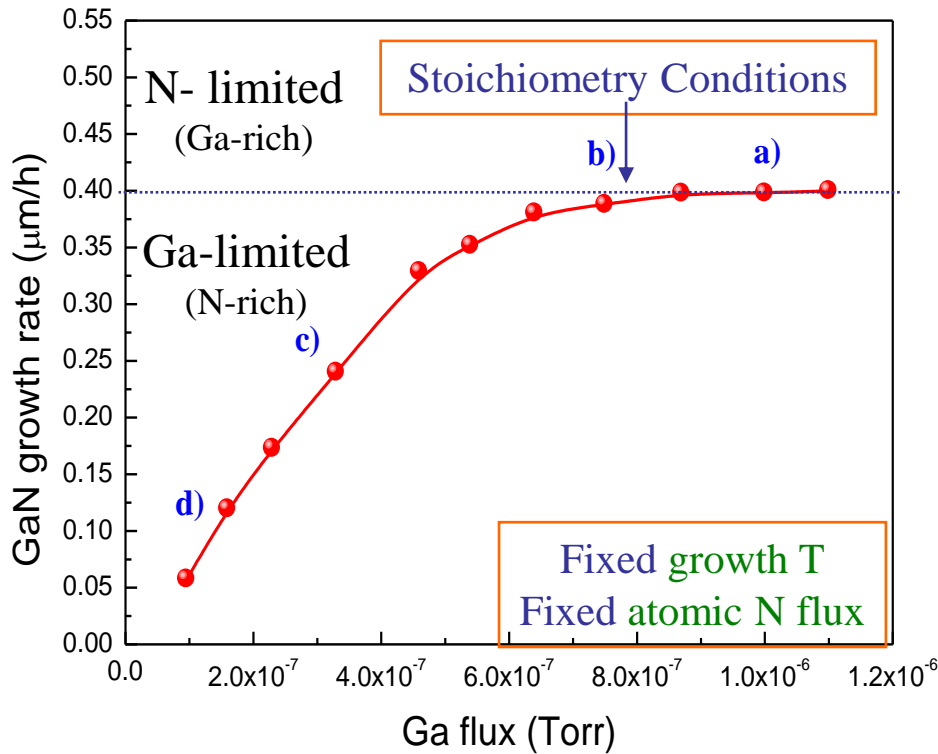
nanodziurki porządkują położenie kolumn



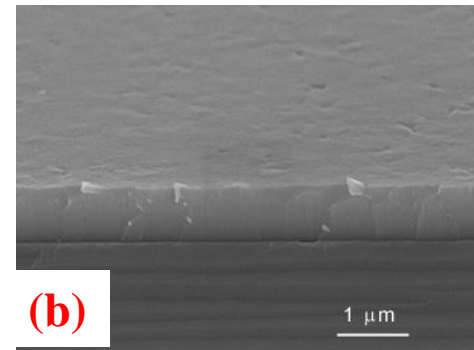
średnica nanodziurki \Rightarrow średnica nanokolumny \Rightarrow długość fali emitowanego światła



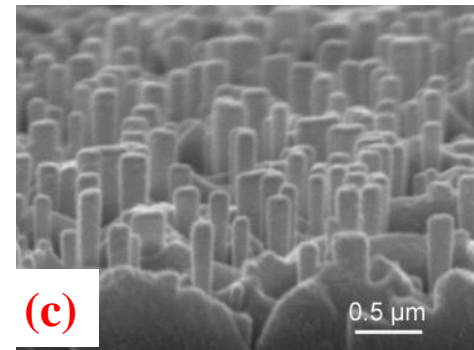
emisja z nanokolumn InGaN/GaN wzrastających na tej samej płytce z różnym wzorem nanodziurek w masce Ti



