

# Crystal Growth: Physics, Technology and Modeling

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## Lecture 4. Molecular beam epitaxy

15 March 2023

<http://www.unipress.waw.pl/~stach/cg-2022-23>

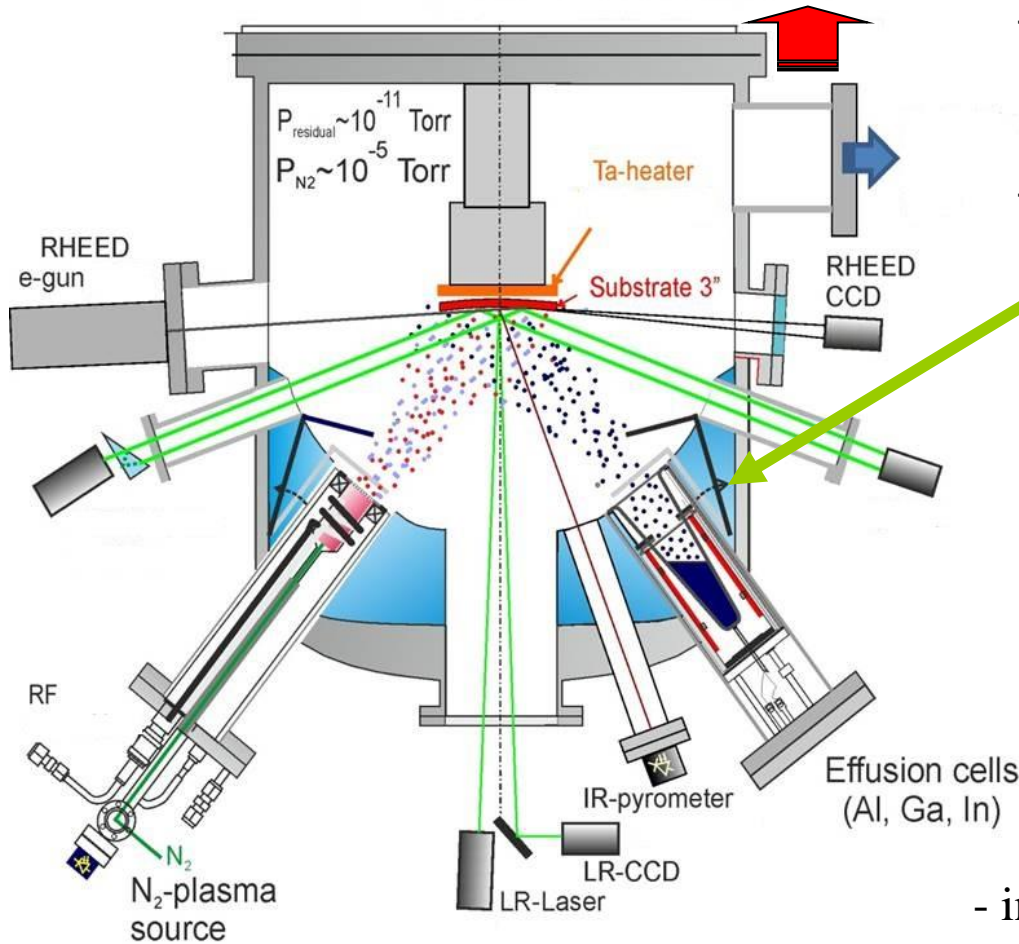
# **Molecular beam epitaxy - MBE**

pol. epitaksja metodą (z) wiązek molekularnych

## **Outline:**

- **idea and physical background of MBE**
- **technical aspect of MBE growth**
- ***in situ* monitoring of MBE growth**
- **examples of MBE grown structures**
  - **low-temperature growth**
  - **superlattices**
  - **quantum dots and nanowires**
- **summary**

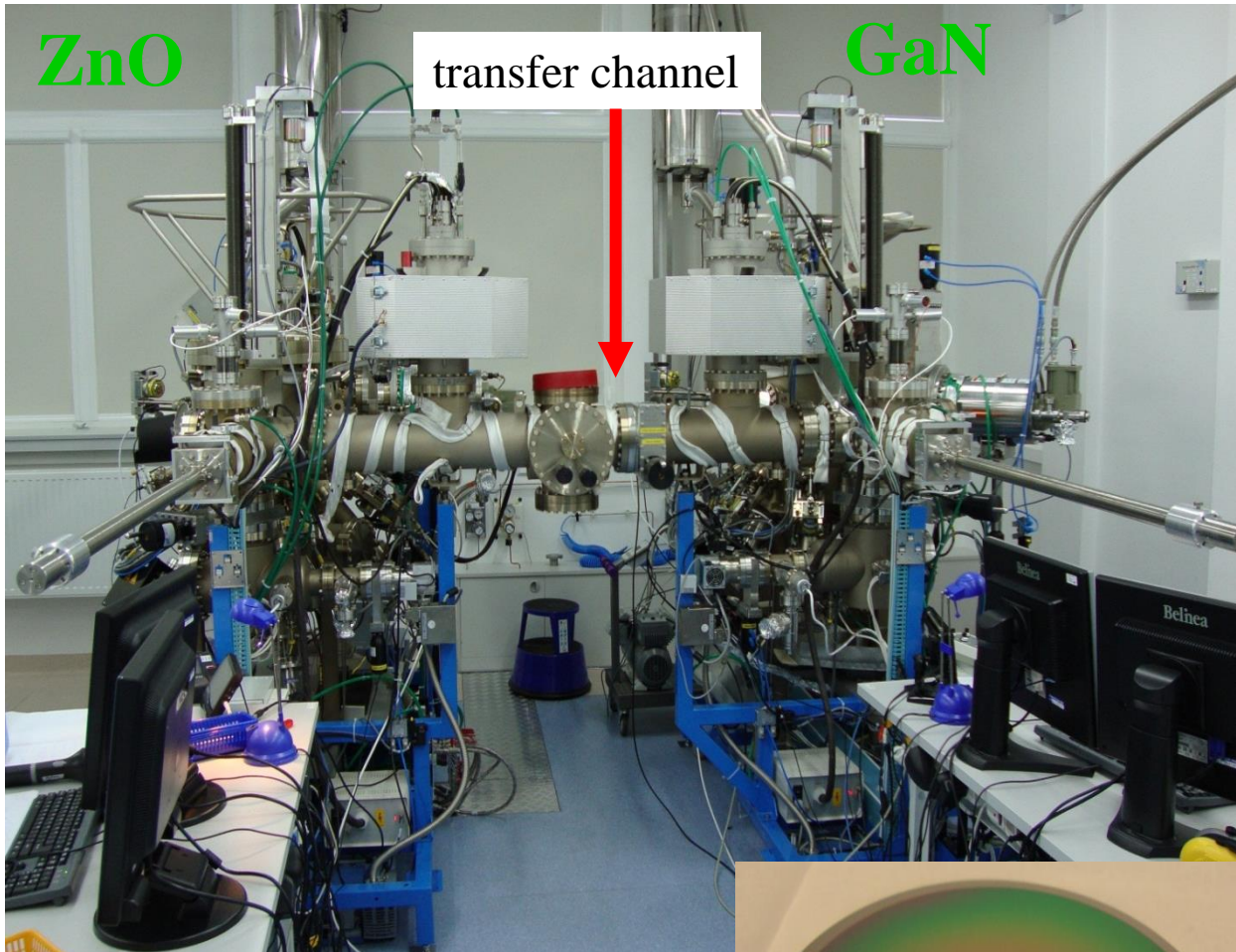
## pumping system



- UHV conditions
  - background pressure  $10^{-10} - 10^{-11}$  Tr
  - $\sim 10^{-5}$  Tr inside the molecular beam
- LN<sub>2</sub> filled cryoshroud :
  - additional pumping
  - freezing out atoms on the walls
  - lower „memory effect”
  - thermal separation of the sources

- independent sources of atoms/molecules;
  - the flux usually controlled by  $T_{\text{source}}$
- measurement of the flux – flux monitor
- mechanical shutters to open/close the source
- substrate heated by the heater
  - $T = \sim 200 \text{ }^\circ\text{C} - \sim 1000 \text{ }^\circ\text{C}$
- many *in-situ* diagnostic tools available

# Plasma-Assisted MBE (PAMBE) Riber Compact 21

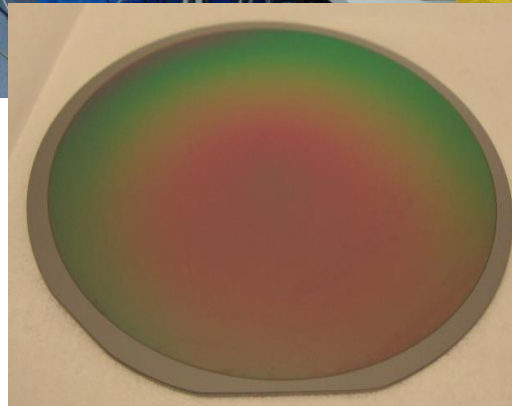


transfer channel

ZnO

GaN

growth on  
3" substrates  
(4" possible)



## TOOLS:

- ▶ optical pyrometer
- ▶ RHEED (k-Space)
- ▶ laser reflectometry
- ▶ LayTec EpiCurve TT (temperature, wafer curvature)
- ▶ line-of-sight quadrupole mass spectrometry (QMS)

## SOURCES:

- ▶ Ga x2
- ▶ Al x2
- ▶ In
- ▶ **RF nitrogen source**
- ▶ Si x2
- ▶ Mg
- ▶ Fe

## UHV in MBE – meaning how much ?

condition 1: mean free path of atoms > source – substrate distance

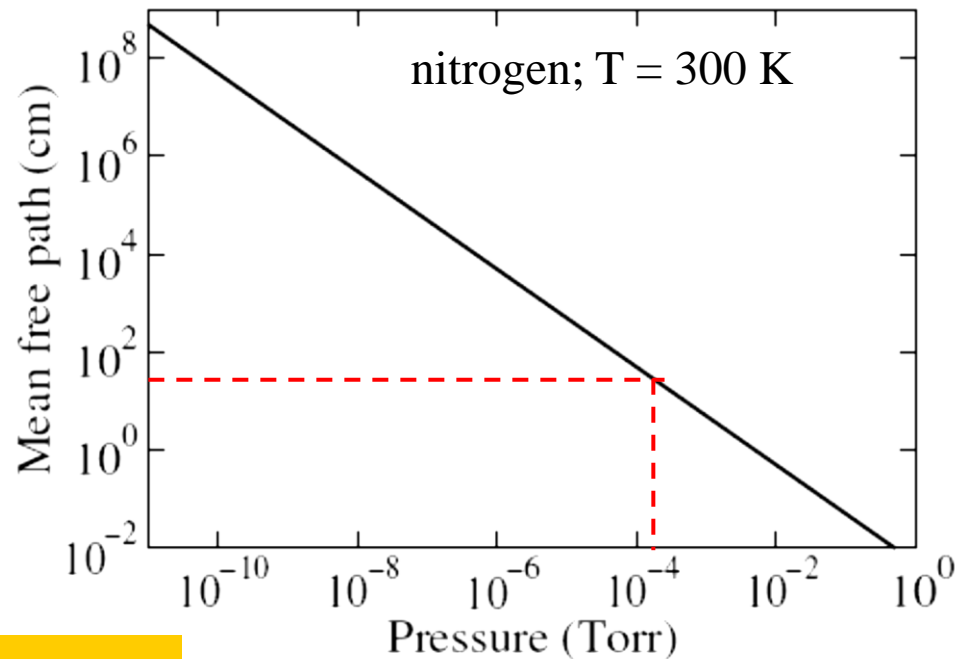
mean free path of atoms  $\lambda$  in gas phase under pressure  $p$

$$\lambda \approx \frac{5 \times 10^{-4}}{p[\text{Tr}]} [\text{cm}]$$

**$p = 10^{-4} \text{ Tr} \leftrightarrow \lambda = \sim 50 \text{ cm}$**

$p = 10^{-7} \text{ Tr} \leftrightarrow \lambda = \sim 0.5 \text{ km}$

$p = 10^{-11} \text{ Tr} \leftrightarrow \lambda = \sim 5\,000 \text{ km}$



ballistic atom transport (no collisions) in MBE  
- no UHV conditions needed

# UHV in MBE – meaning how much ?

condition 2: high purity of MBE grown layer

**assumption: all particles/atoms arriving stick to the substrate (no desorption)**

flux of particles in a gas under pressure  $p$  on area of  $1 \text{ m}^2$  in 1 sec.

$$J = \frac{p \cdot \sqrt{N_{Av}}}{\sqrt{2\pi m k_B T}} \quad \text{if } m=28 \text{ (N}_2\text{); } T=300\text{K then } J = 2 \times 2.8 \cdot 10^{22} \times p \left[ \frac{\text{atoms}}{\text{m}^2 \text{s}} \right]$$

number of lattice places on Si surface  $N = 2.8 \cdot 10^{19} [\text{atoms} \cdot \text{m}^{-2}]$

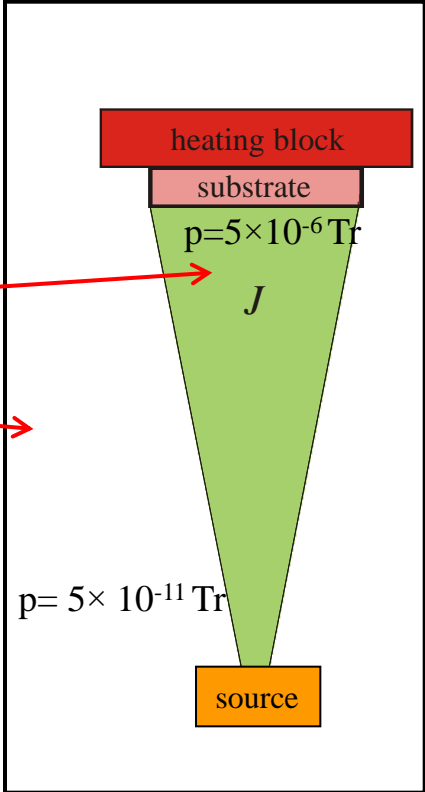
time to cover Si by 1 monolayer (ML) of particles  $\tau = \frac{N}{J} = \frac{5 \times 10^{-4}}{p [\text{Pa}]}$

$$p = 5 \times 10^{-6} \text{ Tr} \leftrightarrow \tau = 1 \text{ sec}$$

$$p = 5 \times 10^{-11} \text{ Tr} \leftrightarrow \tau = 10^5 \text{ s} \approx 28 \text{ h}$$

$$p = 5 \times 10^{-11} \text{ Tr} \Rightarrow 1 \text{ impurity atom per } 10^5 \text{ Si atoms}$$

impurity concentration in the bulk  $\sim 10^{17} \text{ cm}^{-3}$



**under real conditions:** layers more pure ( $\sim 10^{14} \text{ cm}^{-3}$  possible) since:  
 - often sticking coefficient  $< 1$  (desorption important)



# Three-chamber configuration of MBE system

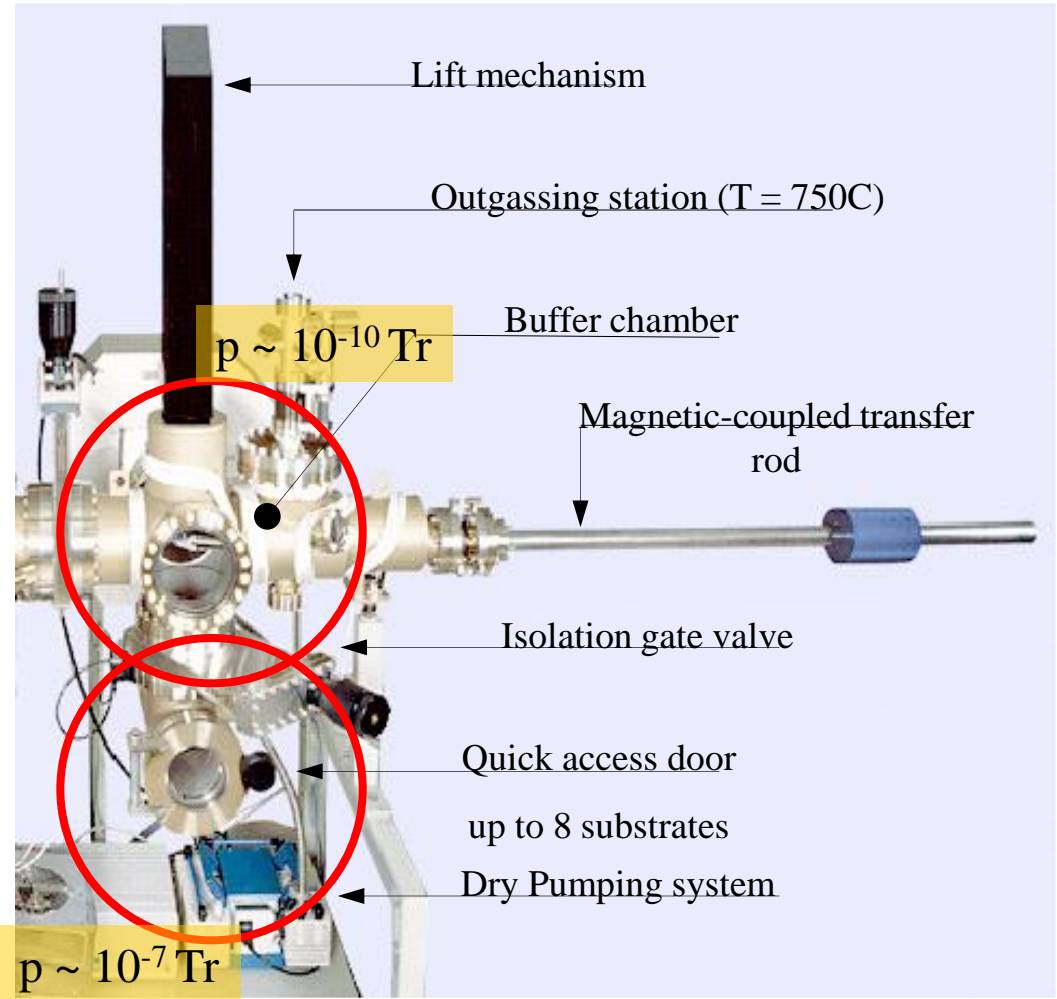
(example - Riber Compact 21)

growth chamber

loading and pre-annealing of the substrate



$p \sim 10^{-11} \text{ Tr}$



each chamber equipped with its own pumping system

## Generation of vacuum

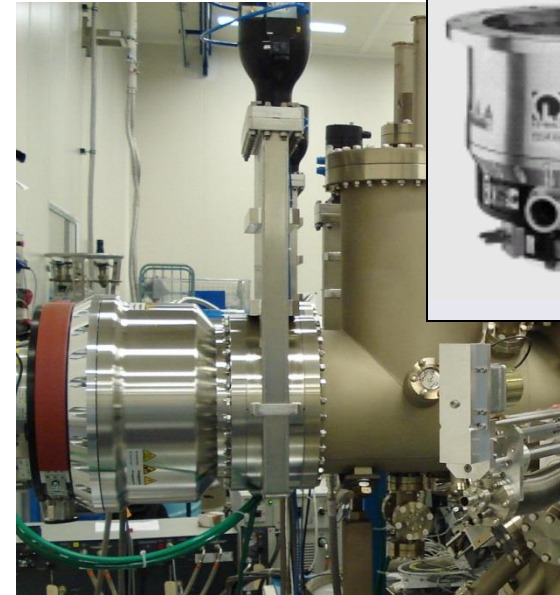
- mechanical pumps - rough pumps (rotary, Scroll, membrane, ..) and turbomolecular (UHV)
- cryopumps
- ion and titanium pumps



pumping speed (N<sub>2</sub>) 1200 l/sec



Helix CTI-10; pumping speed (N<sub>2</sub>) 3000 l/sec



pumping speed (N<sub>2</sub>) 2800 l/sec



long annealing of all chambers at T ~ 150° C after each opening of the system to remove residual gases and water





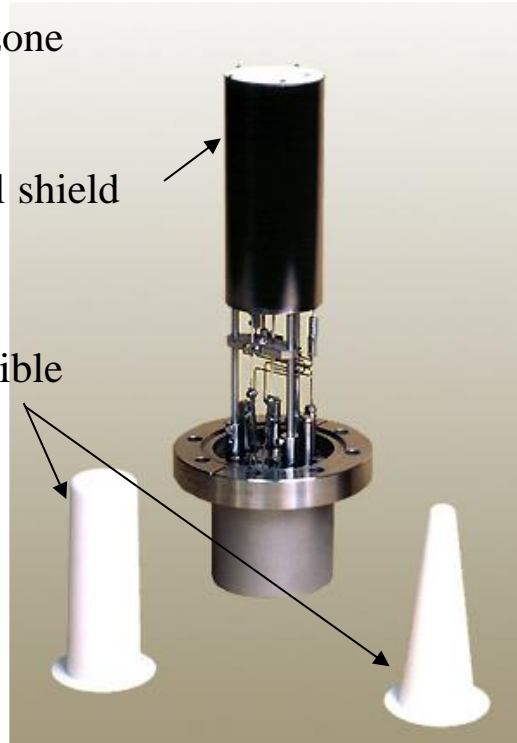
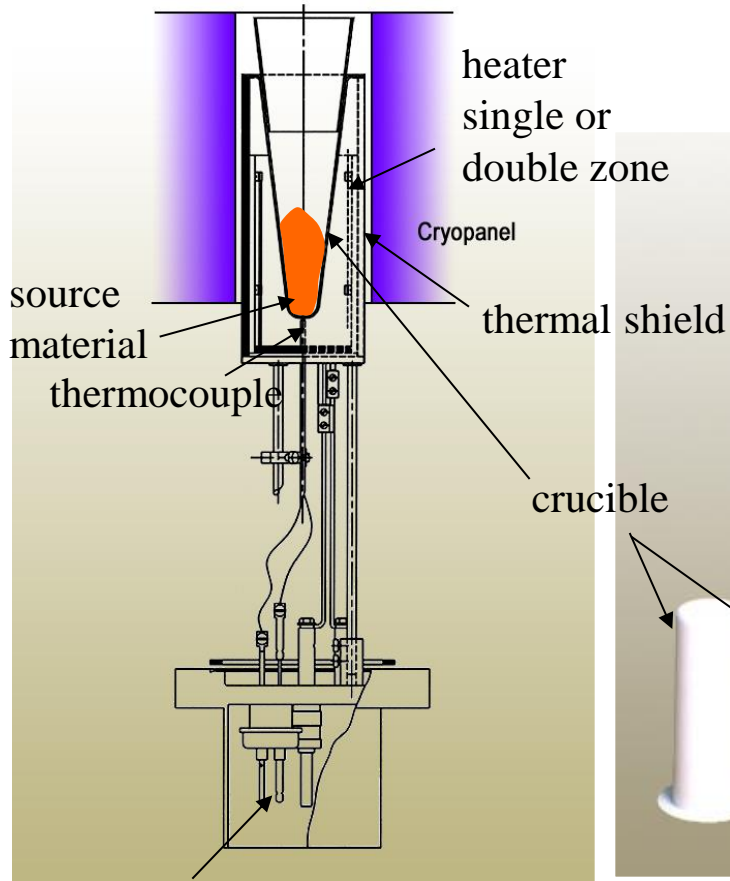
# Generation of molecular beams – effusion (Knudsen) cell

shutter



## Modern molecular sources:

- many sources in one system (10 in Compact 21 Riber)
- sources centered at the substrate  $\Rightarrow$  flux uniformity
- high flux stability;
  - flux drift  $< 1\%/day \Rightarrow \Delta T < 1^\circ C @ T \sim 1000^\circ C$
- each cell equipped with its own shutter
- cells thermally separated from each other



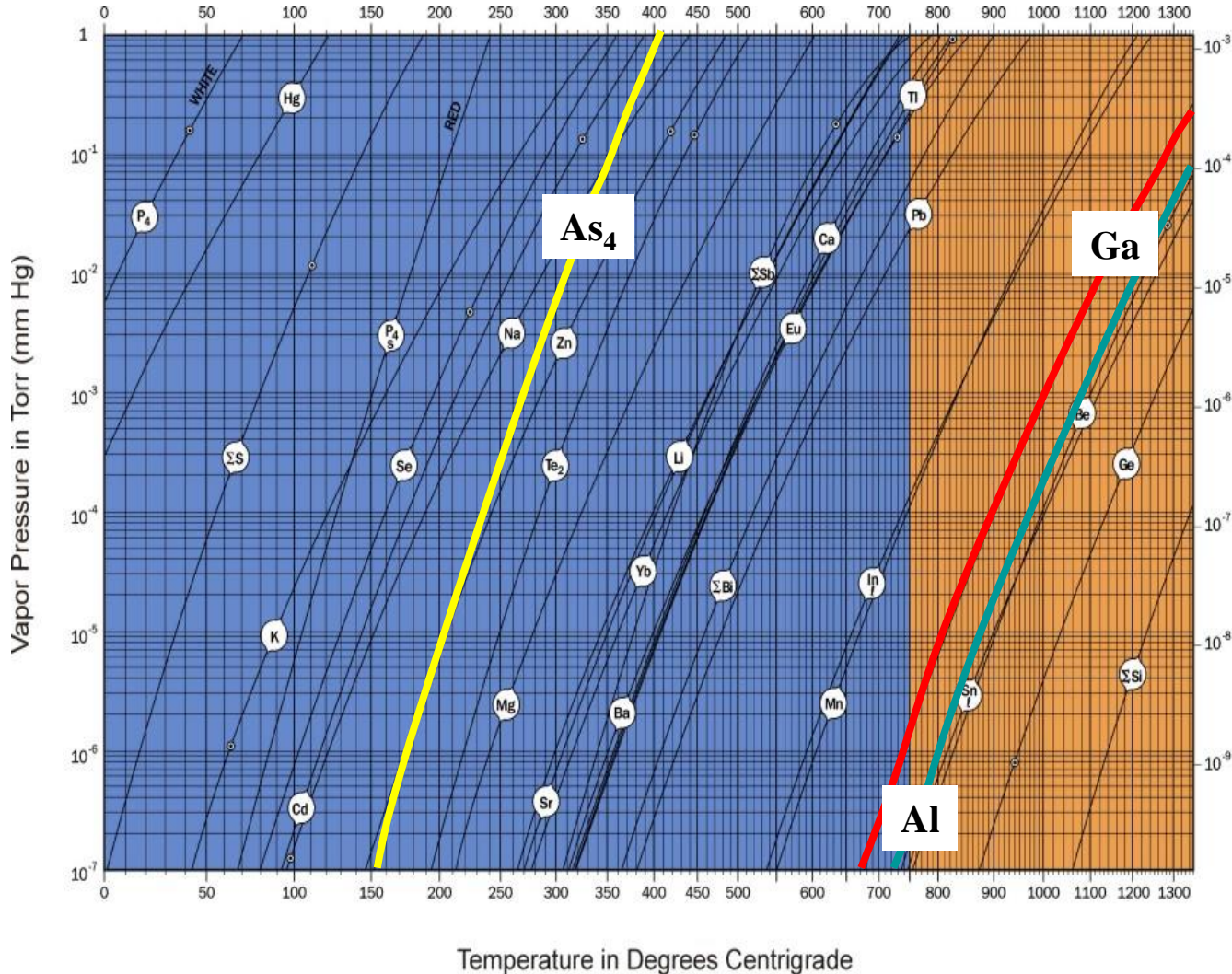
holes for sources and shutters in cryopanel of Compact 21 Riber system



# Generation of molecular beams – effusion (Knudsen) cell

**assumption:** vapor – liquid/solid equilibrium in the cell

vapor – liquid/solid equilibrium plots for selected elements



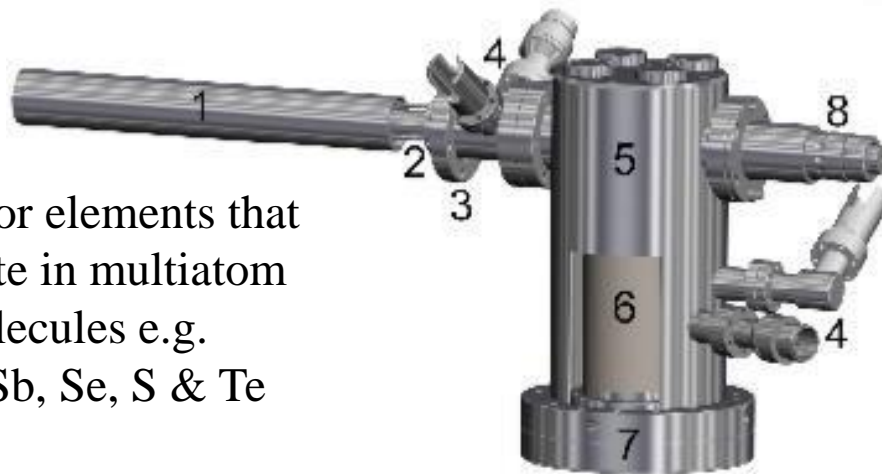
by changing  $T_{\text{source}}$  the flux from the cell  $p$  is controlled

# Generation of molecular beams –special sources

## valved cracker

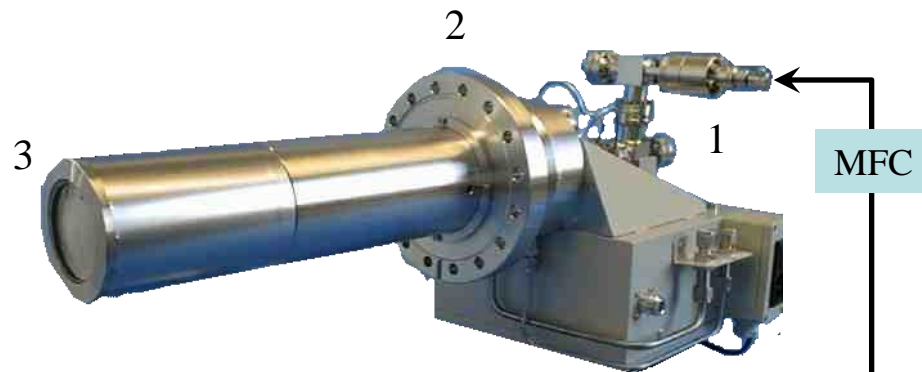
1. Cracking zone  $As_4 \rightarrow As_2$
2. connector + needle valve
3. main flange
4. power and TC connection
5. generation of  $As_4$  vapor
6. crucible with solid As

source for elements that sublime in multiatom molecules e.g.  
As, P, Sb, Se, S & Te



## plasma source

1. inlet of purified gas (MFC)
2. RF cavity
3. exit aperture (plate with small holes)



stable molecules  $N_2$ ,  $O_2$ , etc.  
excited in the cavity to produce  
active gas species

## gas injectors

gas sources with needle valve to deliver precursors used in Gas Source MBE (e.g.  $SiH_4$ ) or metalorganics in MO MBE

# growth rate in MBE – example GaAs

growth under As-rich conditions;  
 growth rate  $V_{gr}$  controlled by Ga flux;  
**assumption:** no Ga desorption