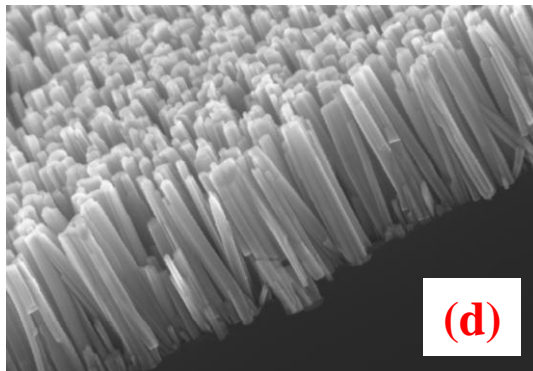
III/V > 1



III/V ≈ 1



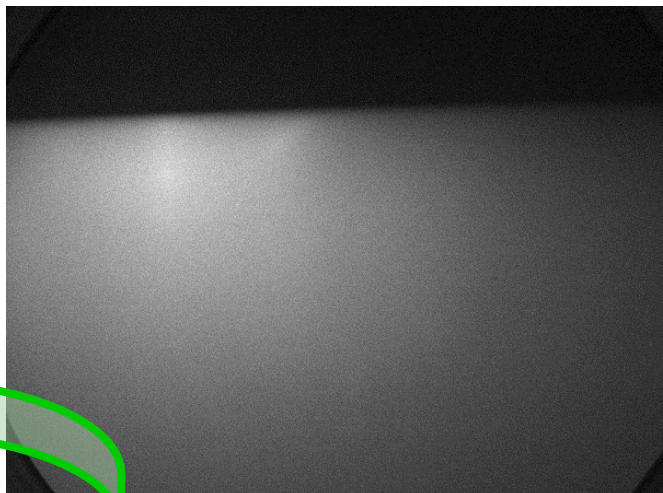
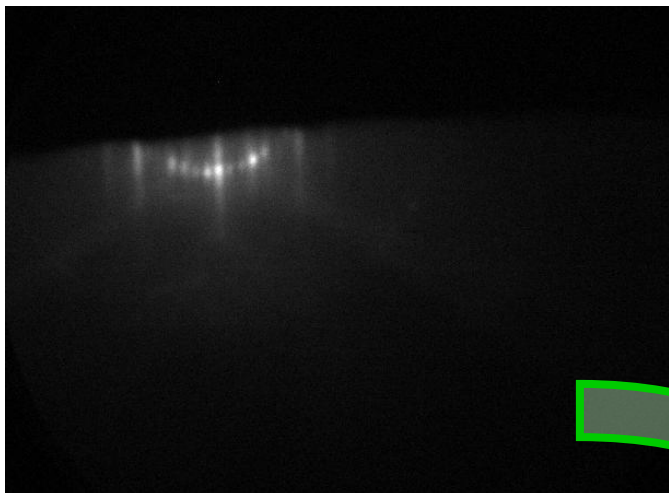
III/V < 1



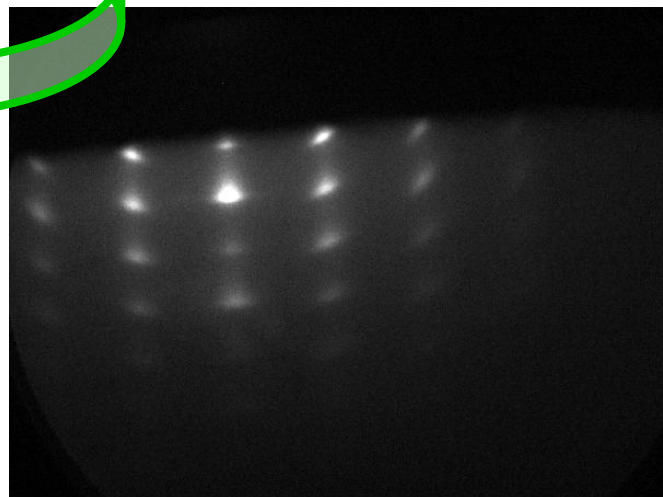
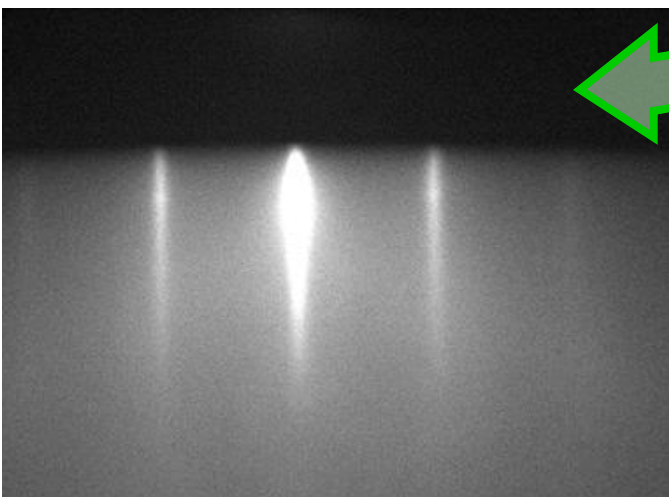
zmieniając stosunek III/V zmieniamy mod wzrostu