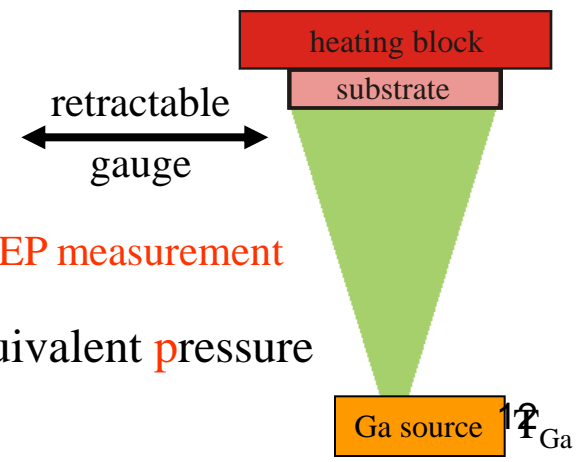
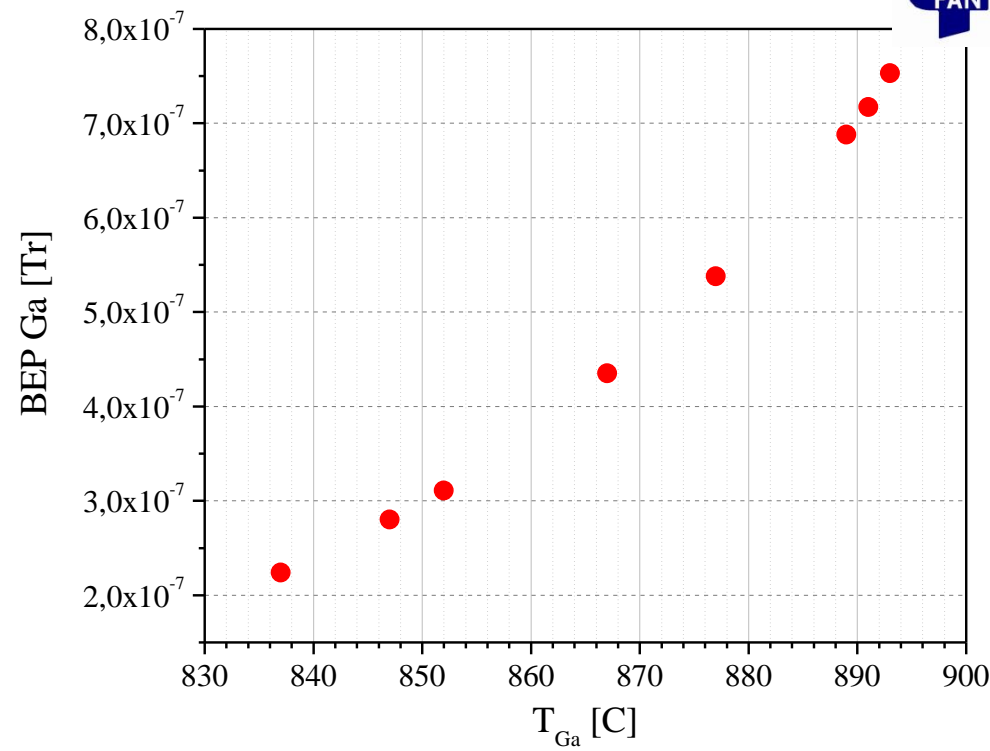
Ga flux  $J = 1.18 \times 10^{15} \frac{at}{cm^2 s}$

spec. volume of GaAs  $\Omega_0 = 2.27 \times 10^{-23} cm^3$

$$V_{gr} = J\Omega_0$$

$$V_{gr} = 2.67 \text{ \AA/s} = 1 \text{ ML/s} = 0.96 \text{ }\mu\text{m/h}$$

controlled growth of very thin (~1 ML) layers and epitaxial structures



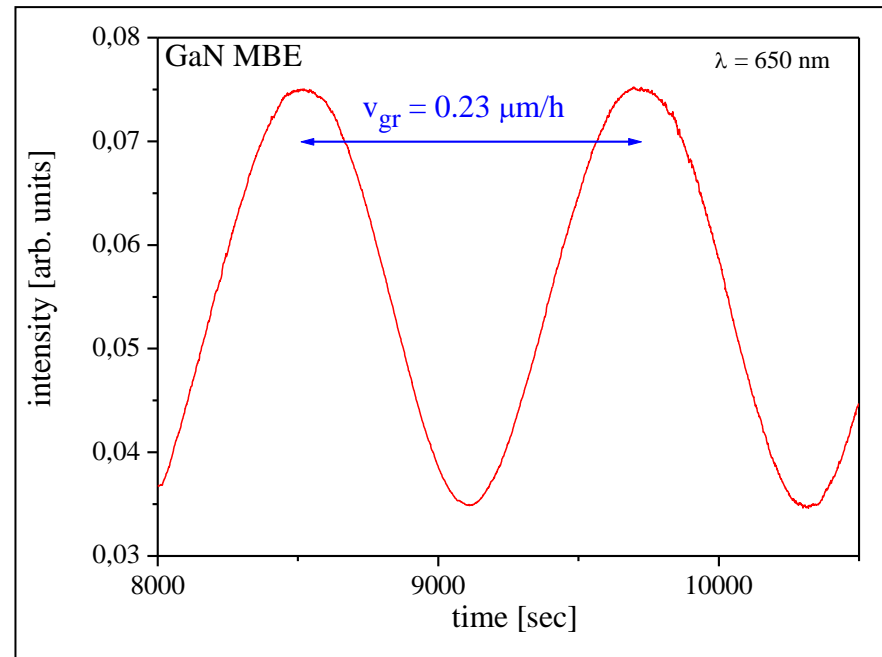
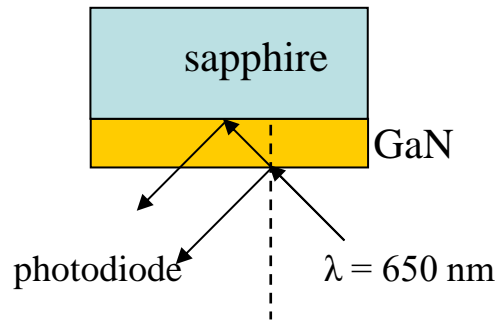
**BEP = beam equivalent pressure**

## *in situ* growth monitoring

vacuum is transparent for light, electrons, X-rays, ...  
 wide possibilities to „observe” surface of growing epilayer

### laser reflectometry

*growth rate, evolution of surface roughness, ...*



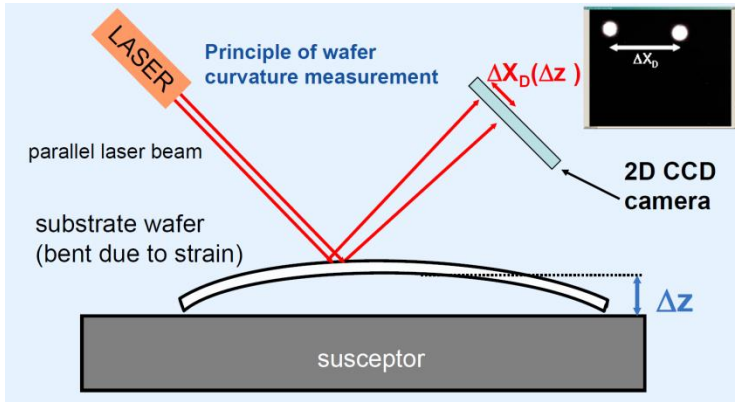
optical pyrometry *measurement of sample T based on the black body emission spectrum*

to be discussed later

ellipsometry



# in situ growth monitoring

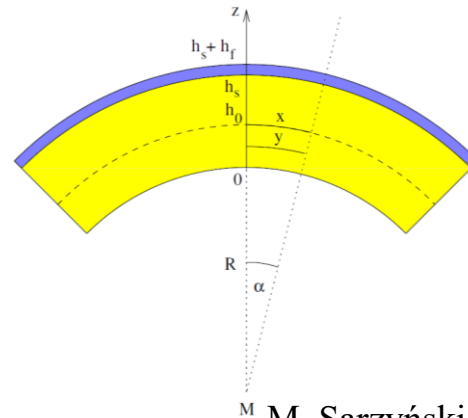


laytec.com

# sample curvature

stress and bow evolution in real time

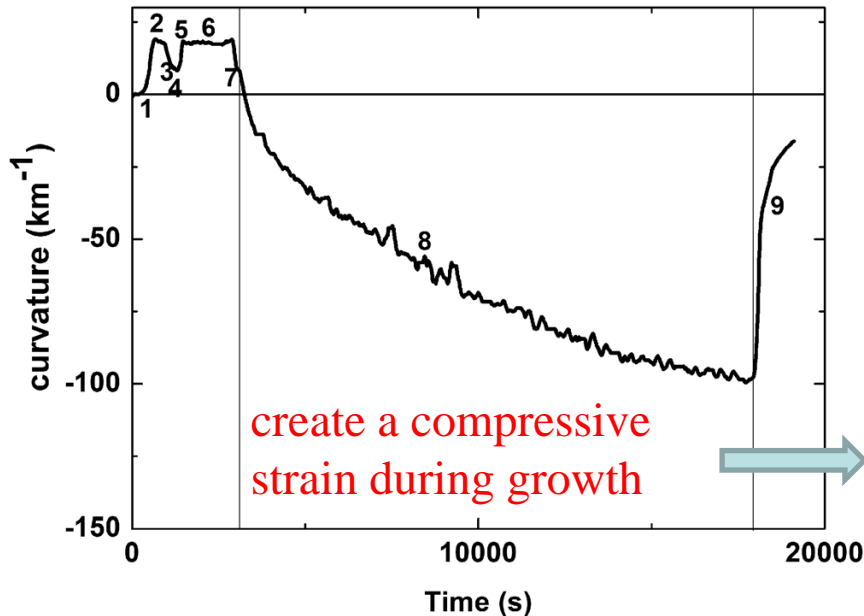
LayTec – EpiCurve; Riber – EZ-Curve, k-Space - kSA MOS...



$$1/R \propto \frac{\epsilon \times h_e}{h_s^2}$$

Stoney formula

M. Sarzyński PhD 2007

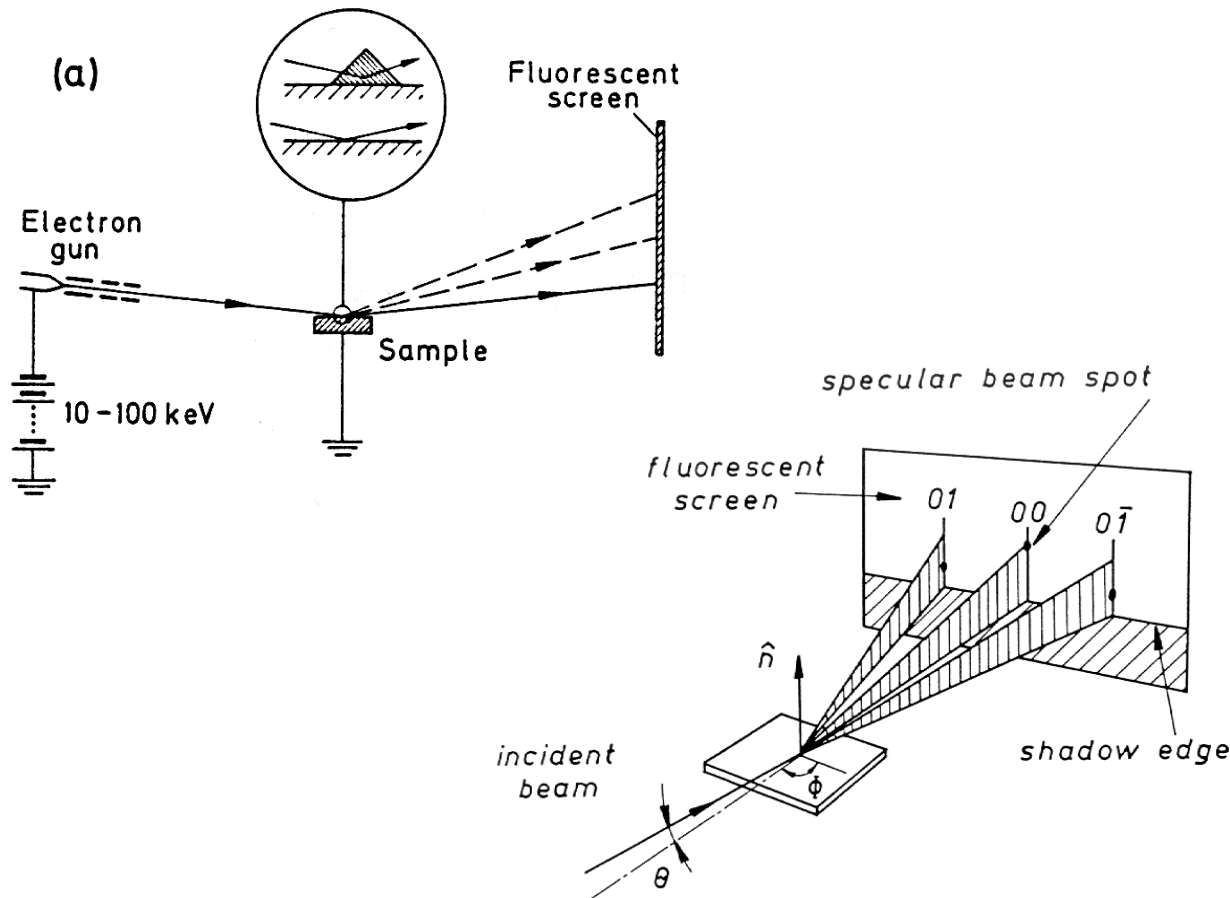


PAMBE growth of GaN/AlGaN on Si(111): mismatch of thermal contraction leads to tensile strain and cracking upon cooling

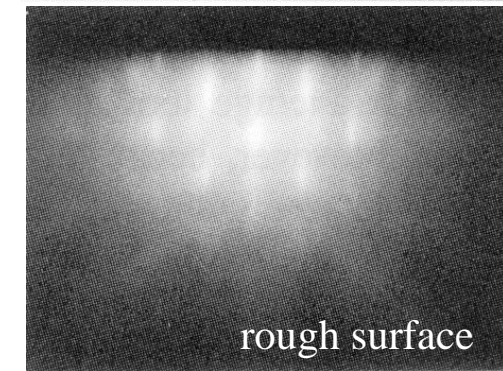
thick GaN buffers can be grown on Si w/o cracks

# *in situ* growth monitoring - reflection high energy electron diffraction (RHEED)

- probing of surface by diffraction of electron beam striking the sample at a very small angle relative to the sample surface ( $1^\circ - 3^\circ$ )
- electron energy 5 – 20 keV; wavelength  $\sim 0.1 \text{ \AA}$
- ideal 2D surface – set of parallel streaks



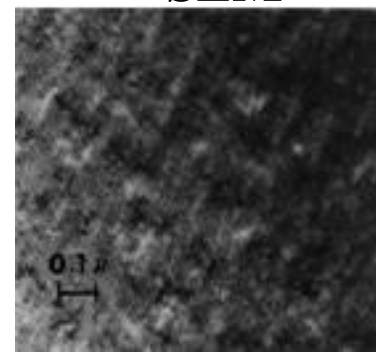
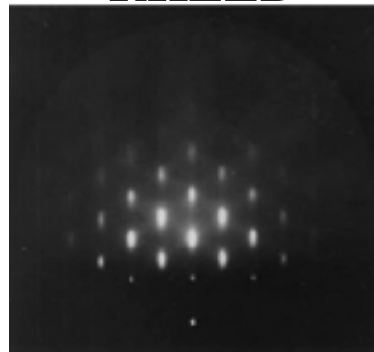
## Si(001) RHEED patterns sputter-cleaned surface



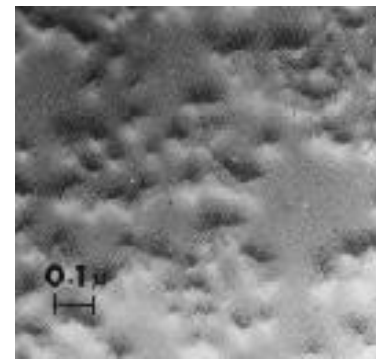
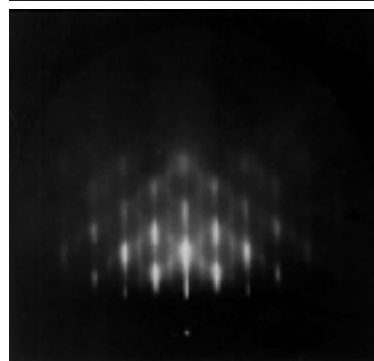
GaAs substrate after oxide desorption

**RHEED**

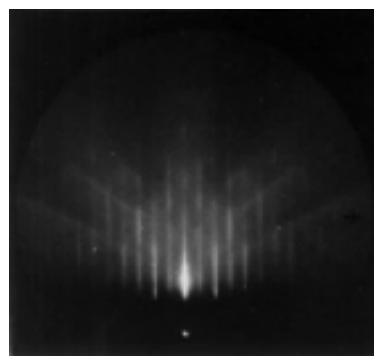
**SEM**



+ MBE growth of 15 nm GaAs

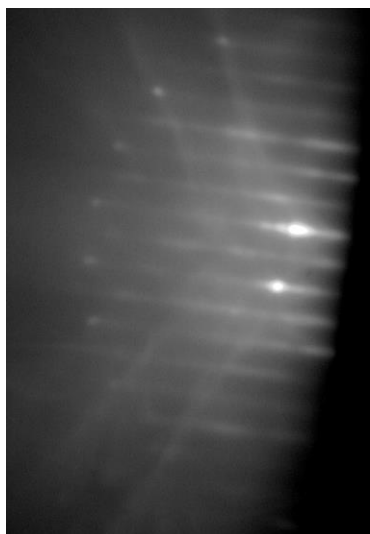


+ MBE growth of 1 μm GaAs

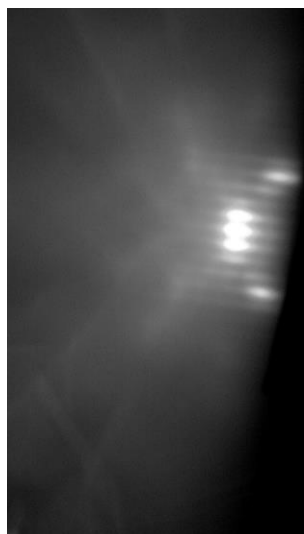


# in situ growth monitoring – RHEED – (2x4) GaAs surface reconstruction

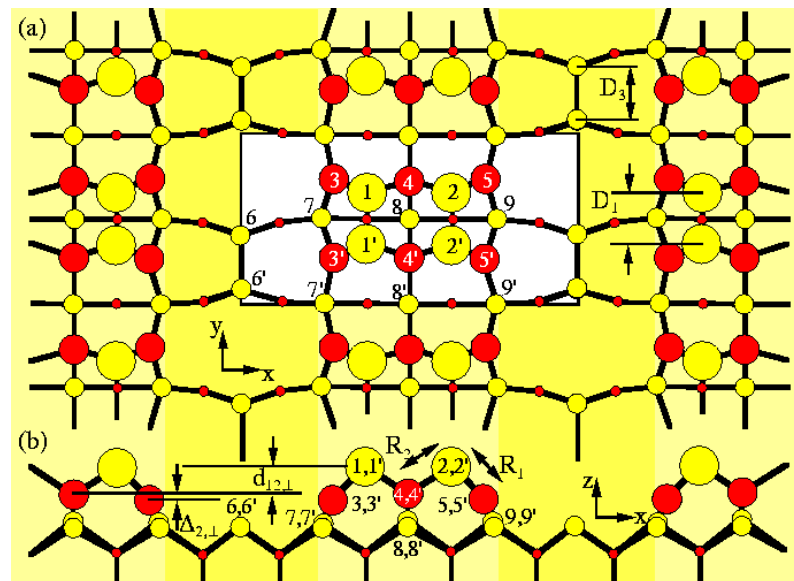
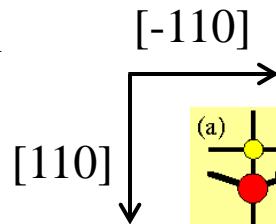
RHEED pattern depends on the azimuth



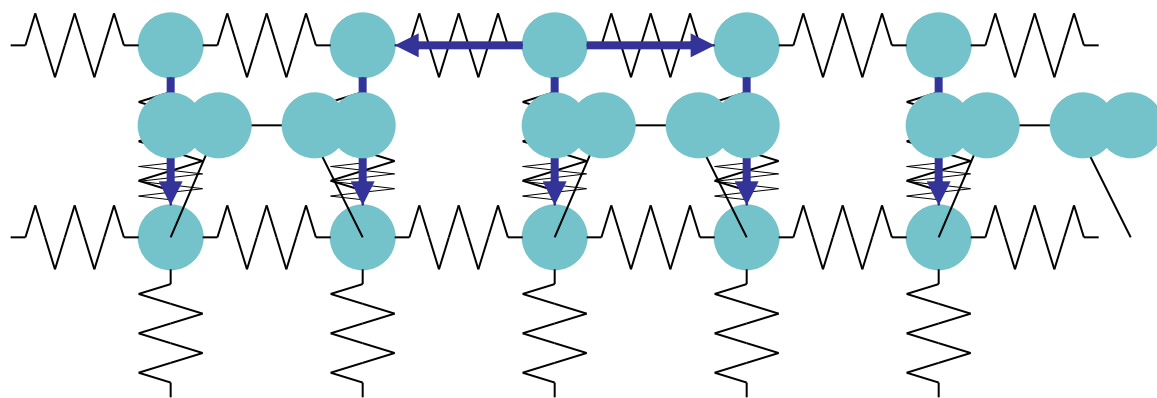
azimuth [110] (2x)



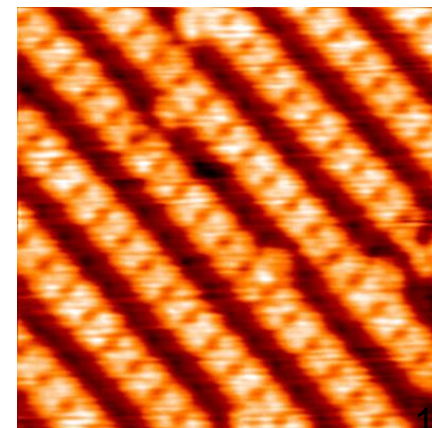
azimuth [-110] (4x)



## surface reconstruction – change of periodicity

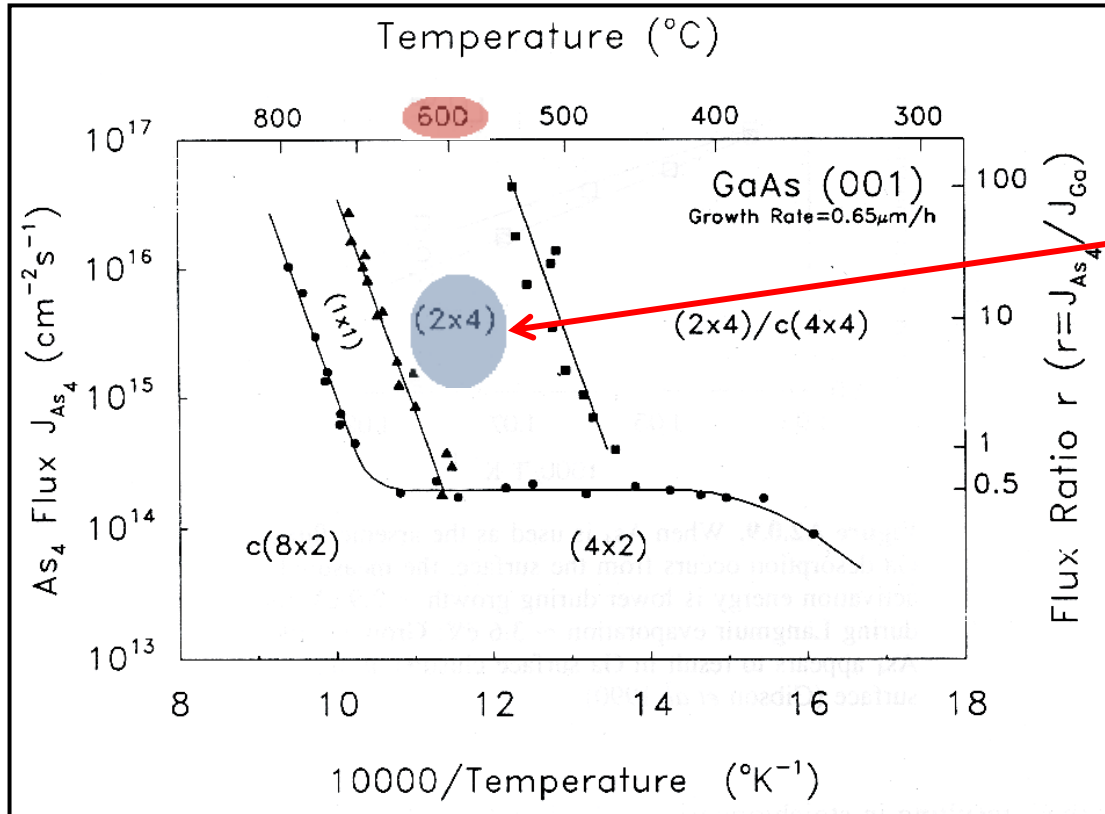


GaAs(001) - STM



# RHEED – surface phase diagram of GaAs

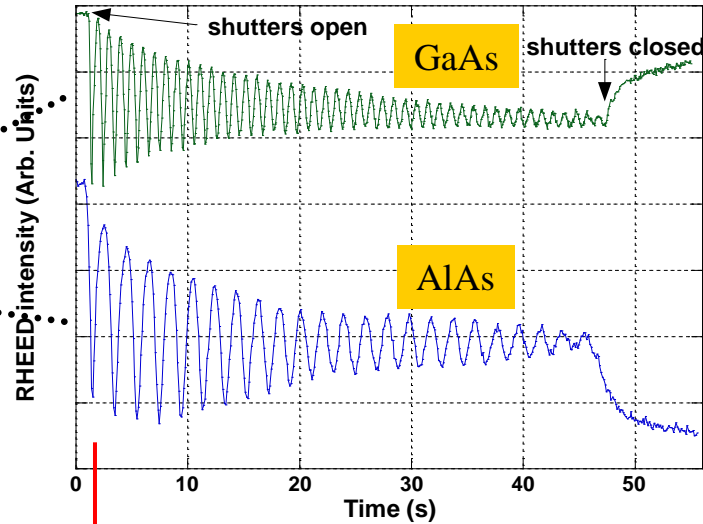
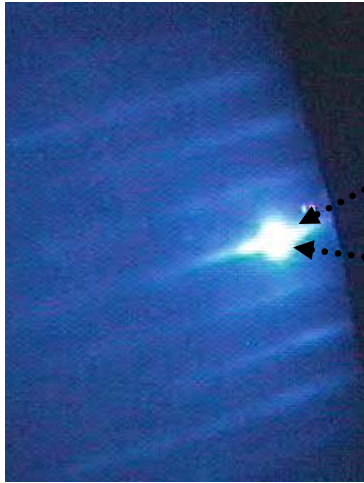
various surface reconstructions depending on T, As/Ga ratio, ...



- **As-stable (2 × 4):** typical GaAs growth conditions by MBE
- surface reconstruction strongly depends on surface temperature – RHEED as surface thermometer



# RHEED – growth rate measurements (RHEED oscillations)



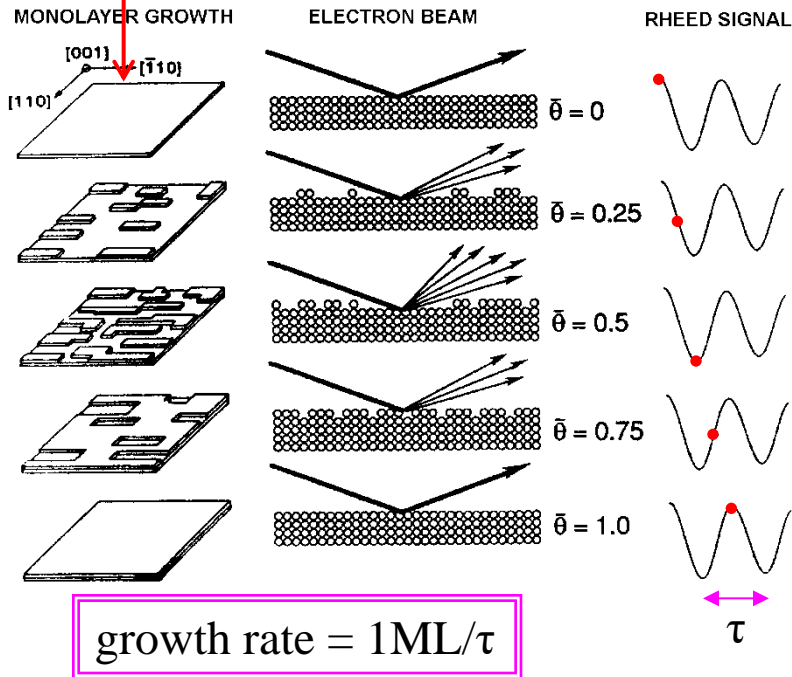
Ga shutter closed:

**GaAs:** signal recovery  $\Rightarrow$  high surface mobility of Ga adatoms, smoothing of the surface

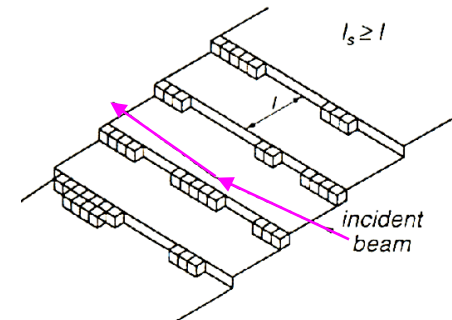
Al shutter closed:

**AlAs:** signal damping  $\Rightarrow$  no surface smoothing; low surface mobility of Al adatoms

start of growth



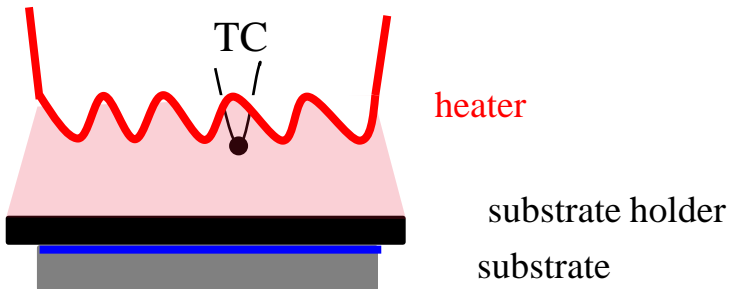
- RHEED oscillations – observation of periodic change of surface roughness as the layer grows
- required: 2D nucleation – layer by layer growth
- no RHEED oscillations for step flow growth



- V element-rich conditions

# substrate temperature measurement in MBE

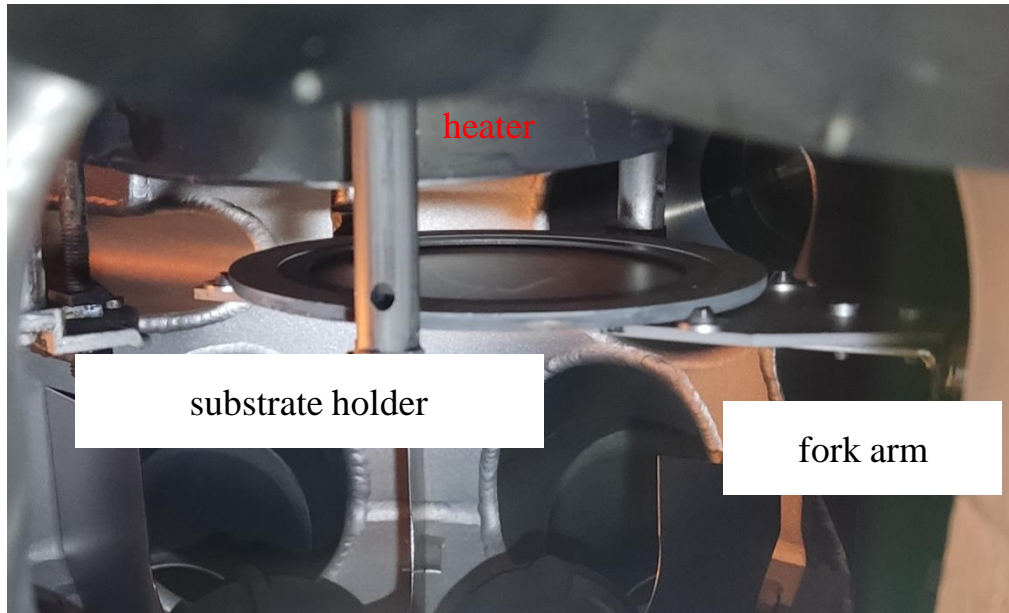
under vacuum heat transport by radiation only



for low T growth substrate glued by In or In/Ga to improve thermal contact of the substrate and the holder

often TC reading used as the measure of the substrate surface T

- TC measures T of the heater !!!
- (TC has no thermal contact with the substrate)
- even if reproducible (?) then for the particular system only
- T values useless for others