clean Si
(7 × 7)

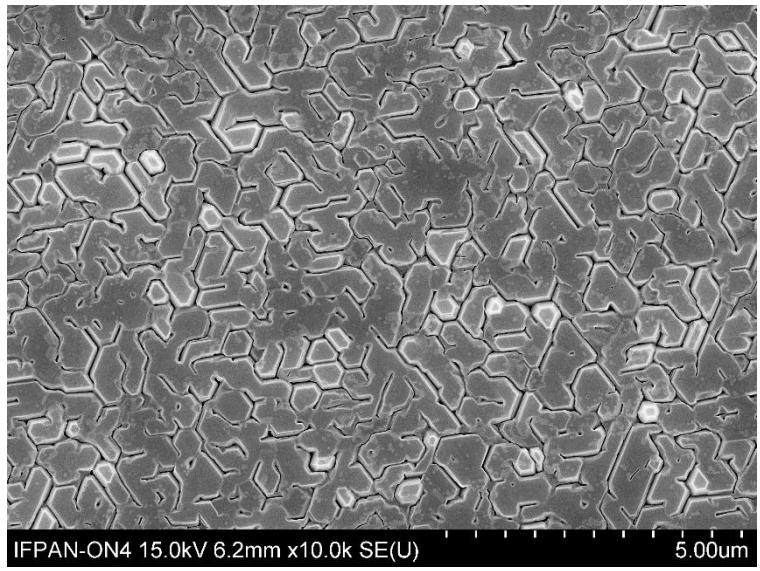
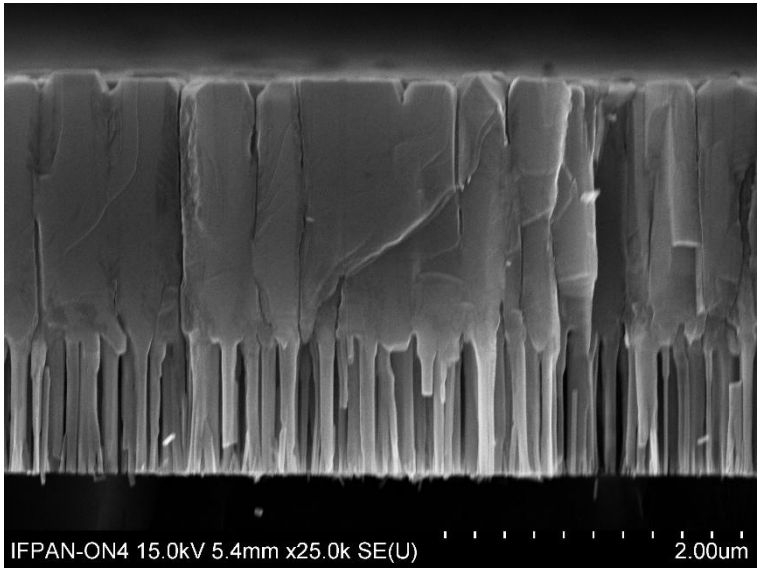


$T_{Ga} \uparrow$
 $T_{sub} \downarrow$
(Ga-rich)



GaN NWs
growth
 $T_{Ga} = 800C$
 $T_{sub} = 740C$

planaryzacja powierzchni



ważne:

- ✓ dobra jednorodność długości drutów
- ✓ wysoka koherencja twistu NWs
- ✓ potrzebna wysoka lateralna prędkość wzrostu (Mg doping?)
- ✓ łatwiejsze zarastanie w MOVPE lub HVPE