**Do not do that !!!!**

*ethical guidelines in science require that your data are presented in the way allowing others to repeat (verify) your experiment !!!!*

# substrate temperature measurement in MBE

optical pyrometry in IR  $\lambda = 1 - 3 \mu\text{m}$  measurement of sample  $T$  based on the black body emission spectrum

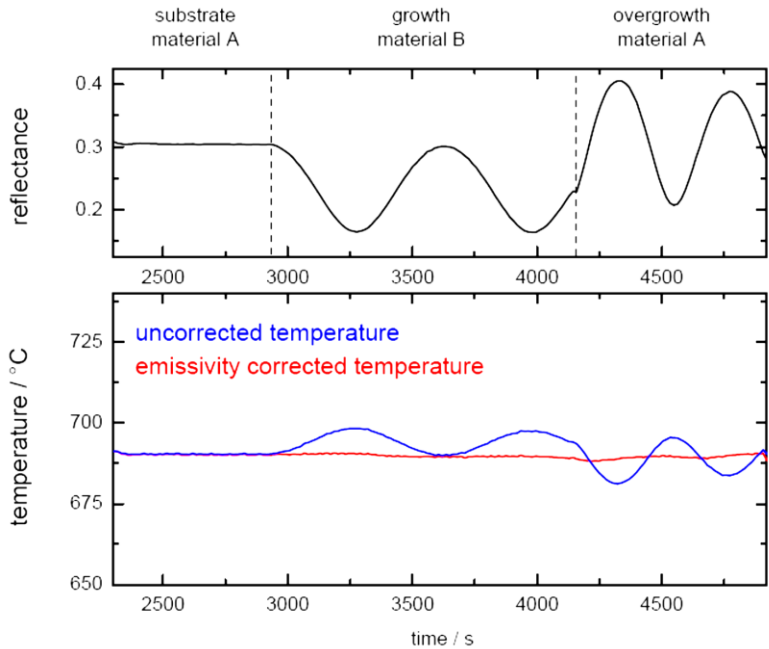
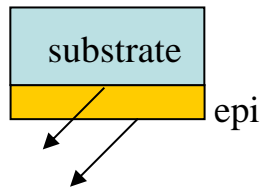
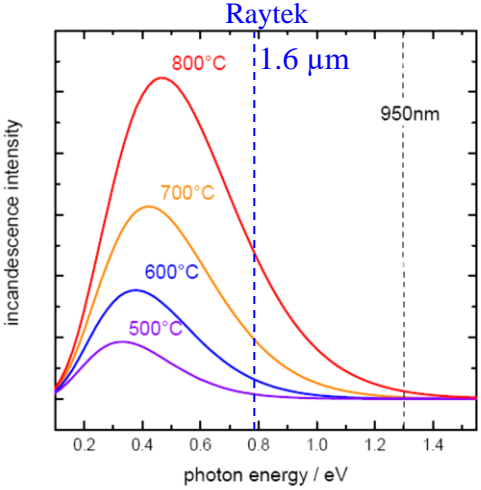


Fig. 1: Emission of a black body according to Planck's law: the emitted radiation power increases together with the temperature, the maximum of the curves shifts to higher photon energies.

IR light interference leads to oscillations of the signal; thus to artificial oscillations of T reading  
 emissivity corrected pyrometry – simultaneous measurement of pyrometry signal and light reflection; allows to eliminate interference-induced oscillations  
 - still the problem exists if the viewport transmission changes during a long growth campaign

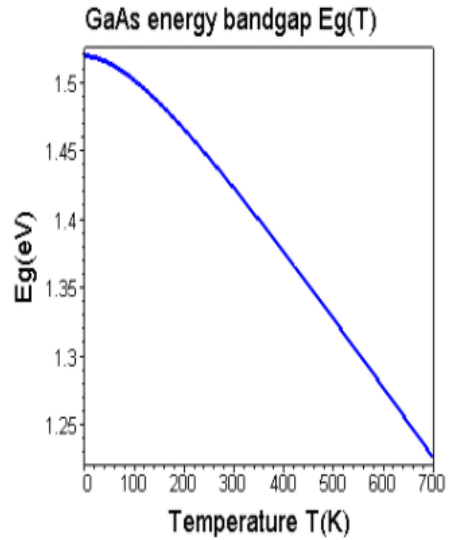
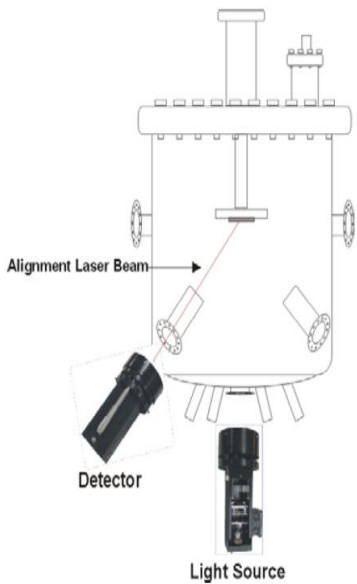


- T of which part pyrometry measures if the substrate is transparent (e.g. sapphire, GaN, ZnO, ....) ?

T of the substrate holder !!!!!  
 it may differ from the substrate surface T by tens of °C

# substrate temperature measurement in MBE

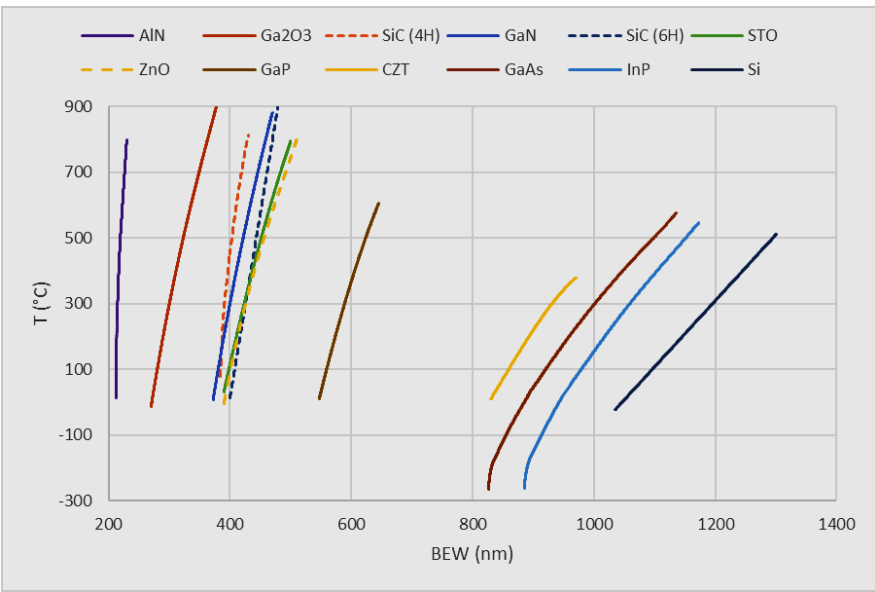
BandiT – k-Space Inc.



- measurement of the **substrate optical absorption edge (not the light intensity)**, the absolute temperature of the wafer can be determined. This absorption edge, which is directly proportional to the band gap of the material, is temperature dependent.

$$E_g(T) = 1.519 - 5.408 \cdot 10^{-4} T^2 / (T + 204)$$

J. S. Blakemore J. Appl. Phys. 53 (1982)



- immune to changing viewport transmission, stray light, and signal contribution from substrate or source heaters, all sources of measurement error for pyrometers

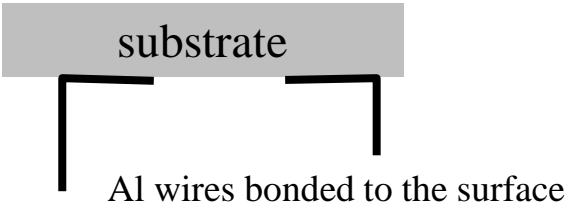
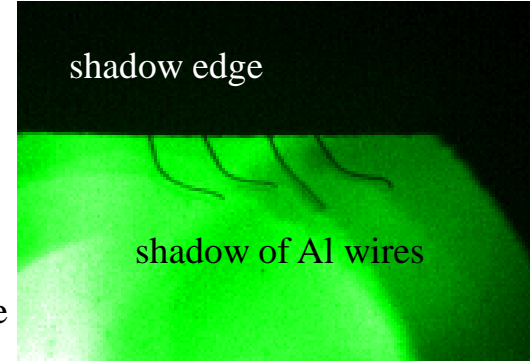
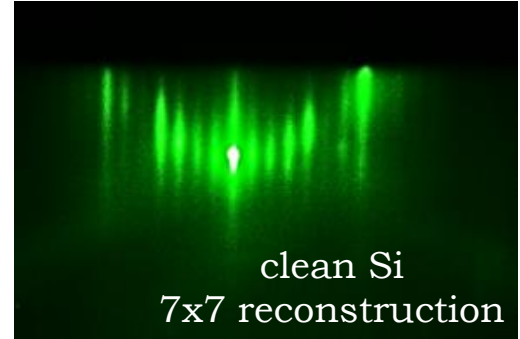
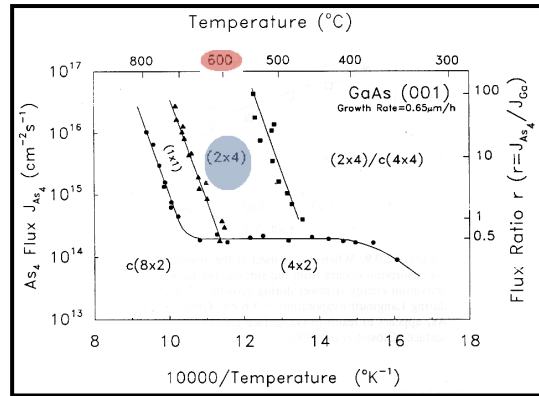
- requires that your substrates are carefully calibrated (factory calibration files)
- different substrate doping can change the result
- does it measure substrate *surface* T ?

# substrate temperature measurement in MBE

rely on surface phenomena if possible

some examples:

1. oxide desorption from GaAs (RHEED observation)  $\sim 560^\circ\text{C}$
2. change of surface reconstruction (RHEED pattern) of GaAs
3. change of surface reconstruction of Si(111)  $7\times 7 \rightarrow 1\times 1$  @  $830^\circ\text{C}$
4. melting point of suitable metal (Al  $660.3^\circ\text{C}$ )
5. desorption rate of deposited metal (e.g. Ga from GaN surface)
6. others, depending on your material system



Try to calibrate surface T as precisely as possible. For you the run to run T reproducibility is the most important. In some cases differences in T readings between various MBE systems as large as  $\pm 50^\circ\text{C}$  are standard. It is crucial to describe in details the way of substrate T measurement in your experiment !!!



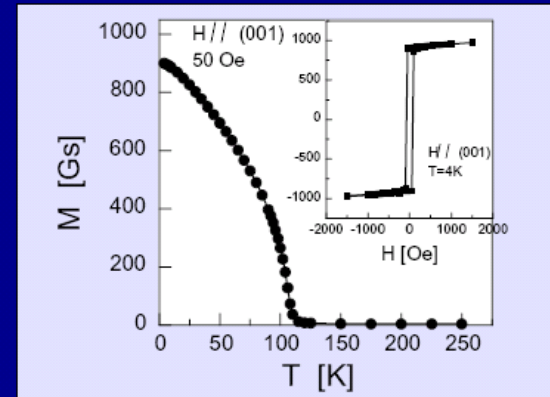
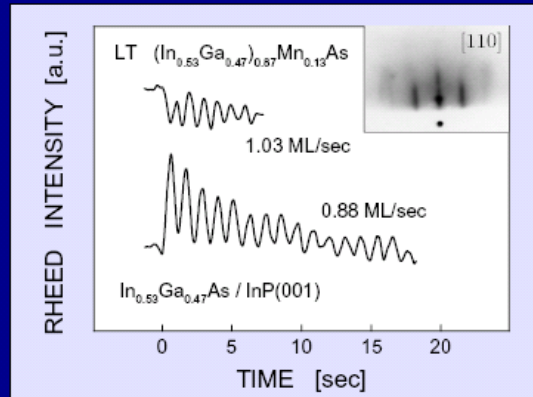
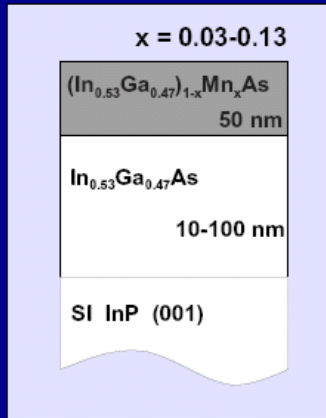
# Application of MBE: incorporation of Mn above the solubility limit in III-V's

nonequilibrium growth in MBE  $\Rightarrow$  possibility to grow (Ga, In)As layers with high concentration of randomly distributed Mn

## FERROMAGNETIC $(\text{In}_{0.53}\text{Ga}_{0.47})_{1-x}\text{Mn}_x\text{As}$ GROWN ON InP(001)



CURIE TEMPERATURE  $T_c \sim 100\text{-}110\text{ K}$

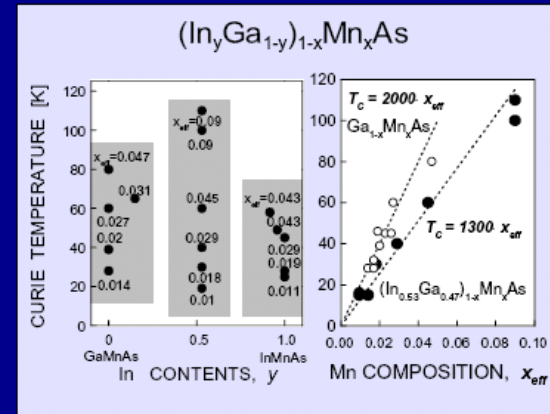
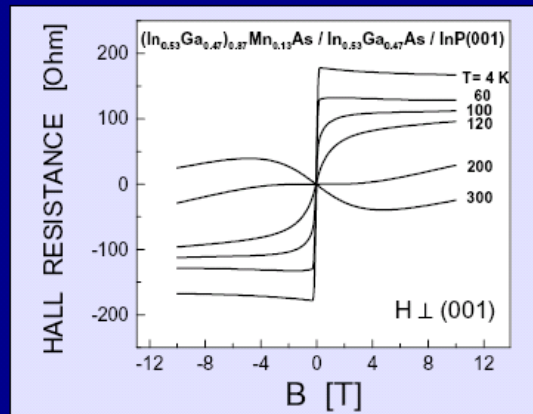


near-lattice matched to InP (001) substrate

$T_s = 200\text{-}260\text{ }^\circ\text{C}$

- optimized vs. Mn

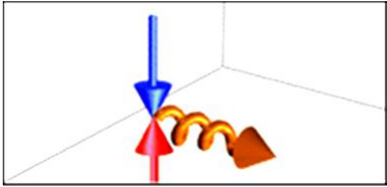
easy axis in-plane



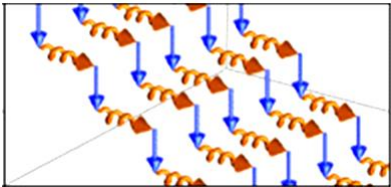
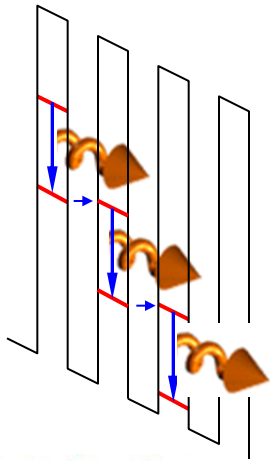
# Application of MBE: superlattices in optical devices

cascade IR laser GaAs/AlGaAs (~9μm)

conventional laser



cascade laser



„cascade of electrons”

multiple photon emission from one electron

[www.bell-labs.com/org/physicalsciences/projects/qcl/qcl2.html](http://www.bell-labs.com/org/physicalsciences/projects/qcl/qcl2.html)

GaAs:Si	1,0μm	$n = 6,0 \cdot 10^{18}$
GaAs:Si	3,5μm	$n = 4,0 \cdot 10^{16}$

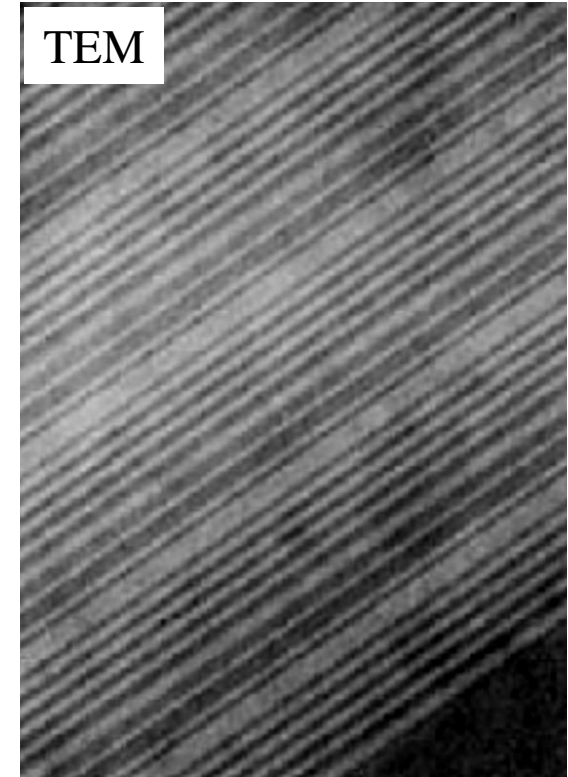
Active region  
1,62μm  
(45nm x 36)

AR = 36 x this segment

GaAs:Si	3,5μm	$n = 4,0 \cdot 10^{16}$
GaAs:Si	1,0μm	$n = 6,0 \cdot 10^{18}$
GaAs:Si	1,0μm	$n = 2,0 \cdot 10^{18}$
GaAs:Si	450μm	$n = 2,0 \cdot 10^{18}$

- GaAs
- AlGaAs x(Al)=45%
- Active Region (AR)
- substrat

2,8 -underline means Si doped



ITE Warszawa - Kosiel et al. EuroMBE 2009, Zakopane

MBE allows growth of complicated set of superthin epilayers with very high crystallographic quality

# Application of MBE: modulation doping ( $\delta$ -doping)

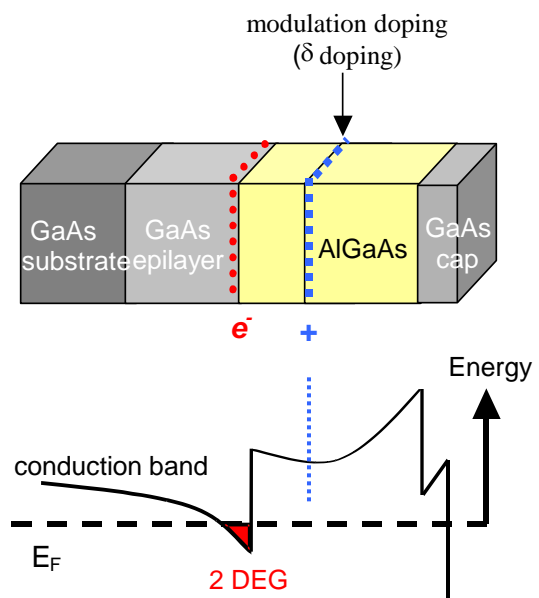
**problem:** doping required for high electrical conductivity

**BUT**

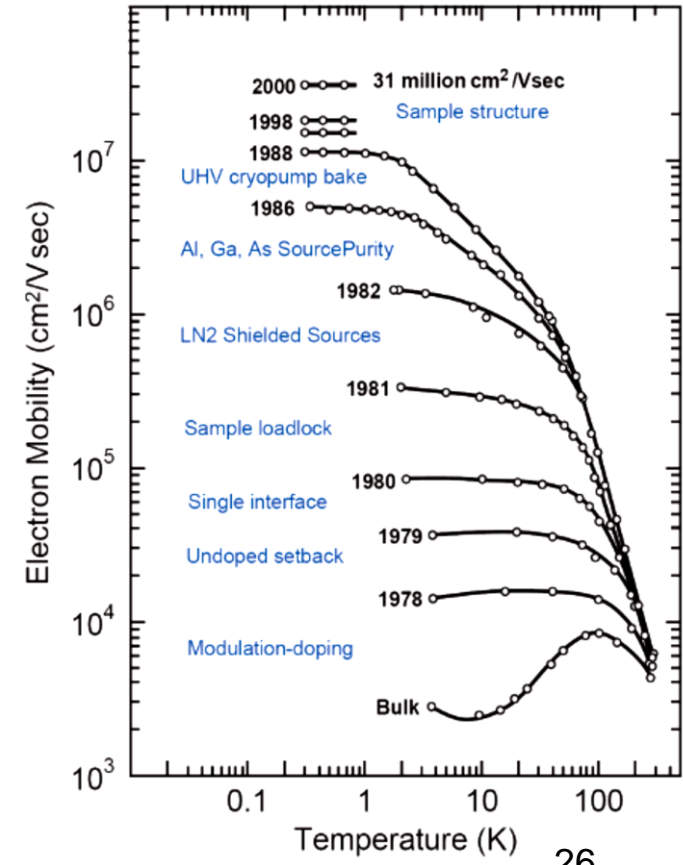
impurities scatter charge carriers  $\Rightarrow$  lower mobility at low T

**solution:** separate in space source of charge carriers (dopants) from the electrical conductivity channel

70ties, Art Gossard i Horst Störmer z Bell Labs.

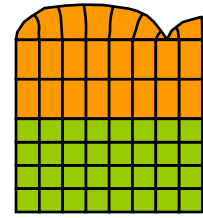


transfer of charge carriers into the 2D channel and their separation from dopants  $\Rightarrow$  higher  $\mu$

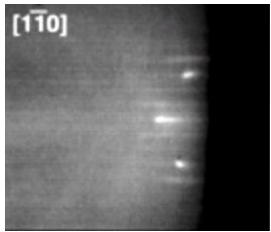


# Application of MBE: self-organized quantum dots (QD)

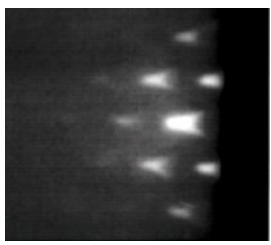
lecture 8.03.2022 - surface deformation as the way to relax lattice mismatch strain  
 7% lattice mismatch in the InAs/GaAs system



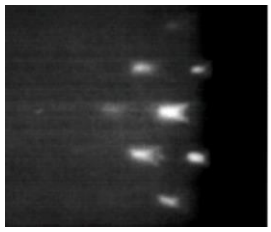
InAs/(001) GaAs  
 azimuth [1-10]



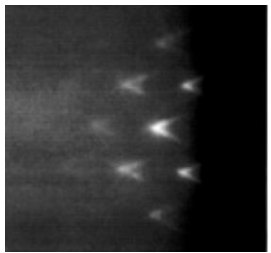
1 ML InAs



2 MLs InAs



3 MLs InAs

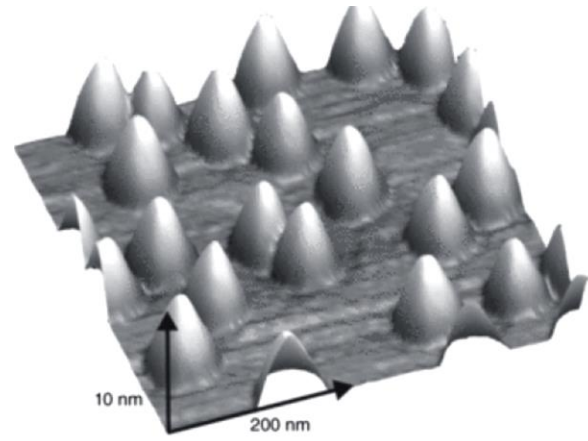
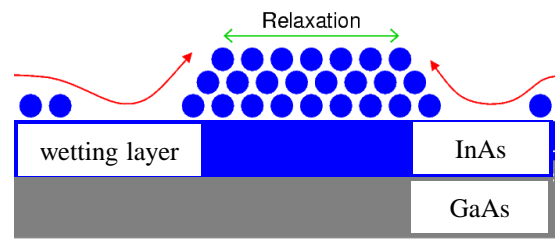


30 MLs InAs

growth 3D

## growth modes:

- Frank-van der Merwe (layer-by-layer)
- Stranski-Krastanov (layer + island)**
- Volmer-Weber (island)



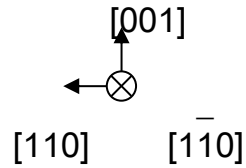
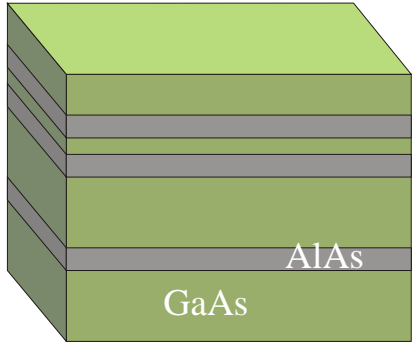
InAs QDs on GaAs:

- dislocation free
  - diameter ~20nm
  - height a few nm
  - broad distribution of dimensions
  - random positions on the substrate
- (self-organization)

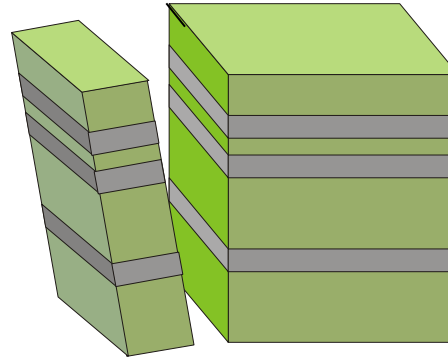
# Application of MBE: organized quantum dots (QD)

E. Uccelli et al. EuroMBE 2009, Zakopane

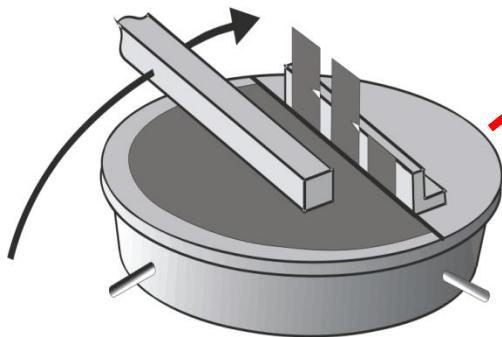
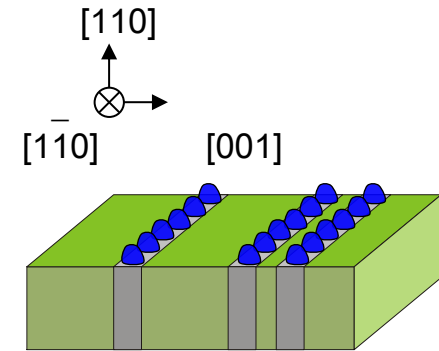
1. growth of AlAs/GaAs (001) layers



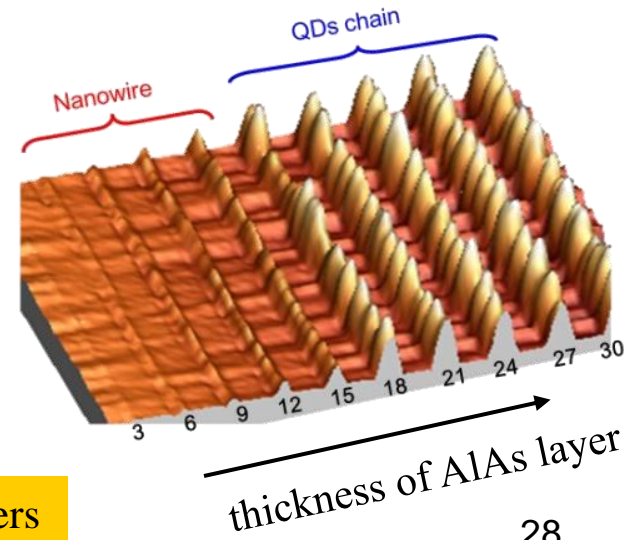
2. *In situ* cleavage: (110) flat surface



3. growth of InAs on the cleaved (110) surface



shutter blade cuts the wafer



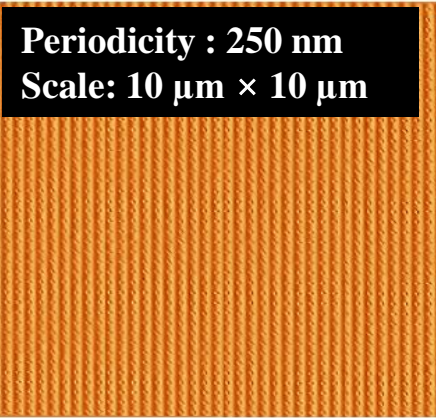
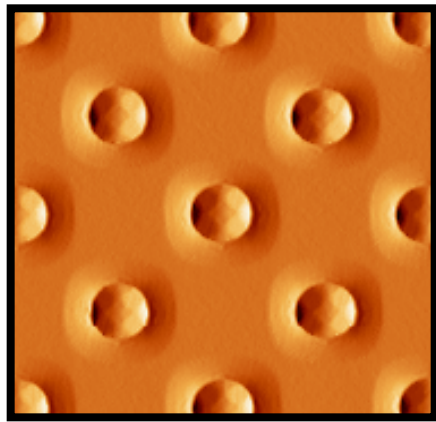
- in-plane nanowires for thin AlAs layers
- QDs for thicker AlAs layers



# Application of MBE: organized quantum dots (QD)

- advantages of ordering:
- better uniformity of QDs' dimensions (more uniform light emission  $\lambda$ )
  - single QD can be addressed
  - ...

G. Chen (EuroMBE 2009, Zakopane)  
E-beam lithography + RIE



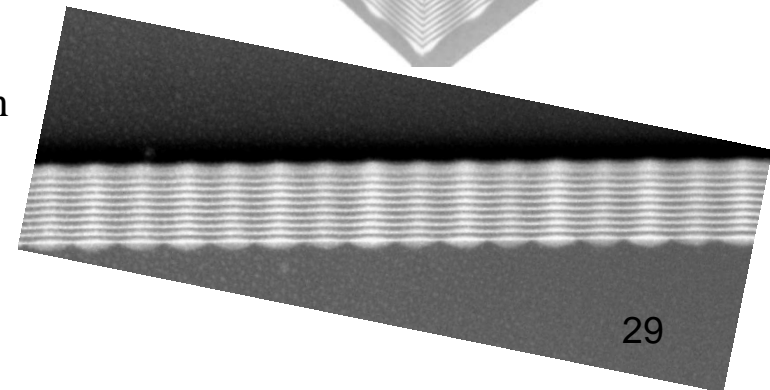
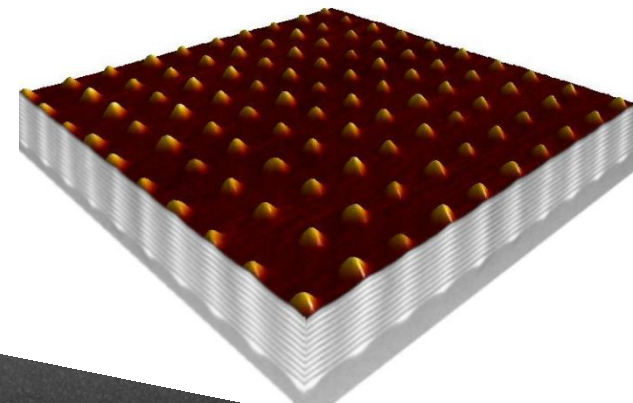
**Periodicity : 250 nm**  
**Scale: 10  $\mu\text{m}$   $\times$  10  $\mu\text{m}$**

## Ge QDs on Si substrate

- lithography of the substrate (e-beam or X-Ray)
  - etching of the pattern (RIE)
  - MBE growth of QDs
- 
- positions of QDs in the next layer reflects their distribution in the layer underneath (coupling via the strain field)

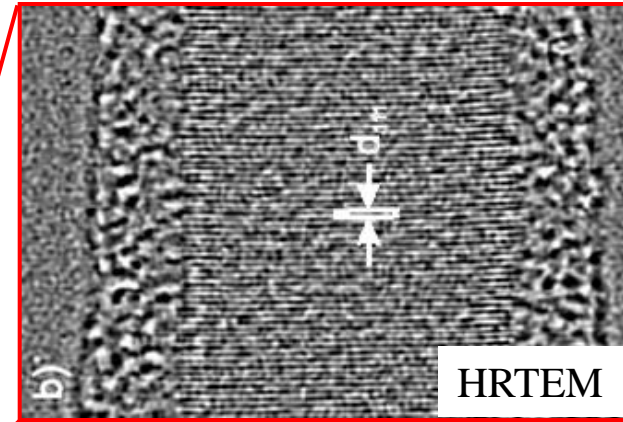
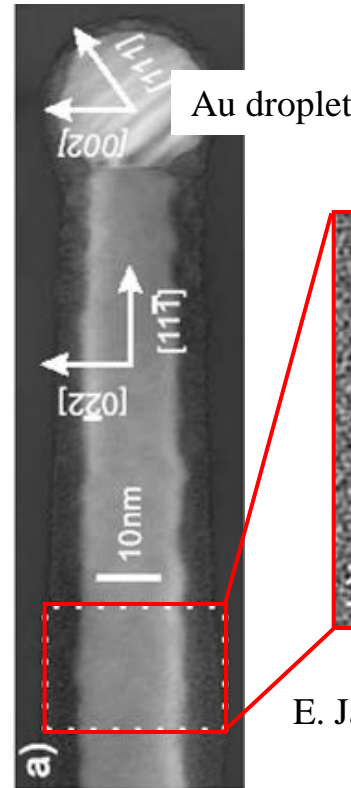
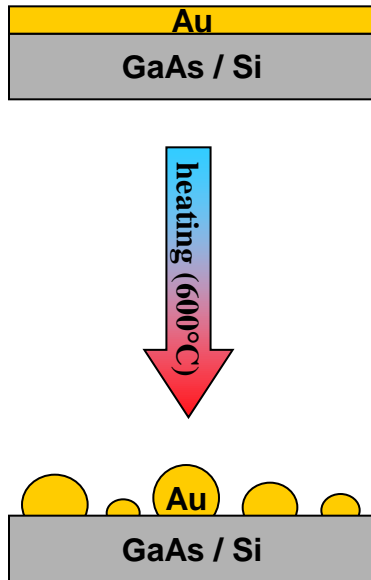
G. Mussler (EuroMBE 2009, Zakopane)  
X-ray lithography + RIE

crystal of Ge QDs



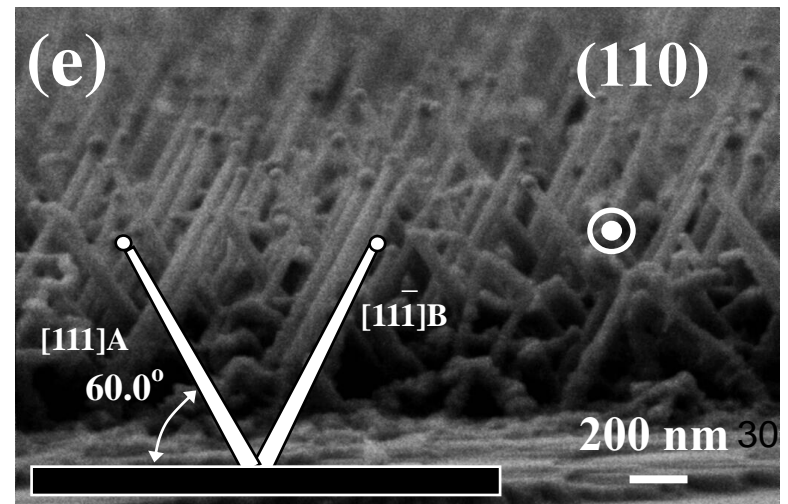
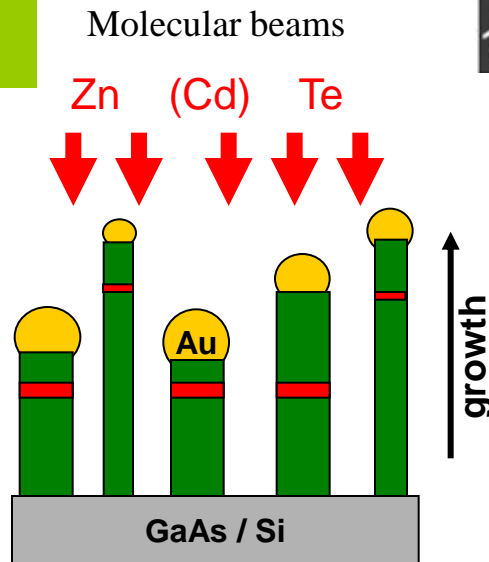
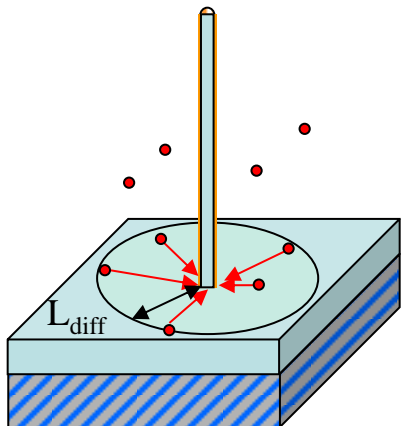
# Application of MBE: self-organized nanowires (NWs)

ZnTe NWs on GaAs



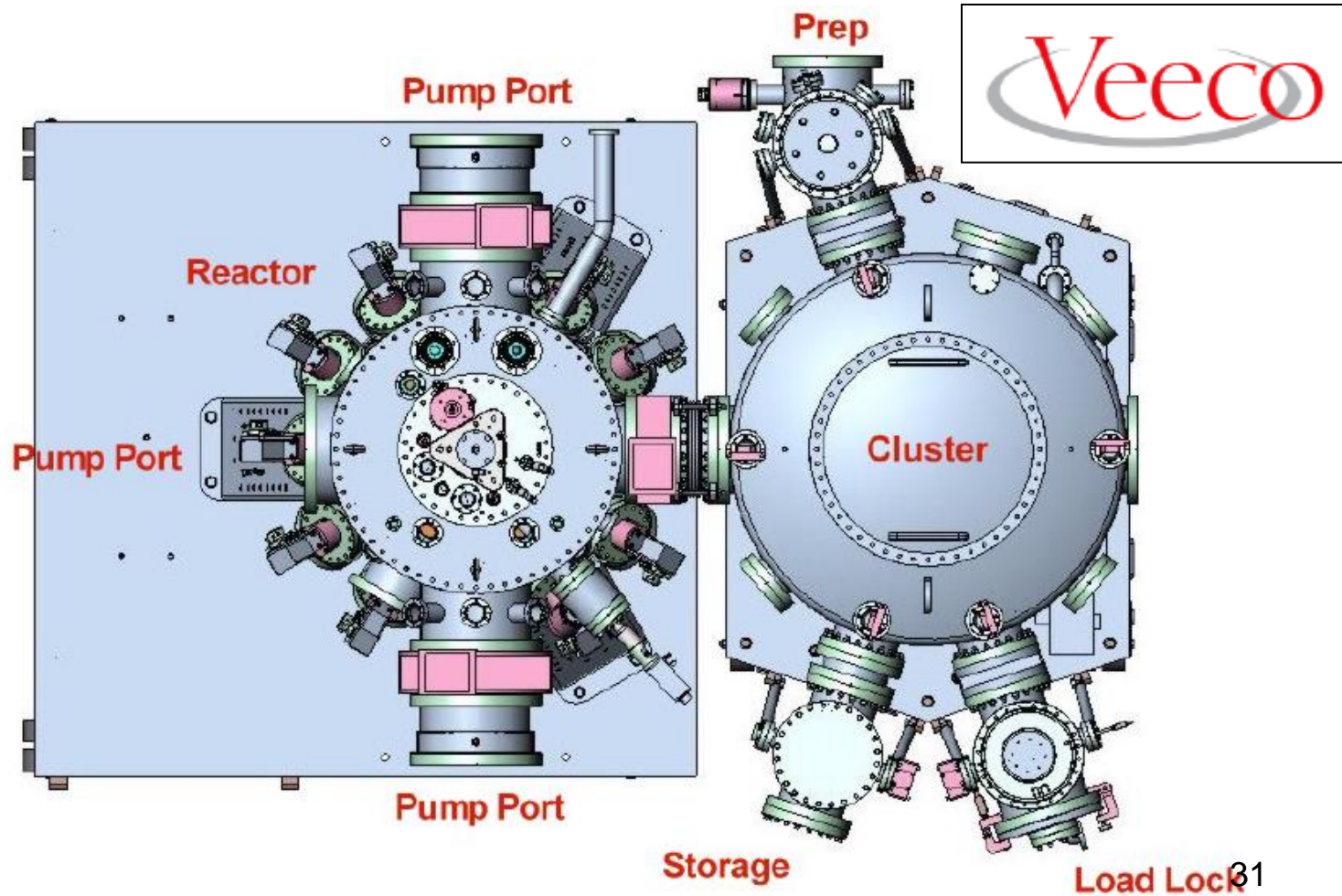
E. Janik, *et al.* APL **89**, 133114 (2006)

growth mode:  
vapor – liquid – solid VLS



## New generation of MBE systems - clusters

- simultaneous growth on large substrates; multi-wafer substrate holders; to increase yield
- min 12 source ports + extra ports for surface analysis tools
- cluster design – independent growth and surface preparation (plasma etching, metallization, ...) chambers
- additional chambers for surface analysis (STM, ...); sample transfer under UHV

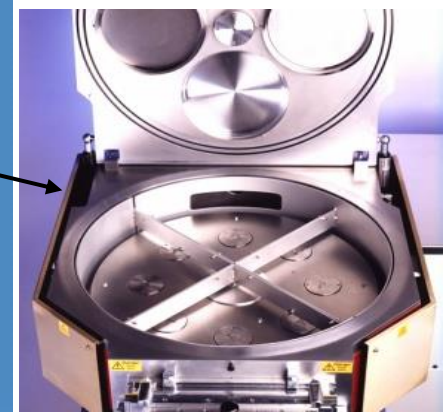
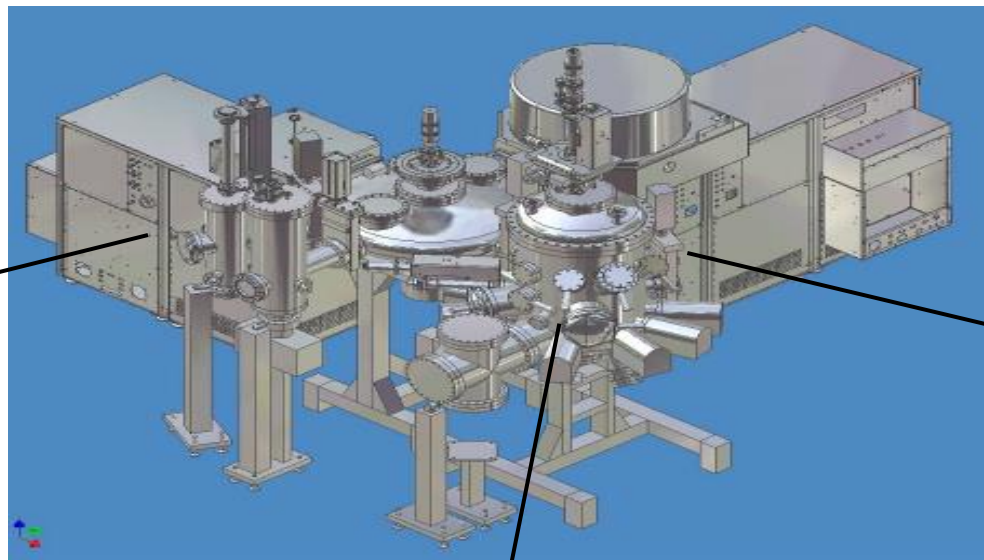




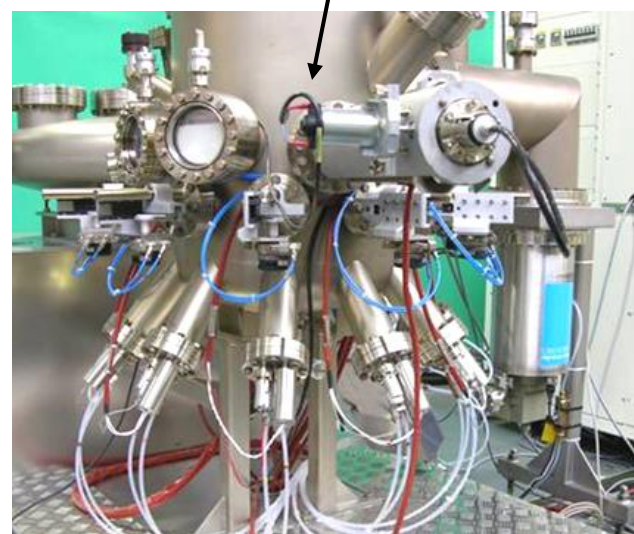
# New generation of MBE systems - clusters



Etch Module (ICP)  
for Clusterlab 600



Deposition Module  
(RF Magnetron Sputter)  
for Clusterlab 600



Epitaxial Growth Module  
(MBE V60) for  
Clusterlab600

# Summary

## advantages of MBE:

- high purity of epilayers
- precise growth control
- perfect for fabrication of low-dimensional structures; sharp interfaces
- wide range of *in situ* growth monitoring tools; important for R&D
- large variety of materials/compounds that could be grown
- strongly nonequilibrium growth technique; solubility limit can be overcome

## disadvantages of MBE:

- measurement of REAL substrate surface temperature difficult
- high costs (purchase, installation and every-day use)
- high failure rate (typical for complicated UHV systems)
  - Most Broken Equipment
  - Multi Bucks Evaporator .....
- selective area growth (SAG) difficult

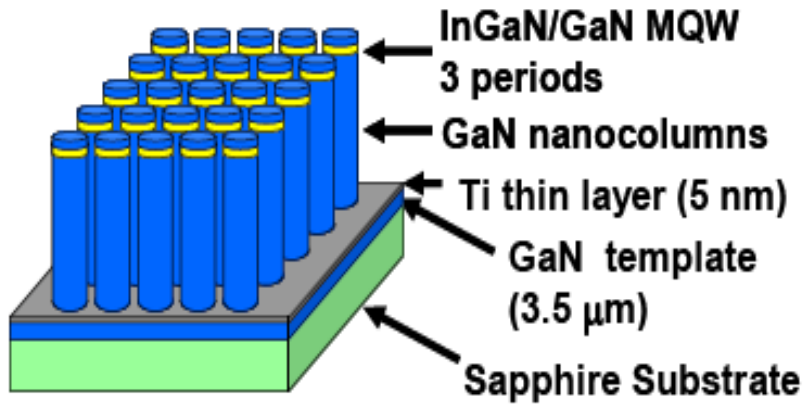


## For further reading

- 1) *M.A. Herman, H. Sitter "Molecular Beam Epitaxy, Fundamentals and Current Status", Springer, 1996*
- 2) *ed. A. Cho "Molecular Beam Epitaxy", AIP, 1994*
- 3) *review papers by T. Foxon; B.A. Joyce; etc.*

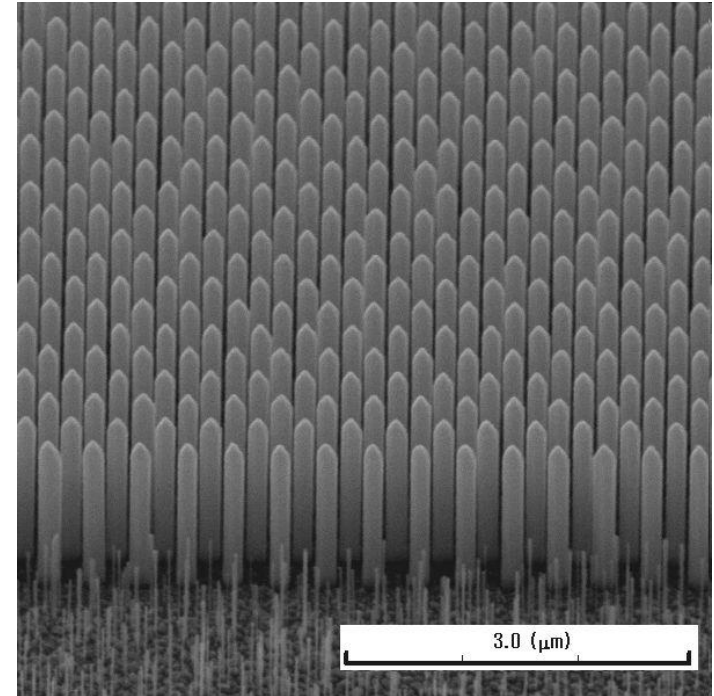
# Przykładowe wykorzystanie MBE: uporządkowane NWs (białe nanoLEDs)

H. Sekiguchi et al., IWNS 2008 Montreux, Switzerland

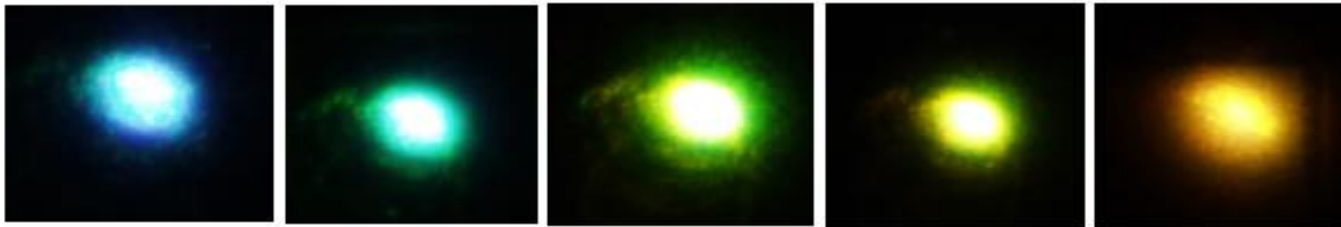


nanodziurki o różnych  
średnicach w masce Ti

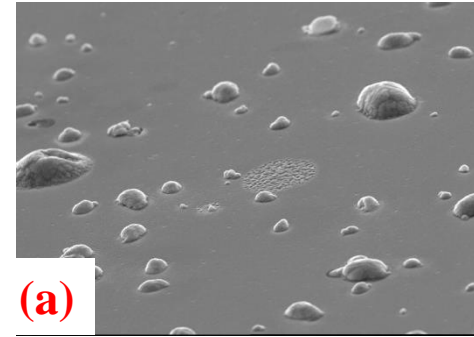
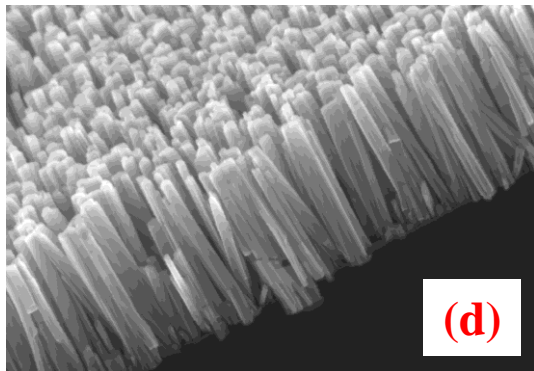
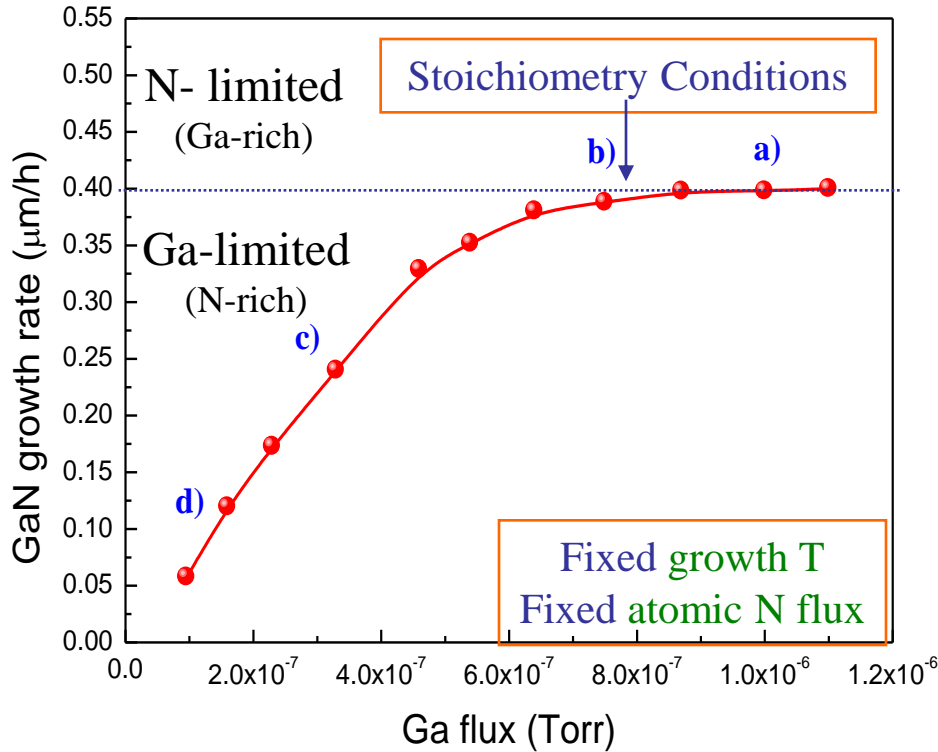
nanodziurki porządkują  
położenie kolumn



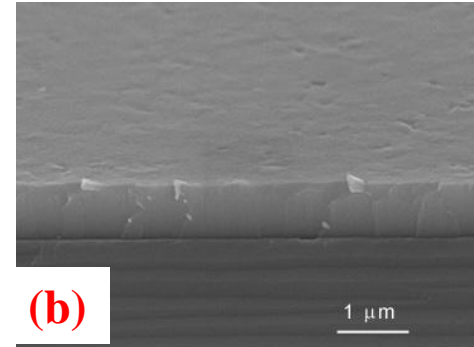
średnica nanodziurki  $\Rightarrow$  średnica nanokolumny  $\Rightarrow$  długość fali emitowanego światła



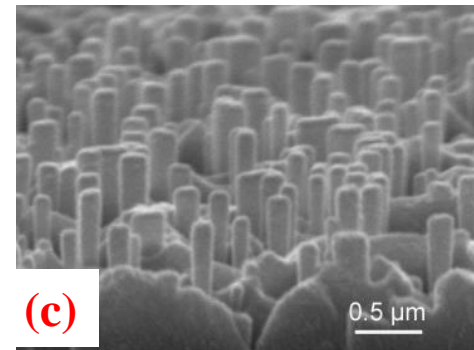
emisja z nanokolumn InGaN/GaN wzrastających na tej samej płytce  
z różnym wzorem nanodziurek w masce Ti



III/V > 1



III/V ≈ 1

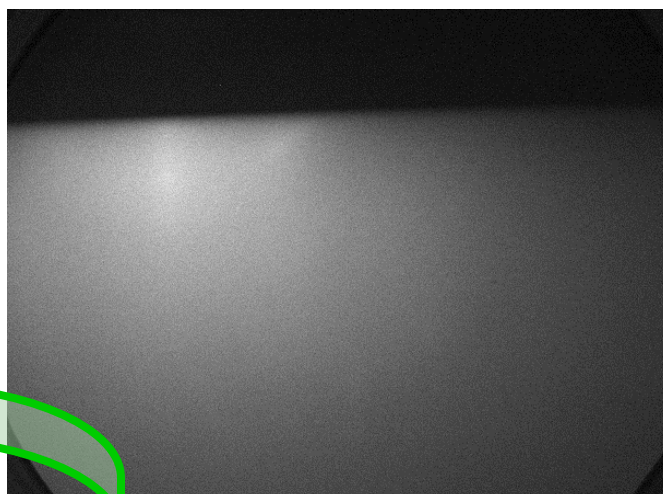
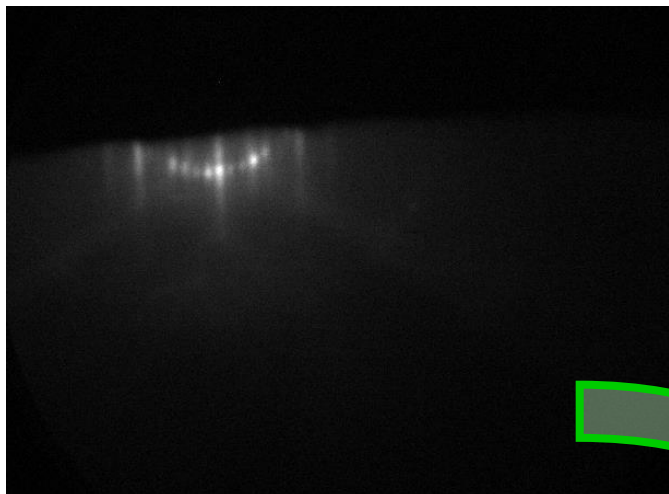


III/V < 1

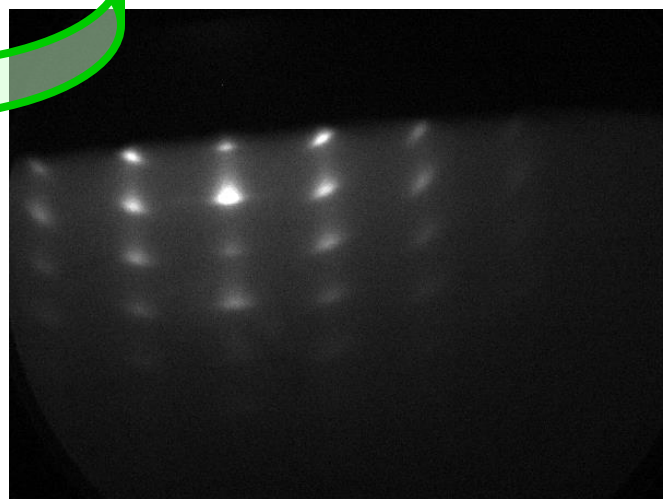
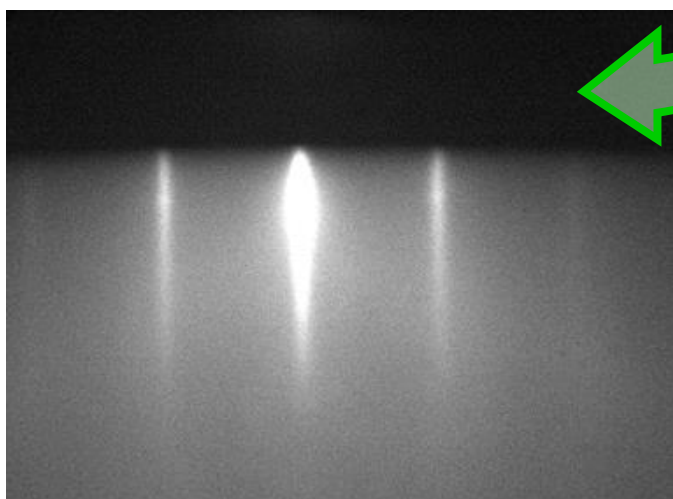
zmieniając stosunek III/V zmieniamy mod wzrostu



clean Si  
(7 × 7)

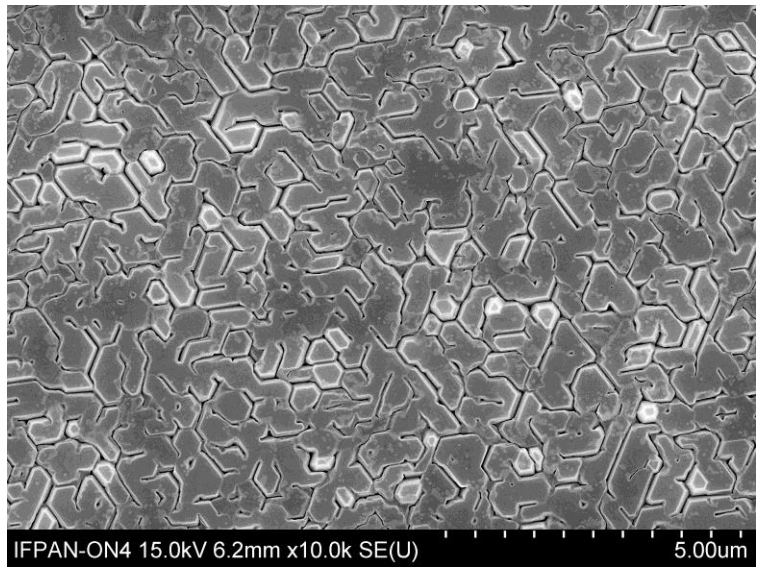
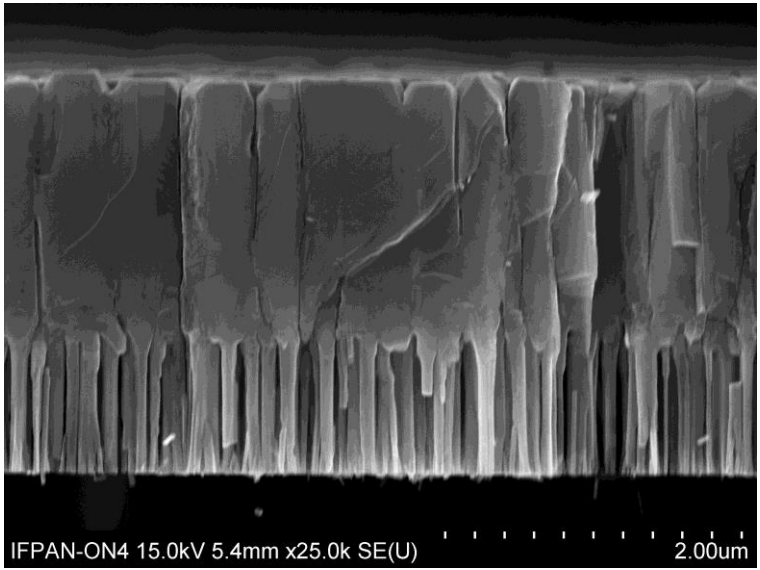


$T_{Ga} \uparrow$   
 $T_{sub} \downarrow$   
(Ga-rich)



GaN NWs  
growth  
 $T_{Ga} = 800C$   
 $T_{sub} = 740C$

planaryzacja powierzchni



ważne:

- ✓ dobra jednorodność długości drutów
- ✓ wysoka koherencja twistu NWs
- ✓ potrzebna wysoka lateralna prędkość wzrostu (Mg doping?)
- ✓ łatwiejsze zarastanie w MOVPE lub HVPE